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I. KAMENICHNY

SHORT HANDBOOK
OF
HEAT
TREATMENT

I. KAMENICHNY

A SHORT HANDBOOK OF HEAT TREATMENT

PEACE PUBLISHERS
Moscow

TRANSLATED FROM THE RUSSIAN BY I. S A V I N DESIDNED BY V. N O V O S E L O V A

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PREFACE

The present handbook aims at giving the operating personnel of heat-treatment shops a reference book which would serve to overcome difficulties arising in everyday practice.

The handbook describes heat-treatment charts, heat-treatment procedures for tools, parts of lathes, automobiles, and agricultural machinery.

In addition the handbook deals briefly with the following subjects: quality control of metals before and after heat-treatment; heat-treatment and case-hardening of steel; heat-treatment of casi iron, non-ferrous metals, and alloys. The book also contains some practical hints on safety engineering.

Chapter I

PHYSICAL CHARACTERISTICS

1. MECHANICAL CHARACTERISTICS

Term	Definition
Deformation	Change of shape or dimensions of a body without alteration of its mass
Elastic deformation	Deformation of a body which occurs when a force is applied, and which disappears upon removal of the force
Permanent deformation	The part of the deformation which remains after the stress is removed
Strength	Ability of a material to withstand stresses without rupturing
Elasticity	Property of a material to recover its origi- nal configuration when the forces are removed
Ductility	Property of a material to withstand considerable permanent deformations without fracturing
Brittleness	Property of a material to fracture without noticeable plastic deformation
Resilience	Work required for rupturing a sample
Hardness	Property of a material to resist indentation by some other body
Fatigue	Development and propagation of cracks in a material as a result of a great number of repeated alternating stresses
Endurance	Property of a material to withstand rup- turing by fatigue stresses
Стеер	Property of a material to flow gradually and continuously under constant stress, especially at elevated temperatures
Wear	Gradual reduction of the size of a part at a result of friction

Term	Definition				
Wear resistance Strain hardening Limit of proportionality σ _p , kg/mm² Elastic limit σ _e , kg/mm² Yield point σ _s , kg/mm² Tensile strength, σ _b kg/mm² Elongation δ, in % Relative reduction in area ψ, in % Specific resilience α _k , km/cm² Endurance limit σ _w , kg/mm² Creep limit σ _c , kg/mm²	Property of a material to resist wear Change in properties and structure of a metal due to deformation Highest stress prior to which deformation increases proportionally to the load applied Stresses at which permanent elongation is negligible (up to 0.02%) Least strain at which the sample flows without noticeable increase in stress Highest stress preceding fracture of a sample The ratio of the increase in length of a sample, produced by tensile stresses, to its original length The ratio of the reduction of area at the point of fracture of a sample to its original cross-section area The ratio of the work required to rupture a sample to the cross-sectional area at the point of fracture Highest stress which a metal can withstand without evidence of fatigue failure Highest stress at which the rate of creep or total creep, in a given time interval, does not exceed a specified value				
	^t er − − − − − − − − − − − − − − − − − − −				

2 THERMAL CHARACTERISTICS

Term	Definition			
<u> </u>				
Calorie (cal)	The amount of heat required to raise the temperature of 1 gram of water 1°C			
Kilocalorie (kcal)	The amount of heat required to raise the temperature of I kilogram of water 1°C			
Heat capacity C, cal/°C	The amount of heat required to raise the temperature of a material I°C			
Specific heat c, cal/g degrees C	The amount of heat required to raise the temperature of 1 gram of a material 1°C			

Term	Definition
Heat conductivity	The ability of a body to conduct heat from one part to the other, as well as to an adjacent body when in physical contact with the latter
Convection	Transfer of heat by the flow of a heated agent (air, water, etc.)
Heat radiation	Transfer of heat by heat rays emitted by an incandescent (hot) body
Calorific power Q, kcal/kg	The amount of heat generated by burning I kg of a fuel
Temperature t Absolute zero	The thermal condition of a body Temperature equal to —273.2°C
Thermodynamic temperature scale, °K	Temperature equal to —270.2 C Temperature expressed in terms of degrees Centigrade and referred to absolute zero as a standard
Coefficient of linear expansion, mm/m degrees C or cm/cm degrees C	Increment of length per unit of length for a temperature rise of 1°C

3. ELECTRICAL CHARACTERISTICS

Term	Definition			
Electric current	The flow of electrons along a conductor			
Current I,	The quantity of electricity passing through			
_ amperes _	a conductor in 1 second			
Resistance R, ohms	The ability of a material to impede the flow of electric current			
Resistivity e, ohm·mm²/m	The resistance of a conductor 1 m in length, 1 mm ² in cross-section, at 20°C			
Voltage E, volts	The difference of electrical potential across the terminals of a conductor			
Power P, watts	The work performed when I ampere flows through a potential difference of 1 volt			
	monga a potential difference of 1 4415			
	·			

4. SOME PROPERTIES OF ELEMENTS AND MATERIALS DEALT WITH IN HEAT-TREATMENT OF METALS

Materia!	Formula	Specific gravity	Melting or fusing point, °C	
Aluminium Ammonium chloride Antimony Asbestos Barium Barium chloride Beryllium Birch Bismuth Bone Borax (anhydrous) Boron	Solids Al NH ₄ Cl Sb — Ba BaCl ₂ Be — Bi — Na ₂ B ₄ O ₇ B	2.7 1.53 6.68 2.4-2.55 3.86 1.82 0.51-0.77 9.8 1.7 2.37 2.32	658 — 630 1480-1510 850 962 1350 — 271 — 711 2300	
Brass Bronze, aluminium Bronze, phosphorus Bronze, tin Brick Cadmium Calcium Calcium carbide Caliche (sodium nitrate) Carboloys, BK type Carboloys, TK type Charcoal, lumps Charcoal, crushed Chromium Clay, dry Coal Cobalt Coke, lumps Coke, crushed Copper Corundum Diamond Duralumin Fireclay (chamotte) German silver Glass Gold Graphite Grey cast iron Ice at 0°C	Cd Ca CaC2 NaNOs — — — — — — — — — — — — — — — — — — —	8.1-8.6 7.7 8.8 8.7 1.4-2.0 8.64 1.55 2.22 2.26 14.3-14.9 9.5-11.0 0.4 1.4-1.5 6.92 1.8 1.2-1.5 8.9 0.6 1.25-1.4 8.92 3.9-4.0 3.51 2.6-2.8 1.8-2.2 8.5 2.4-2.6 19.3 2.25 7.0-7.2 0.917	880-910 950-970 321 850 2300 308 1615 1490 1083 About 2050 3500 1100 1063 1063	

Material	Formula	Specific gravity	Melting or fusing point, °C	
Iridium Iron Iron Iron carbide Lead Magnesium Malleable cast iron Manganese Molybdenum Nickel Niobium Oak Potassium nitrate (niter) Phosphorus, yellow Pine Platinum Potassium	Ir Fe Fe,C Pb Mg Mn Mo Ni Nb KNO, P	22.42 7.86 7.4 11.34 1.74 7.2-7.6 7.2 10.2 8.9 8.4 0.7-1.0 2.11 1.82 0.31-0.76 21.45 0.86 1.52	2450 1535 — 327 651 — 1260 2625 1452 2500 — 333 44 — 1774 62 634	
Potassium cyanide Potassium ferricyanide Potassium ferrocyanide Potassium hydrate Rhodium Salt, common (sodium chloride) Sand, dry Silicon Silicon carbide	KCN K ₄ Fe (CN) ₆ KOH Rh NaCl Si	1.32 1.93 1.85 2.04 12.5 2.16 1.4-1.6	360 1966 800 — 1420	
(carborundum) Silver Soda ash Soda, caustic Sodium Sodium cyanide Steel, carbon Steel, grade P9 Steel, grade P18 Sulphur Tellurium Tin Titanium Tungsten Tungsten carbide	SiC Ag Na ₂ CO ₃ NaOH Na NaCH — S Te Sn Ti W WC	3.17 10.5 2.53 2.13 0.97 	2700 min 961 851 318 97 564 — 115-119 452 232 1800 3370 2777 1720	
Vanadium Wood, air-dry Zine	Zn	7.14	419	

Material	Formula	Specific gravity	Melting of fusing point, °C
	Liquids		
Benzene	1 -	0.88	
Gasoline	-	0.7-0.75	_
Glycerine	i –	1.26	
Hydrochloric acid	HCI	1.19	-
Kerosene	_	0.8-0.82	_
Machine oil	-	About 0.9	-
Mercury	Hg HNO,	13.55	39
Nitric acid	HNO ₃	1.5	_
Petroleum	_	0.89 max	_
Sulphuric acid	H ₂ SO ₄ H ₂ O	1.84	_
Water at 4°C	H ₂ O	1.0	_

 $\it Note.$ Specific gravities and melting points of alloys are approximate and may vary depending on the composition.

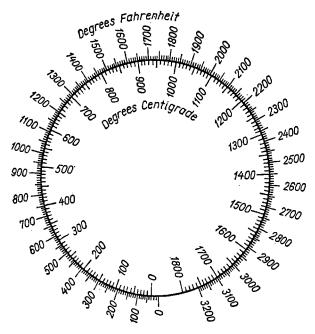


Fig. 1. Conversion of degrees Centigrade to degrees Fahrenheit

Chapter II

QUALITY CONTROL OF METALS

1. HARDNESS TESTS

Hardness may be measured by various methods, depending on the condition of the metal (hardened, annealed, etc.), and on the size of the specimen.

Brinell hardness test. This method is chiefly used to test rolled and forged pieces, castings, as well as dies and tools whose hardness does not exceed 450 units.

The Brinell hardness number (abbreviated to H_B and expressed in kg/mm²) is determined by forcing a hardened ball 10.5 or 2.5 mm in diameter into the metal being tested under a load of 15.6 to 3,000 kg in a special press (see Table 1).

- \cdot When testing specimens for hardness the following conditions are to be ensured:
- 1. The point of indentation should be free from scale and/or decarburised layers.
- 2. Rounded surfaces should be so machined as to present a plane to the indentor (a flat should be filed).
- 3. If the bottom or lateral walls are strained following indentation, the test should be repeated by the use of a ball of a smaller diameter and an appropriate load.
- 4. This method cannot be used to determine the hardness of carburised or nitrided surfaces.
- 5. The centre of the impression should be situated at least one ball diameter from the edge of the specimen, and at least two diameters from an adjacent impression.
- The diameter of the impression is measured by means of a special magnifying glass, and the hardness number is then read in the table.

When testing for hardness with a 5-mm ball the actual diameter of the impression is to be multiplied by 2; when a 2.5-mm ball is used, the result is to be multiplied by 4.

Example:

Ball diameter	5 mm	2.5 mm
Load	750 kg	187.5 kg
Impression diameter		
Multiplication	$.8\times2=$	$1.1\times4=$
=	3.6 mm =	=4.4 mm
These diameters correspond to the		
following hardness numbers (see	000	107
Table 2)	286	18/

Table 1

Specimen Thickness and Load Application Time in Brinell
Hardness Test

Material	Hardness number H	Specimen thickness, mm	Ball diameter, mm	Load P, kg	Load application time, sec
Ferrous metals	150-450	Over 6 3-6 Below 3	10 5 2.5	3,000 750 187.5	10 10 10
retrous metals	Below 150	Over 6 3-6 Below 3	10 5 2.5	1,000 250 62.5	10 10 10
Copper, brass, bronze, magne- sium- and alumin- ium-base alloys	35-130	Over 6 3-6 Below 3	10 5 2.5	1,000 250 62.5	30 30 30
Aluminium, anti- friction alloys	8-35	Over 6 3-6 Below 3	10 5 2.5	250 62.5 15.6	60 60 60

 ${\it Table~2} \\ {\it Conversion~of~Hardness~Numbers~and~Their~Correlation~with} \\ {\it Tensile~Strength} \\$

Brinell ness n		Rockwo	ell har umber	dness	yramid umber,	umber,	Tens	ile stren kg/mm²	gth.
Impres- sion dia., mm	Hard- ness number		Scale		Diamond pyramid hardness number, H_V	Shore scleroscope hardness number, H _S	Carbon steel	Chromium steel	Nickel and chrome nickel steel
lmp slon	# 2 2	_ c	A	В	- ZEE	325	ا ين ا	0.4%	Z 5 E \$
2.2 2.25 2.35 2.45 2.55 2.65 2.75 2.80 2.95 3.00 3.02 3.15 3.25 3.35 3.40 3.45 3.55 3.55 3.65 3.75 3.75	780 745 712 682 653 627 601 578 555 534 495 477 461 444 429 415 409 415 338 375 363 352 341 331 321 311 321 321 321 321 321 321 32	72 70 68 66 62 60 55 50 48 46 54 43 42 41 40 39 33 33 31 30 29 82 27	89 87 86 85 84 83 82 81 79 77 76 77 77 71 70 69 68 67 67 67 67 67 67 67 67 67 67 67 67 67		1,224 1,116 1,022 941 868 804 746 694 606 587 551 534 450 435 423 410 390 380 361 344 335 320 312 305 278 272 261	90 86 84 81 78 75 71 69 66 65 64 61 59 58 57 55 52 51 50 49 48 47 40 39	178 172 165 160 155 149 147 144 139.5 120.5 122.5 111.5 103.0 99.5 97.0 94.5	173 167 156 150 145 143 139.5 131.5 127 123.0 119.0 116.5 102.5 102.5 102.5 102.5 102.5	168 161 146 141 139 136.5 1127.5 1123.5 119.5 116.0 109.0 105.5 109.0 105.5 100.0 97.5 94.0 91.5 89.5

	l hard- number	Rock	well i	nardness	yramid umber,	oscope umber,	Ter	nsile strei kg mm²	ngth,
Impres- sion dia,	Hard- ness number	C .	Sca!	В	Diamond pyramid hardness number,	Siore scleroscope hardness number, H _S	Carbon steel	Chromium steel	Nickel and chrome cnickel steel
4.30 4.35 4.40 4.45 4.50 4.55 4.60 4.65 4.75 4.80 4.85 4.90	255 248 241 235 228 223 217 212 207 202 196 192 187 174 170 166 163 159 156 153 149 146 143 140 137 134 131 128 124 121 118 118 119 119 119 119 119 119 119 11	26 25 24 23 22 21 20 19 18 16 15 15 	643 633 622 661 660 558 557 566 55 5 5 5 5 5 5 5 5 5 5 5 5 5 5		255 250 240 235 226 221 217 213 209 201 197 190 186 183 177 174 171 165 162 159 154 157 117 111	38 37 36 35 34 33 32 32 31 31 30 	92.5 89.5 87.0 84.5 80.0 74.5 66.0 66.5 66.0 66.5 57.5 66.0 60.5 55.5 55.5 55.5 55.5 55.5 55	89.0 87.0 84.5 82.5 80.5 776.0 72.5 63.5 65.5 65.5 65.5 65.5 55.5 55.5 55	86.5 84.5 82.0 80.5 776.5 66.5 65.5 66.5 65.5 66.5 57.5 57

î,

Brinell ness n	hard- umber	Rockw	ell ha	rdness er	pyramid number,	scleroscope ss number,	Tens	Tensile strengt			
Impression dia.,	Hard- ness number								uou	Chromium steel	Nickel and chro- me-nickel steel
Imp sion	Hard ness numb	С	A	В	Diar hard H	Shore hardne H _S	Carbon steel	Chro	Nick and me-n steel		
5.70 5.75 5.80 5.85 5.90 5.95 6.00 6.10 6.20 6.36 6.48 6.56	107 105 103 101 99 97 96 92 88 84 80 78	11111111111	11111111111	59 58 57 56 54 53 52 49.5 47 43.5 40.5 38.5			38.5 38.0 37.0 36.5 35.5 35.0 34.5 33.0 30.0 29.0 28.0				

Note. Brinell hardness numbers were obtained with a 3,000 kg load and a 10-mm ball.

Rockwell hardness test. This method is used to determine the hardness numbers of metals (abbreviated to R_C , R_A and R_B) by forcing into them a conical diamond with rounded point, called the penetrator, under a load of 150 or 60 kg or a 1.59-mm steel ball to which a 100 kg load is applied. Test conditions are described in Table 3.

Rockwell Hardness Testing Conditions

KOCK WEIT 112	ii uiiess	restring C	onu	110112		
Metal	Approximate Brinell hardness number	Type of penetrator	Load, kg	Rockwell hardness number designation	Working range of scale	Scale
Annealed steel, brass, bronze, aluminium and magnesium hard alloys	60-230	Steel ball, dia. 1.59 mm	100	R_B	2 5- 100	Red
	230- 700	Diamond cone	150	R_{C}	20-67	Black
3. Thin plates, case-carbu- rised and nitrated arti- cles, carboloys		Ditto	60	R_A	Over 67	Black

- On testing for hardness the following requirements should be met:
 1. The piece being tested should be free from scale, decarburised layers, dents, grease or traces of machining.
- 2. Hollow thin-walled specimens should not be tested since the results would be wrong (because the piece will spring back or cave in).
- 3. For the same reason no pads should be inserted between the piece being tested and the anvil of the testing machine.
- 4. The piece to be tested should be thick enough to prevent bulging on the reverse side.
- 5. The centre of the impression should be spaced from the edge of the specimen not less than 2.5 mm for scales A and C, and not less than 4 mm for scale B.
- 6. Round-shaped specimens being tested for hardness by means of a diamond cone should be no less than 15 mm in diameter. When testing pieces of smaller diameters Correction Tables 4 and 5 should be consulted.

Table 4

Correction Values for Hardness Numbers Measured on Cylindrical Surfaces of Specimens by a TK Testing Machine (Rockwell Test)

Diameter of	Hardness of cylindrical surface (R_C)									
specimen, mm	58	59	62	63						
	Values to be added to the hardness number obtained									
6 7 8 9 10 11 12 13 14 and 15	2.5 2.0 2.0 — — — —	2.0 2.0 1.5 1.5 1.0 1.0 1.0	2.0 1.5 1.5 1.0 1.0 1.0 1.0	2.0 1.5 1.5 1.0 1.0 1.0 0.5	2.0 1.5 1.5 1.0 1.0 1.0 0.5 0.5	1.0 1.0 1.0 0.5 0.5				

Table 5
Correction Values for Hardness Numbers as Measured on Spherical Surfaces by a TK Testing Machine (Rockwell Test)

	1	Ha	rdness	of st	heric	al sur	face,	R .	
Ball diameter, mm	56	57	58	59	60	18	62	63	64
	Va	lues	to be		i to ti btaine		rdness	num	er
Up to 4 From 4 to 6		5.5 4.5		5.0	4.5 3.5		3.5	_	_
, 6 to 8.5	4.5	3.5	3.5 2.5	3.0 2.5	$\frac{3.3}{3.0}$	3.0	2.5	1.5	_
" 11.5 to 15.0		_	$\tilde{2}.\tilde{0}$	1.5	1.5	1.0	1.0	1.0	0.5

Table 6

Conversion of Rockwell Hardness Numbers Measured by the Use of Diamond Cone under Loads of 150 kg (Scale C) and 15, 30 and 45 kg (Microhardness Scale)

30 and 45 kg (Microllandiess Scale)										
Rockwell hardness number, R _C	Microhar under	dness nu loads of,		Rockwell hardness number, R _C	Microhardness number under loads of, kg					
Roc har nun R _C	15N	30N	45N	Roc har num R _C	15 N	30N	45N			
68 67 66 65 64 63 62 61 60 59 57 56 55 54 55 51 50 49 48 47 46 45 44	93.2 92.9 92.5 92.5 91.8 91.1 90.7 90.2 89.8 89.3 88.9 87.9 87.4 86.4 85.9 85.5 85.5 85.5 83.5 83.5 83.5 83.5	84.4 83.6 82.8 81.9 81.1 79.3 78.4 77.5 76.6 75.7 74.8 73.9 73.0 72.0 70.2 69.4 68.5 67.6 65.8 64.8 64.0 63.1	75.4 74.2 73.3 72.0 69.9 68.8 67.7 66.6 564.3 63.2 62.0 60.9 59.8 65.5 57.4 55.0 53.8 52.5 4 50.3 49.0 47.8	43 42 41 40 39 38 37 36 35 34 33 32 31 30 29 28 27 26 25 24 23 22 21 20	82.0 81.5 80.9 80.4 79.9 79.4 78.8 78.3 77.7 77.2 76.6 75.0 74.5 73.3 72.8 72.2 71.6 69.9 69.4	62.2 61.3 60.4 59.5 58.6 57.7 56.8 55.9 55.0 54.2 53.3 50.4 49.5 44.0 43.2 41.5	46.7 45.5 44.3 43.1 41.9 40.8 39.6 38.4 37.2 36.1 34.9 33.7 32.5 31.3 22.5 27.8 26.7 25.5 24.3 23.1 22.0 20.7 19.6			

To determine the hardness of thin layer of metals, a special testing machine "Super-Rockwell" is used.

The test is carried out by means of a diamond cone or a 1.59-mm ball

under loads of 15, 30 and 45 kg (Tables 6 and 7).

Hardness test by a diamond pyramid penetrator, H_V in kg/mm². The hardness of metals is determined by indenting a tetrahedral diamond pyramid penetrator under loads of 5, 10, 20, 30, 50, 100 and 120 kg. Loads should be selected by referring to Tables 8 and 9.

The diamond pyramid can be used to test soft, as well as heat-treated metals. The diamond pyramid test is widely used to determine the hardness of thin specimens and articles with hard superficial layers.

Table 7 Conversion of Rockwell Hardness Numbers Determined by the Use of a Steel Ball under Loads of 100 kg (Scale B) and 15, 30 and 45 kg (Microhardness Scale)

			•			_	
Rockwell hardness number, R	Microhar under i	dness nu oads of,		Rockwell hardness number, R		nardness num r loads of,	
Ro han R	15T	30T	45T	Ro han R	15 T	30T	45T
100 99 98 97 96 95 94 93 92 91 90 89 88 87 86 85 84 83 82 81 80 77 76	93.0 92.5 92.5 92.0 91.5 91.0 90.5 90.0 89.5 89.5 89.0 88.5 88.0 87.5 87.0 86.5 86.0 86.0	83.0 82.5 81.0 80.5 80.0 79.0 78.5 78.0 77.0 75.0 74.5 74.0 73.5 73.0 72.0 71.5 70.0 69.0 68.5 68.0	73.0 72.0 71.0 69.0 66.0 66.0 65.0 62.0 60.0 59.5 58.5 57.5 54.5 53.5 52.5 52.0 50.0	75 74 73 72 71 769 68 67 66 63 62 61 59 85 57 55 49 48 47 42 40	85.5 85.0 84.5 84.0 83.5 83.5 82.0 82.5 82.0 81.5 81.0 80.0 79.5 79.0 79.5 77.5 77.5 77.5 75.5	67.5 66.5 66.0 65.5 63.0 63.5 63.0 61.5 60.5 59.0 58.0 57.0 55.5 55.0 55.0 50.0 50.0 50.0 50	49.0 48.5 47.5 46.5 44.5 42.5 42.0 41.0 40.0 38.0 37.0 36.0 35.0 32.5 33.0 32.5 28.5 28.0 21.0 21.0 18.0 16.0

 ${\it Table~8} \\ {\it Recommended~Loads~for~Diamond~Pyramid~Hardness~Testing}$

Thickness of	Hardness number H_{V}									
specimen, mm	20.50	50-100	100-300	300-900						
		Recommer	ided loads, kg							
0.3-0.5 0.5-1.0 1.0-2.0 2.0-4.0 Over 4	5-10 10-12 20 and higher	 5-10 20-30 30 and higher	— . 10-20 20-50 50 and higher	5-10 10-20 — 20-50						

Table 9

Recommended Loads for Diamond Pyramid Hardness Testing
of Tubes and Various Thin-walled Articles

		Wa	ll thick	ness of t	ube or	article,	mm				
Outer diameter of tube or	0.5	0.75	1.0	1.25	1.5	2.0	2.5	3.0			
article, mm	Recommended loads, kg										
0-10 10-20 20-30 30-40 40-50 50-60 60-70 70-80	10 5 	10 5 5 -	20 20 10 10 5 5	- - - 10 10 5 5	30 30 20 20 20 20 10	30 30 30 30 30 20 20	30 30 30 30 30 30 30	30 30 30 30 30 30 30 30			

The surface to be tested should be dry, clean, free from pores, scale and decarburised layers and bear no rough traces of machining. The diagonal of the impression is measured with a microscope attached to the testing machine; the hardness number is determined by referring to a special table.

The $T\Pi$ testing machine can be used for Brinell hardness testing with 5- and 2.5-mm balls, which are part of the machine, under loads of 62.5 and 15.6 kg.

Shore scleroscope hardness test. A well-finished part (specimen) is set in the machine and an indentor is dropped onto the surface from a certain height. Upon rebounding the indentor deflects the pointer on a dial, thus indicating the hardness number of the part tested.

Shore hardness number is abbreviated to H_{S} .

Approximate hardness testing by ball impact indentation. The procedure employs a portable testing machine intended for the approximate hardness testing of bulky articles. This is accomplished by pressing a ball, with an impact of medium force, simultaneously into the article being tested and into the standard specimen; the hardness number is then determined by referring to a special table attached to the testing machine.

The hardness number in the table is found at the intersection of the columns giving the impression diameters of the article tested and the standard specimen. The precision of the method is within $\pm 7\%$.

Filing hardness test. Tools and articles (such as taps, reamers, drills, small-size tools) whose hardness cannot be determined with testing machines are tested by filing with smooth files and by comparing the abrasion with that produced on standard specimens. The latter consist of a set of quenched rings of varying hardness ranging from 45 to $63\ R_C$ with gradations every 3-5 units. The hardness number is determined with adequate precision by alternately filing the article and the standard specimen and by comparing the forces applied.

2. Detection of Cracks in Metals

The most widely used methods of finding surface cracks are:

(a) magnetic-particle crack detection (applicable only to steel and cast iron);

(b) deep etching;

(c) kerosene or hot oil tests or blacking-in;

(d) fluoroscopic crack detection.

The method of magnetic-particle crack detection is performed as follows: the article under test is magnetised in the crack detector, after which it is dipped into a bath or flushed with a suspension of magnetic particles. The magnetic particles will settle on the cracks, if there are any. After the test, the article is demagnetised.

The first step in the preparation of the magnetic-particle powder is to thoroughly mix finely crushed ferric oxide or ochre with kerosene or oil to a paste of medium consistency. The paste is then packed into a crucible or tube, which should be tightly closed, luted and held in a furnace at 600-800°C until the kerosene is completely burned, after which the paste is allowed to cool down to room temperature. The resultant magnetic powder is thoroughly ground with a small amount of kerosene or oil to a thin paste, after which kerosene or oil is added to obtain 1 litre of suspension for every 25-40 grams of powder.

A non-corrosive liquid designated KMO is used as an aqueous suspension of magnetic particles to reveal very fine cracks; it has the following composition:

denaturated alcohol				
water			150	millilitres
sodium hydroxide NaOH			40	grams
oleic acid				
naphthenic acid			200	millilitres

The KHO liquid is added in the proportion of 40 millilitres per 1 litre of water, containing 2 grams of soda ash, after which magnetic powder is introduced into the mixture.

Fluoroscopic crack detection. This method is used to reveal cracks in non-magnetic metals, as well as in various other materials (e.g., plastics, etc.). The test is performed by the use of the AIOM-1 testing machine manufactured by medical equipment plants. The piece to be tested is immersed in a special liquid after which it is flushed with water, wiped dry, sprinkled with magnesium oxide (the excess particles being blown off) and then examined in the ultra-violet light. If there are any cracks, the solution contained therein wets the magnesium oxide and the cracks are exposed upon examination in a dark room under the rays of a mercury vapour-quartz lamp.

The liquid used is a mixture of 15% (by volume) of transformer oil

and 85% of kerosene.

The detection of cracks and other faults in steel by the deep etching method requires the use of the etchants listed in Table 10.

The following etchants are used for copper-base alloys:

a) 10-20% aqueous solution of ammonium persulphate;

 b) 10% solution of hydrogen peroxide in saturated aqueous solution of ammonium;

 c) solution of ferric chloride (10 grams) and hydrochloric acid (30 cu cm) in water (120 cu cm).

The latter etchant is also used for nickel-base alloys.

To etch duralumin, the following etchant is used: 16.5% hydrochloric acid, 16.5% nitric acid, 4.5% hydrofluoric acid, and 62.5% water. The etchants act quickly; following etching the specimen should immediately be washed and dried.

Table 10
Etchants Used for Deep Etching of Steel

	Etchar	nt comp (by vo		п %	Etching schedule			
Material etched	Hydro- chloric acid HCl	Nitric acld HNO ₃	Sulphuric acid H ₂ SO ₄	Water	Tempera- ture, °C	Holding time		
Cook or steel			15		00	TT 1 0 1		
Carbon steel .		_	17	83	60	Up to 2 hrs		
Ditto	50	-	-	50	60-70	10-45 min		
Ditto	_	5 0	_	50	70-80	1-2 hrs		
Alloy steel	17	—	33	50	Up to 100	20-60 min		
Ditto	66	_	10	24	95-98	20 min-2 hrs		

Table 11
Characteristics of Grinding Wheels for Spark Testing of Steel

Wheel characteristics	For testing rods, castings and forgings	For testing finished articles		
Wheel diameter, mm Wheel thickness, mm Grain size	300-350 40-60 36-40 Grade Ст.1 steel	150-200 25-40 60-80 Grade Cr.1 steel		

Deep etching is usually applied when testing parts for possible cracks. Since acids tend to dissolve metals, the parts may decrease in size as a result of etching.

Crack detection by blacking-in. The piece is immersed in a solution, then thoroughly washed with cold water, coated with a thin film of aqueous suspension of white clay, and finally dried in an air jet. The solution forced out of the cracks colours them brightly.

Composition of the solution:

kerosene .										volume
transformer	oi	1						30%	_	1)
turpentine								5%		1)

Sudan III, sudan II, sudan I or fatty orange are added as colouring

Kerosene or hot oil testing. The parts are immersed in kerosene or hot oil for 10-20 min, after which they are sand-blasted or wiped dry and rubbed with chalk. The kerosene and oil retained by the cracks make them appear as thin streaks.

Sonic inspection. Cracks can be detected by experienced operators by striking suspended parts. A cracked part gives off a dull sound when hit with a hammer.

3. APPROXIMATE DETERMINATION OF STEEL COMPOSITION

Steeloscope inspection. Spectroscopic analysis is being more and more widely used in heat-treatment shops, as a means of determining the approximate composition of steel. In the spectroscopic method an electric arc is struck between the piece being tested and an electrode. The light of the arc is directed by a lens, into the slit of a special apparatus—the steeloscope—where the spectrum is examined. Most of the components of steel emit characteristic spectral lines on burning. The pattern and intensity of these lines when referred to the special tables accompanying the apparatus, give information as to the approximate content of the various elements in the steel. The procedure takes

only a few minutes and no damage is done to the piece tested. Carbon, sulphur and phosphorus cannot be determined by this method.

Spark test. The determination of the composition of steel by the spark test requires blacking out of the site of the test, adequate grinding

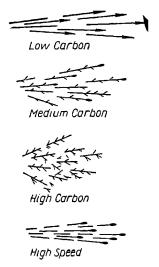


Fig. 2. Pattern of sparks produced by various grades of steels

wheels (Table II), as well as branded specimens of the steels used in the plant. By examining the colour and pattern of the sparks emitted from the part being tested and a standard specimen, one may determine approximately the steel grade.

Low-carbon steel throws off a long beam of light-yellow sparks with no explosions. As the carbon content increases the beam becomes shorter and wider and the number of explosions increases (Fig. 2).

As compared to carbon steel the colour of the sparks produced by chromium steel is darker, with a lesser number of explosions. Steel containing tungsten throws off dark-red sparks.

STEEL

1. CONSTITUENTS OF STEEL

The internal arrangement of steel and other alloys is called their structure. The structure visible only under a microscope is called microstructure. To determine the microstructure of a steel specimen, the latter should be ground, polished, etched and examined with the aid of a microscope. Table 12 indicates the constituents of steel and presents some data on their formation, physical properties, etc.

2. STRUCTURE OF IRON AND STEEL. EQUILIBRIUM DIAGRAM OF IRON-GARBON ALLOYS

The crystal lattice of pure iron changes its structure, i. e., the arrangement of its atoms, with temperature. Various lattice structures are presented in Fig. 3. Data on lattice structure modifications on heating and cooling are given in Table 13.

Fig. 4 presents an equilibrium diagram of iron-carbon alloys. The temperatures at which solid-phase changes occur in iron and its alloys are called critical points (Table 14).

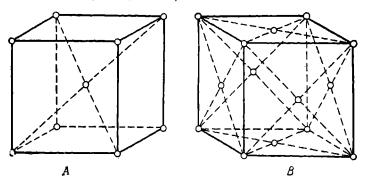


Fig. 3. Structure of crystalline lattice: a-alpha- and delta-iron; b-gamma-iron

Structural Constituents of Iron-carbon Alloys

ounditions C	;
Definition formations of	tion
Solid solution of car- On heating above Ac, Soft, bon and other elements in gammanicon. Carbon con-	a Jint
Solid solution of car- bon and other ele- ments in alpha-iron. Carbon content up to 0.000	au w c hy teel
Chemical combination Rejected by liquid of iron and carbon. and solid solutron carbide Fe ₃ C tions on slow contains 6.67% car-	liq s s
Eutectoid mixture of Formed on decay cementite and fer- of austenite rite	de te
Solid solution of carbon and other elerapid cooling of ments in alpha-iron with distorted latters tice	lling fr criti res

Table 12 continued

Brinell hardness number, H_B	330-400	Harder than troostite	270-320	
Physical properties	Highly dispersed mix- Formed on heating Approx. up ture of ferrite and carbides the 250-400° C range or on slow cooling of aus-	Formed on isother- Up to 500 Hard, fairly ductile, Harder than mal transformation of austerite in the 250-400° C	Up to Ac ₁ Ductile and resilient. Less stronger and harder than troostite Magnetic	Brittle
Stable at tem- peratures, °C	Approx. up	Up to 500		Below 1130
Conditions of formation	Formed on heating of martensite in the 250-400° C range or on slow cooling of aus-	tenite Formed on isothermal transformation of austenite in the 250-400° C	Formed on heating of martensite in the 400° C-Ac, range or on very	<u>щ</u>
Definition	Highly dispersed mix- ture of ferrite and carbides	Difto	Fine mixture of ferrite and cementite	An austenite-cementite eutectic mixture containing 4.3% carbon
Structure	Troostite	Acicular troostite (bainite)	Sorbite	Ledeburite

Notes, 1. Steel structures composed of martensite and troostite, as well as of troostite and sorbite are called troostomartensite and troosto-sorbite structures respectively and have properties intermediate between the two named structures.

2. The word "dispersed" denotes fine-crushed.

Stability Temperature Ranges of Various Crystalline Forms of Pure Iron

On heating		On cooling				
Stable lattice	Temperature range, °C	Stable lattice	Temperature range, °C			
Body-centered cu- bic alpha-iron Face-centered gam- ma-iron Body-centered cu- bic delta-iron	Up to 910 910-1390 1390-1 5 39	Body-centered cu- bic delta-iron Face-centered cu- bic gamma-iron Body-centered cu- bic alpha-iron	1539-1390 1390-898 Below 898			

Notes. 1. Disagreement between transformation temperatures on heating and cooling is called the thermal hysteresis.

2. The difference in the structures of alpha- and detla-iron is attributed only to the distance between the atoms, the mutual arrangement of atoms of both structures being identical.

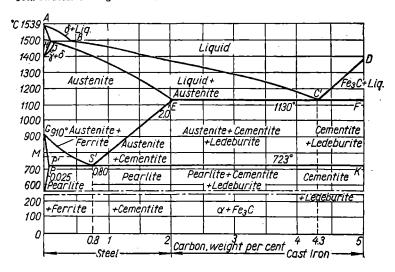


Fig. 4. Iron-carbon equilibrium diagram

Table 14
Principal Phase Transformations in Iron-carbon Alloys on Slow
Heating and Cooling

Line of trans-	Temperature of transfor- mation, °C	Nature of transformation	of cr	nation itical ints
forma- tion	Tempe of trai mation	Nature of transformation	on heatin g	on cooling
PSK	723	Transformation of pearlite to austenite Transformation of austenite to pearlite	Ac ₁	Ar,
МО	768	Loss of magnetism for steels containing approximately up to 0.5% carbon Regaining of magnetism for the same steels	Ac ₂	Ar ₂
GOS	910- 723	The end of the dissolution of ferrite in austenite in hypoeutectoid steels The beginning of the separation of ferrite from austenite in hypoeutectoid steels	Ac,	Ar ₃
SE	723- 1130	The end of the dissolution of cementite in austenite in hypereutectoid steels The beginning of the rejection of cementite from austenite in hypereutectoid steels	Ac _m	Ar _m
IE		The beginning of melting of steel on heating The end of solidification of steel on cooling	_	
ECF	1130	The beginning of melting of cast iron on heating The end of solidification of cast iron on cooling	_	_
ABCD	_	The end of melting of steel and cast iron on heating The beginning of solidification of steel and cast iron on cooling		_

3. Alloying Elements and Their Effect on the Properties of Steel

Alloying elements are added to steels with the aim of imparting specific mechanical and physicochemical properties to the latter (see Tables 15 and 16).

Table 15
Conventional Symbols for Elements Contained in Steels

Element	Designation of ele- ments in tables of chemical composition	Designation of ele- ments in Soviet Stand- ards for marking steel grades
Aluminium Boron Carbon Chromium Cobalt Copper Manganese Molybdenum Nickel Niobium Phosphorus Silicon Sulphur Titanium Tungsten Vanadium	AI B C Cr Co Cu Mn Mo Ni Nb P Si S Ti W	ЮРУХКДГМНБПС ТВФ

Alloying elements comprise aluminium, boron, vanadium, tungsfen, cobalt, molybdenum, nickel, titanium, chromium, etc. Silicon and manganese are considered alloying elements, only if they are contained in the steel in amounts higher than usually.

4. Chemical Composition and Hardness of Market Steels

Steels are divided into three classes according to their use:

1. Structural steel is used mainly for the manufacture of machine parts and various structures.

2. Tool steel is used for the manufacture of tools.

3. Steels with specific physical and chemical properties are used for the manufacture of special machinery and machine parts.

Structural and tool steels are divided into carbon and alloy grades

according to their chemical composition.

Tables 17 to 36 present chemical composition, hardness numbers of various grades of steels (manufactured according to Soviet Standards and, partly, according to ministerial specifications), as well as critical temperatures of steels on heating.

Effect of Alloying Elements on Properties of Steels

Table 16 continued

Element	Tendency to overheating	Harden- ability	Annealing, normalising, and hardening temperatures	Hardness and strength	Ductility	High-tempera- ture strength
Molybdenum	Influences slightly	Increases	Increases	Increases	Increases when added in amounts	Increases
Nickel	Influences	Increases	Lowers	Increases	up to 0.6% Increases slightly	Influences slightly
Niobium Silicon	Influences	 Increases	Increases Increases	Lowers Increases	Increases Lowers	Increases
Titanium	slightly Lowers		Increases consider-	Increases slightly	Increases slightly	sugnery Influences slightly
Tungsten	Lowers	Increases	ably Increases	Increases	Increases slightly	Increases consider-
Vanadium	Lowers	l	Increases	Increases	when accelulation amounts up to 1% Increases	Influ- ences slightly

Note. Boron content in steel averages 0.002-0.005%.

 ${\it Table~17}$ Chemical Composition of Plain Carbon Steel

	Chemical composition, %							
Grade	Carbon	Manganese	Silicon					
	Open-heart	h steel						
МСт.0 МСт.1КП МСт.2КП МСт.3КП МСт.3 МСт.4КП МСт.4 МСт.5 МСт.6 МСт.7	○ 0.23 0.06-0.12 0.09-0.15 0.14-0.22 0.14-0.27 0.18-0.27 0.18-0.27 0.28-0.37 0.38-0.49 0.50-0.62	$\begin{array}{c} \\ \leqslant 0.05 \\ \leqslant 0.07 \\ \leqslant 0.07 \\ 0.12 \cdot 0.30 \\ \leqslant 0.07 \\ 0.12 \cdot 0.35 \\ 0.15 \cdot 0.35 \\ 0.15 \cdot 0.35 \\ 0.15 \cdot 0.35 \\ 0.15 \cdot 0.35 \\ \end{array}$	0.25-0.50 0.25-0.50 0.30-0.60 0.40-0.70 0.40-0.70 0.50-0.80 0.50-0.80					

Notes. 1. Phosphorus content should not exceed 0.07% for steel MCr.0, and 0.045% for other grades; corresponding amounts of sulphur are 0.06% and 0.055% respectively.

2. Table 32 lists chemical composition of open-hearth steels.

Table 18

Chemical Composition and Hardness Numbers of Free-cutting Steels

0.1		Chemical composition, %							
Grade	Carbon	Silicon	Manganese	Sulphur	Phosphorus	hardness number			
A12 A20	0.08-0.16 0.15-0.25	0.15-0.35 0.15-0.35	0.6-0.9 0.6-0.9	0.08-0.20 0.08-0.15	0.08-0.15 Below 0.06	167-217 167-217			
A30	0.25-0.35	0.15-0.35	0.7-1.0	0.08-0.15		174-223			
Α40Γ	0.35-0.45	0.15-0.35	1.2-1.55	0.18-0.3	Below 0.05	179-229			

Table 19
Chemical Composition and Mechanical Properties of Quality Carbon Steel

			Steel	,		
	Chem	ical composit	ion, %	Me	chanica	l properties
Grade	Carbon	Silicon	Manganese	Tensile strength, kg/mm²	Elonga- tion, %	Impression diameter, mm, minimum Hardness number, HB, maximum
			Group I			

05КП 08КП*	€ 0.06 0.05-0.11	€0.03 €0.03	€ 0.40 0.25-0.50	30	$\frac{-}{35}$	- 5.2	— 131
08 10КП* 10КП	0.05-0.12 0.07-0.14 0.07-0.14 0.12-0.19	0.17-0.37 0.07 0.17-0.37 ≤0.07	0.35-0.65 0.25-0.50 0.35-0.65 0.25-0.50	33 32 34 36	33 \ 31 \ 29 \	5.1	137
15КП* 15* 20КП*	0.12-0.19 0.12-0.19 0.17-0.24	0.07 0.17-0.37 ≤0.07	0.35-0.65 0.25-0.50	38 39	27 } 27 } 27 }	5.0	143
20* 25*	0.17-0.24 0.22-0.30	0.17-0.37 0.17-0.37	0.35-0.65 0.50-0.80	42 46	25 } 23	4.8	156 170
30* 35* 40	0.27-0.35 0.32-0.40 0.37-0.45	0.17-0.37 0.17-0.37 0.17-0.37	0.50-0.80 0.50-0.80 0.50-0.80	50 54 58	21 20 19	4.5 4.4 4.4	179 187 187
45 50	0.42-0.50 0.47-0.55	0.17-0.37 0.17-0.37	0.50-0.80 0.50-0.80	61 64	16 14	$\frac{4.3}{4.2}$	197 207
55 60	0.52-0.60 0.57-0.65 0.62-0.70	0.17-0.37 0.17-0.37 0.17-0.37	0.50-0.80 0.50-0.80 0.50-0.80	66 69 71	13 12 10	4.1 4.0 4.0	217 229 229
65 70 75	0.67-0.75 0.72-0.80	0.17-0.37	0.50-0.80 0.50-0.80	73 110	9 7	4.0 3.9	229 241
80 85	0.77-0.85 0.82-0.90	0.17-0.37 0.17-0.37	0.50-0.80 0.50-0.80	110 11 5	6 6	3.9 3.8	241 255
	Į.	l			l i	ļ	i

...Group II.

$\begin{array}{c ccccccccccccccccccccccccccccccccccc$	6 24 0 22 5 20	4.7 4.3 4.2 4.4 4.3	197 207 187
---	----------------------	---------------------------------	-------------------

	Chemi	ical compositi	on, %	Me	chanica	l propert	ies
Grade	Carbon	Silicon	Manganese	Tensile strength, kg/mm²	Elonga- tion, %	Impression diameter, mm, minimum	Hardness number, H _B ,
40Γ 45Γ 50Γ 60Γ 65Γ 70Γ	0.37-0.45 0.42-0.50 0.48-0.56 0.57-0.65 0.62-0.70 0.67-0.75	0.17-0.37 0.17-0.37 0.17-0.37 0.17-0.37 0.17-0.37	0.70-1.00 0.70-1.00 0.70-1.00 0.70-1.00 0.90-1.20 0.90-1.20	60 63 66 71 75 80	17 15 13 11 9	4.2 4.1 4.1 4.0 4.0 4.0	207 217 217 219 229 229 229

Notes. 1. Sulphur and phosphorus content for all grades of steel averages <0.04%, except for grades $05 \, \mathrm{KH}$, 08 and 10 where phosphorus content amounts to <0.035%. 2. Hardness numbers of steel grades marked with an asterisk refer to non-annealed steel, the balance representing hardness numbers of annealed steel.

Table 20

Chemical Composition and Hardness Numbers for Plough Share Rolled Steel

	Chen	nical compositi	on, %	Brinell hardness number as supplied				
Grade	Carbon Manganese		Other elements	Minimum impression diameter, mm	Maximum hardness number			
Л53	0.47-0.57	0.5-0.8	Silicon 0.15-0.4 Sulphur below 0.05	3.8	255			

Table 21
Chemical Composition and Hardness Numbers of Alloy Structural Steel

		Chemical o	composition.	%	Brinell hardness number after an- nealing
Grade	Carbon	Manganese	Chromium	Other elements	Minimum impression diameter, mm Maximum hard- ness number
35Γ2 40Γ2 45Γ2 50Γ2 15X 15XP 20X 30XPA 35XPA 35XPA 35XYA 40XP 45XI 50X 15XΓ 18XΓ 18XΓ 18XΓ 18XΓ 18XΓ 20XΓ 20XΓ 20XΓ 20XΓ 40XC 27CΓ	0.31-0.39 0.36-0.44 0.41-0.49 0.46-0.55 0.12-0.18 0.17-0.23 0.25-0.33 0.27-0.33 0.31-0.39 0.33-0.40 0.34-0.42 0.36-0.44 0.41-0.49 0.41-0.49 0.41-0.20 0.15-0.21 0.17-0.23 0.15-0.25 0.18-0.24 0.24-0.32 0.37-0.45 0.37-0.45 0.37-0.45 0.37-0.45 0.37-0.45 0.37-0.45 0.37-0.45 0.37-0.45 0.29-0.37 0.34-0.42 0.37-0.45	1.4-1.8 1.4-1.8 1.4-1.8 1.4-1.8 0.4-0.7 0.5-0.8 0.5-0.8 0.5-0.8 0.5-0.8 0.5-0.8 0.5-0.8 0.5-0.8 0.5-0.8 0.5-0.8 1.1-1.4 0.9-1.2 0.8-1.1 0.9-1.2 0.7-1.1 1.6-1.9 0.3-0.6 0.3-0.6 1.1-1.4			4.2 207 4.1 217 4.0 229 4.0 229 4.5 179 4.4 187 4.5 179 4.4 187 4.1 217 4.0 229 4.1 217 4.0 229 4.1 217 4.0 229 4.1 217 4.0 229 4.1 217 4.0 229 4.1 217 4.0 229 4.1 217 4.1 87 4.3 197 4.0 229 4.0 229 3.9 241 3.8 255 3.8 255 3.8 255 4.1 217

		Chemical co	omposition,	%	Brin hardn numl after neali	ess oer an-
Grade	Carbon	Manganese	Chromium	Other elements	Minimum Impression diameter, mm	Maximum hard- ness number
35CF 36F2C 15XM 20XM 30XM 35XM 35X2MA 35X2MA 35X2MA 35XBA 15XФ 20XФ 40XФ 15HM 20HM 40HM 20XH 40XH 45XH 50XH 12XH2 12XH3A 12X2H4A 20X2H4A 30XH3A 37XH3A 37XH3A 34XH1M 34XH3M	0.31-0.39 0.32-0.40 0.11-0.18 0.15-0.25 0.26-0.34 0.32-0.40 0.35-0.42 0.12-0.18 0.17-0.23 0.37-0.44 0.1-0.18 0.17-0.25 0.37-0.45 0.17-0.25 0.41-0.49 0.46-0.54 0.09-0.16 0.09-0.16 0.09-0.16 0.09-0.16 0.17-0.24 0.16-0.22 0.27-0.34 0.33-0.4 0.3-0.4	1.1-1.4 0.4-0.7 0.4-0.7 0.4-0.7 0.4-0.7 0.4-0.7 0.25-0.5 0.4-0.7 0.5-0.8 0.4-0.7 0.5-0.8 0.4-0.7 0.5-0.8 0.3-0.6 0.3-0.6 0.3-0.6 0.3-0.6 0.3-0.6 0.3-0.6 0.5-0.8	© 0.25 © 0.25 0.8-1.1 © 0.3 0.8-1.1 1.6-1.9 0.9-1.3 0.8-1.1 0.8-1.1 0.8-1.1 0.8-1.1 0.8-5.0.75 0.45-0.75 0.45-0.75 0.45-0.75 0.45-0.75 0.45-0.75 0.45-0.75 0.45-0.75 0.45-0.75 0.6-0.9 1.25-1.65 0.6-0.9 1.25-1.65 0.6-0.9 1.25-1.65 0.6-0.9 1.25-1.65 0.6-0.9 1.25-1.65 0.6-0.9 1.25-1.65 0.6-0.9 1.25-1.65	1.1-1.4 Si 0.4-0.7 Si 0.4-0.55 Mo 0.4-0.55 Mo 0.4-0.55 Mo 0.15-0.25 Mo 0.15-0.25 Mo 0.15-0.25 Mo 0.1-0.2 V 0.1-0.2 V 0.1-0.2 V 1.5-2.0 Ni 0.2-0.3 Mo 1.0-1.4 Ni 1.0-1.5-2 Ni 2.8-3.2 Ni 3.3-3.7 Ni 2.8-3.2 Ni 3.0-3.5 Ni 0.25-0.4 Mo 2.75-3.25 Ni 0.25-0.4 Mo	4.0 4.0 4.5 4.3 4.0 3.9 4.4 4.3 4.3 4.2 4.2 4.2 4.1 3.7 3.9 3.9 3.7	229 179 179 129 241 241 229 187 241 197 241 197 207 207 207 207 207 2217 269 241 269

		Chemical c	omposition,	%	Brine hardn numb after neali	ess er an-
Grade	Carbon	Manganese	Chromium	Other elements	Minimum impression diameter, mm	Maximum hard- ness number
36XН1МФ А	0.32-0.4	0.3-0.5	1.3-1.7	1.3-1.7 Ni 0.3-0.4 Mo		
45ХНМФА	0.42-0.5	0.5-0.8	0.8-1.1	0.08-0.15 V 1.3-1.8 Ni 0.2-0.3 Mo		
30XHBA	0.27-0.34	0.3-0.6	0.6-0.8	0.1-0.2 V 1.27-1.65 Ni	3.9	241
38XHBA	0.34-0.42	0.3-0.6	1.3-1.7	0.5-0.8 W 1.25-1.65 Ni 0.5-0.8 W	3.7	269
40XHBA	0.37-0.44	0.5-0.8	0.6-0.9	1.25-1.65 Ni 0.8-1.2 W	3.7	269
40XHMA	0.37-0.44	0.5-0.8	0.6-0.9	1.25-1.65 Ni 0.15-0.25 W	3.7	269
30X2HBA	0.27-0.34	0.3-0.6	1.6-2.0	1.4-1.8 Ni 1.2-1.5 W	3.7	269
38XH3BA	0.34-0.42	0.25-0.55	0.8-1.2	2.75-3.25 Ni 0.5-0.8 W	3.7	269
18X2H4BA	0.14-0.2	0.25-0.55	1.35-1.65	4.0-4.5 Ni 0.8-1.2 W	3.7	269
25 X 2H 4B A	0.21-0.28	0.25-0.55	1.35-1.65	4.0-4.5 Ni 0.8-1.2 W	3.7	269
30ХНВФА	0.27-0.34	0.3-0.6	0.6-0.9	2.0-2.5 Ni 0.5-0.8 W 0.15-0.3 V		
30Х2НВФА	0.27-0.34	0.3-0.6	1.6-2.0	1.4-1.8 Ni 1.2-1.5 W	3.7	269
38ХН3ВФА	0.34-0.42	0.25-0.55	1.0-1.4	0.18-0.28 V 3.0-3.5 Ni 0.5-0.8 W 0.1-0.2 V		
	1	1,	ľ			

		Chemical c	om position,	%	Brine hardn numl after neali	ess er an-
Grade	Carbon	Мапganese	Chromium	Other elements	Minimum impression diameter, mm	Maximum hard- ness number
20ХН4ФА	0.17-0.2	0.25-0.55	0.7-1.1	3.75-4.25 Ni	3.7	269
38ХНЗМФА	0.34-0.42	0. 25 -0.55	1.2-1.5	0.15-0.3 V 3.0-3.5 Ni 0.35-0.4 M o	3,7	269
15X ГНТ 15X 2 ГН2 Т 15X 2 ГН2 ТРА	0.12-0.18 0.12-0.18 0.12-0.18	0.7-1.0 0.7-1.0 0.7-1.0	0.7-1.0 1.4-1.8 1.4-1.8	0.1-0.2 V 0.06-0.12 Ti 0.06-0.12 Ti 0.002-0.005B	4.0 3.7 3.7	229 269 269
18ΧΓΗ 25Χ2ΓΗΤΑ	0.16-0.22 0.22-0.29	0.8-1.1 0.8-1.1	0.4-0.7 1.3-1.7	0.06-0.12 Ti 0.4-0.7 Ni 0.9-1.3 Ni	3.8 3.7	255 2 6 9
30XFHA 38XFH 30X2FH2 16XCH	0.28-0.35 0.35-0.43 0.26-0.34 0.13-0.20	0.6-0.9 0.8-1.1 0.8-1.1 0.3-0.6	0.9-1.2 0.5-0.8 1.4-1.7 0.8-1.1	0.06-0.12 Ti 0.3-0.6 Ni 0.7-1.1 Ni 1.4-1.8 Ni 0.6-0.9 Ni	4.0 4.0 3.8	229 229 255
18XCHPA	0.16-0.21	0.6-0.9	0.8-1.1	0.6-0.9 Si 0.8-1.1 Ni 0.6-0.9 Si	4.3	197
20XFCA 25XFCA 30XFCA 30XFCHA	0.17-0.23 0.22-0.28 0.28-0.34 0.27-0.34	0.8-1.1 0.8-1.1 0.8-1.1 1.0-1.3	0.8-1.1 0.8-1.1 0.8-1.1 0.9-1.2	0.002-0.005B 0.9-1.2 Si 0.9-1.2 Si 0.9-1.2 Si 0.9-1.2 Si	4.2 4.1 4.0 3.8	207 217 229 255
35ХГСА 38ХЮ 38ХМЮА 38ХВФЮ	0.32-0.39 0.35-0.43 0.35-0.42 0.35-0.43	0.8-1.1 0.2-0.5 0.3-0.6 0.2-0.4	1.1-1.4 1.5-1.8 1.35-1.65 1.5-1.8	1.4-1.8 Ni 1.1-1.4 Si 0.5-0.8 Al 0.15-0.25 Mo 0.2-0.4 W 0.4-0.7 Al	3.6 4.0 4.0 4.0	241 229 229 229 229
13H2XA	0.09-0.16	0.3-0.6	0.2-0.5	0.1-0.2 V 1.7-2.1 Ni	4.2	20 7

Table 22
Chemical Composition and Hardness Numbers of Non-annealed Spring Steels

		Chemical c	om positio	n, %		ihard- umber
Grade	Carbon	Silicon	Manga- nese	Other elements	Minimum impression diameter, mm	Maximum hardness num ber, HB
65 70 75 85 55ΓC 65Γ 50C2 55C2 60C2A 63C2A 70C3A 70C2X A (ЭИ142) 50 XΓ 50 XΓA 50 X Φ A 60C2X Φ A 60C2X Φ A 60C2X Φ A 60C2X Φ A 60C2H2A 55CΓ 60CΓ 60CΓ 60CΓ	0.6-0.7 0.65-0.75 0.72-0.80 0.82-0.90 0.52-0.60 0.6-0.7 0.47-0.55 0.52-0.60 0.57-0.65 0.56-0.64 0.60-0.65 0.46-0.54 0.46-0.54 0.46-0.54 0.46-0.54 0.56-0.64 0.56-0.64 0.56-0.64 0.56-0.64 0.56-0.64 0.56-0.64 0.55-0.60 0.55-0.65 0.56-0.64		0.5-0.8 0.5-0.8 0.5-0.8 0.6-0.9 0.8-1.0 0.8-1.0 0.8-1.0 0.8-1.0 0.8-1.0 0.8-1.0 0.8-1.0 0.8-1.0 0.8-1.0		3.87 3.66 3.67 3.66 3.55 3.55 3.55 3.55 3.55 3.55 3.55	255 269 285 302 285 269 285 285 302 302 302 302 302 302 302 302 302 302

Notes. 1. The 70C2XA (9H142) grade steel is marketed in the form of strips. 2. In addition, high-quality steel differs from quality steel as to lower sulphur and phosphorus content.

Chemical Composition and Hardness Numbers of Ball-bearing Steels

	С	hemical con	mposition,	%	Brinell hardness number			
Grade	Carbon	Manganese	Chromium	Sílícon	Minimum impres- sion di- ameter,	Maximum mum hardness number, H		
ШХ6 ШХ9 ШХ15 ШХ15СГ ШХ10	1.05-1.15 1.0-1.1 0.95-1.1 0.95-1.1 0.32-0.42	0.2-0.4 0.2-0.4 0.9-1.2	0.9-1.2 1.3-1.65 1.3-1.65	0.15-0.35 0.15-0.35 0.15-0.35 0.4-0.65 0.17-0.37		170-207 170-207 170-207 170-207		

Notes. 1. The grade IIIX10 steel contains maximum 0.03% both of sulphur and phosphorus. The remaining grades of steel contain less than 0.027% of sulphur and less than 0.02% of phosphorus.
2. Nickel content may rise up to 0.3%, that of copper up to 0.25%, while their sum should not exceed 0.5%.

Low-coercivity magnetic steels and alloys are used for the manufacture of parts of electrical machines. These steels comprise electric sheet steel, low-carbon electric steel, high permeability alloys and alloys with specific magnetic properties.

Table 24 lists data on electric sheet steels, while Table 25 presents

data on low-carbon electric steels.

High permeability alloys, i. e., permalloys, include iron-nickel alloys of grades 45H, 50H, 50HH, 05HH, 38HC, 50HXC, 79HM, 80H XC, 79HMA.

Table 26 presents data on alloys with specific magnetic properties.

Table 24 Electric Sheet Steel

Grade	Silicon content, %	Degree of alloying
911, 912, 913, 91100, 91200, 91300 921, 922 931, 932, 93100, 93200, 9310, 9320,	0.8-1.8 1.8-2.8	Low Medium

Grade	Silicon content, %	Degree of alloying
9330, 9330A	2.8-3.8	Fairly high
941, 942, 943, 943A, 944, 943, 946, 947, 948	3.8-4.8	High

Note. The steel grade designation is read as follows: Θ - electric; the first figure designates the degree of alloying with silicon; the second figure - guaranteed electric and magnetic properties; the letter "A" after the last figure denotes extra low specific losses; the third figure "0" means that the steel has been cold-rolled and possesses adequate texture; the paired figures "00" denote that the steel is cold-rolled and has no definite texture.

Table 25

Low-carbon Electric Steel

		Chemica	compo	sition,	%. max		Former	na ma	
Grade	Carbon	Manga- nese	Silicon	Phos- phorus	Sul- phur	Copper	0 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6		
Э, ЭА, ЭАА	0.04	0.2	0.2	0.025	0.03	0.15	Armco	iron	

Note. The difference between the grades is set by checking their magnetic properties after annealing. Coercivity of steels of grades 9, 9A and 9AA should not exceed 1.2, 1.0 and 0.8 oersteds, respectively.

Table 26

Alloys with Special Magnetic Properties

				Ja.	me							Avera main	age cont elemen	ent of ts, %
			 _				_			_		Nickel	Cobalt	Iron
Permendur Ditto . Perminvar Ditto . Ditto . Isoperm . Ditto .	•		 			 		 	 	 	 	45 70 45 50 36	50 49 25 7 25 —	50 49 29.4 22.4 21.9 50 55

Table 27

Chemical Composition and Hardness Number of Magnetic Steel as Supplied

			Chemical o	Chemical composition, %			Brinell hard.
Grade	Carbon	Chromium	Tungsten	Cobalt	Molyb-	Other elements	ness number
EX	0.95-1.10	1.3-1.6		 	 	Manganese 0.2-0.4	217-241
EX3	0.9-1.1	2.8-3.6	1	1	1	Silicon 0.17-0.4	229-285
E7B6	0.68-0.78	0 3-0.5	5.2-6.2	1	1	Sulphur 0.02 max	255-321
EX5K5	0.90-1.05	5.5-6.5	1	5.5-6.5	1	Phosphorus 0.03 max	269-341
EX9K15M	0.90-1.05	8.0-10.0	1	13.5-16.5	1.2-1.7		285-341

Note. Steel which is off standard in chemical composition may be accepted provided all its properties are up to specifications.

Table 28

Chemical Composition of Cast Permanent Magnets

		Iron		The balance
		Carbon Manganese	below	0.35
		Carbon	p (0.03
•	Chemical composition, %	Cilicon	3111011	0.15
	са] сошр	Conner) addoo	1 4 8 1 6 4 8
	Chemic	Cobalt	Cobail	74 25 27 27 27 28 28 28 28 28 28 28 28 28 28 28 28 28
		Nickel Aluminium Coholt Conner		12.00 20.00 20.00 20.00 20.00
		Nickel	MICHEL	22 24.5 23.5 19 13.5
	Grade of alloy	PIO PIO	Old	АЛНИ! АЛНИЗ АЛНИЗ АЛНИКО! АЛНИКО!5 АЛНИКО!5 АЛНИКО!6 АЛНИКО!8 АЛНИКО!8
	Grade	mo N	Man	AH1 AH2 AH3 AHK AHK01 AHK02 AHK03

Chemical Composition of Γ13 Steel

			Chemica	Chemical composition, %	%	
Grade	Carbon	Manganese	-	Silicon, max	Sulphur, max	ax Phosphorus, max
F13	1.0-1.3	10-14		0.5	0.03	0.03
-	_		-		-	Table 30
	Cher Non-so	Chemical Composition of Corrosion-resisting, Non-scaling and Heat-resisting Steels and Alloys	tion of Co at-resistin	rrosion-resist g Steels and	ting, Alloys	
			Cher	Chemical composition, %	ilon, %	
Grade of steel or alloy	oy Carbon	Silicon	Manganese	Manganese Chromium	Nickel	Other elements
		Mar	Martensitic Steels	sels		
Ϋ́	1 < 0.15	_	₹0.2	4.5-6.0	-	1
W X	V 0.15	0.5	₹0.5	4.5-6.0	1	Mo 0.45-0.6
XFR	₹ 0.15		₹0.5	4.5-6.0	1	W 0.4-0.7
XECM	V 0.15		<0.7	2.0-6.0	1	Mo 0.45-0.6
1Х8ВФ	0.08-0.15		₹ 0.5	7.0-8.5	1	W 0.6-1.0 V 0.3-0.5
(0) VOC 40)	0 98 0 45		V 0 7	8.0-10.0	!	: 1
4X3C2 (3CA 18)	0.35-0.45	1.9-2.6	V 0.7	9.0-10.5	1	Mo 0.7-0.9
1X12H2BMФ (ЭИ961)			9.0.₹	10.5-12.0	1.5-1.8	W 1.6-2.0 Mo 0.35-0.5
						V 0.18-0.3
2X 13 3X 13	0.16-0.24	24 \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \	№ 0.6	12.0-14.0 12.0-14.0	11	1.1
	_	_		_	_	

Table 30 continued

			i			
or the second second			Che	Chemical composition, %	tion, %	
Orage of steel of alloy	Carbon	Silicon	Manganese Chromtum	Chromtum	Nickel	Other elements
AV 13	0.35.0.44		7 0.6	19.0-14.0	1	. 1
47.13 37.13H7C9 (3M79)	0.55-0.34		× 0.7	12.0-14.0		į
1X13H3	0.08-0.15	€0.6	9.0	12.5-14.5	2.2-3.0	1
1X17H2	0.11-0.17		8.0√	16.0-18.0		{
9X 18 (ЭИ229)	0.1-6.0		<0.7	17.0-19.0	1	1
		Marten	Martensito-ferritic Steels	: Steels		
X6CIO (3M428)	≤ 0.15		< 0.5	5.5-7.0	-	Al 0.7-1.1
1X11M¢	0.12-0.19	≤0.5	€0.7	10.0-11.5		Mo 0.6-0.8
WHASIX!	0 19-0 18	40	0.5-0.9	0.5-0.9 11.0-13.0	0 4-0 8	V 0.25-0.4 W 0.7-1.1
(2007) ±1.11071V1)			Mo 0.5-0.7
			,	($V_{0.15-0.3}$
2Х12ВМБФР (ЭИ993)	0.15 - 0.22	₹0.5	11.0-13.0	11.0-13.0 11.0-13.0	Ī	W 0.4-0.7
						M6 0.4-0.6 N5 0.2-0.4
		_	_		_	V 0.15-0.3
(952MG) WMGBGIAI	0 1-0 17	\ \	α 	0 5 0 9 11 0 13 0		B 0.003
(October (October)		 //	0.0.0.0	0.01		Mo 0.6-0.9
					-	V 0.15-0.3
1X13	0.09-0.15	9.0≽	≤0.6 ≤0.6 12.0-14.0	12.0-14.0		1
		F	Ferritic Steels	Š		
1X 12CIO (3M404)	[0.07-0.12]	_		<0.7 12.0-14.0	1	Ai 1.0-1.8
0X13 (3M496)	80.0≽	9.0≯	9.0≽	11.0-13.0	1	1

						Table 30 continued
Grade of steel or allow			Che	Chemical composition,	tion, %	
	Carbon	Silicon	Manganese Chromium	Chromium	Nickel	Other elements
X14 (3M241)	<0.15	€0.7	≥0.7	13.0-15.0	!	
X17	≤0.12	8.0∜	€0.7	16.0-18.0	-	1
0X17T (ЭИ645)	80.0 8	₩.	€0.7	16.0-18.0	1	Ti 5.C-0.80
X 18CFO (3M484)	\ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \	1.0-1.5	V 0.5	17.0-20.0	j	AI 0.7-1.2
X28 (ЭИ349) X28 (ЭИ349)		0.0 V/ V/	% \	$\begin{vmatrix} 24.0 - 27.0 \\ 27.0 - 30.0 \end{vmatrix}$	1!	Ti 5.C-0.80 —
		Austen	Austeno-martensitic Steels	c Steels		
2:X13H4F9 (ЭИ100)	10.15-0.3	8.0.≽	1 8.0-10.0	8.0-10.0112.0-14.01	3.7-4.7	I
X15H9IO (ЭИ904)	0.03	8.0.8	₹0.8	14.0-16.0		Al 0.7-1.3
X17H7IO (ЭИ973) 2X17H2	≤ 0.09 0.22-0.28	8.0 V V 8.0 8.0 8.0	8 8 V V	16.0-18.0 16.0-18.0	6.5-7.5	Al 0.8-1.3
	-	Auste	Austeno-ferritic	Steels	•	
0X20H14C2 (3M732)	≥0.08	2.0-3.0		119.0-22.0	12.0-15.0	l
X20H14C2 (ЭИ211)	V 0.2	2.0-3.0		19.0-22.0	12.0-15.0	; ; i
0X21H51 (ЭИ53) 0X21H6M2T (ЭИ54)	¥ V 0.088	× 8. V V	8.0 √√ √√ 8.0 8.0 8.0 8.0 8.0 1.0 8.0 1.0 1.0 1.0 1.0 1.0 1.0 1.0 1.0 1.0 1	20.0-22.0	4.8-5.8 5.5-6.5	Ti 0.3-0.6 Ti 0.2-0.4
Wolley Table	000	ه د ا	0		, n	T: /C 0 001.8
X23H13 (9H811)	0.03-0.14 ≪ 0.2	0.0 	× 2.0 × 2.0	22.0-25.0	12.0-15.0	11 (C-0.0Z)3-0.80 —
. Х28АН (ЭИ657)	≤ 0.15	V ≥ 1.0	V 1.5	25.0-28.0	1.0-1.7	ļ
		Aı	Austenitic Steels	sels		
4Х12Н8Г8МФБ (ЭИ481)	0.34-0.4	0.3-0.8	7.5-9.5	0.3-0.8 7.5-9.5 11.5-13.5	7.0-9.0	Mo 1.1-1.4
						V 1.25-1.55

Table 30 continued

Grade of steel or alloy 0X10H20T2 X12H20T3P (ЭИ696) X12H22T3MP (ЭП33)	Carbon Carbon \$< 0.08 \$< 0.1 \$< 0.1 \$< 0.1	1 1 1	Chemical composition Chemical composition Chromium ≤ 2.0 10.0-12.5 ≤ 1.0 10.0-12.5 ≤ 0.6 10.0-12.5 1.0-2.0 13.0-15.0	Chemical composition. % See Chromitam Nickel O 10.0-12.0 18.0-20.0 O 10.0-12.5 18.0-21.0 O 10.0-12.5 21.0-25.0 O 13.0-15.0 14.0-17.0	Chromium Nickel 10.0-12.0 18.0-20.0 10.0-12.5 18.0-21.0 10.0-12.5 21.0-25.0 13.0-15.0 14.0-17.0	Other elements Ti 1.5-2.5 Al < 1.0 Ti 2.6-3.2 Al < 0.8 B 0.003-0.02 Ti 2.6-3.2 Al < 0.8 Al < 0.8 No 1.0-1.6 B 0.02
4P) 95) (695P)	0.07-0.12	\$0.0 \$0.0 \$0.6	1.0-2.0	13.0-15.0 14.0-17.0 13.0-15.0 18.0-20.0 13.0-15.0 18.0-20.0	14.0-17.0 18.0-20.0 18.0-20.0	Nb 0.9-1.3 B 0.005 W 2.0-2.75 Nb 0.9-1.3 W 2.0-2.75 Nb 0.9-1.3
1X14H18B25P1 (ЭИ726) 0.07-0.12	0.07-0.12	9.0≽	1.0-2.0	13.0-15.0 18.0-20.0	18.0-20.0	Ce ≤ 0.02 W 2.0-2.75 Nb 0.9-1.3 B ≤ 0.025 Ce ≤ 0.02
X14F14H X14F14H3T (ЭИ711) 4X14H14B2M (ЭИ69)	≤ 0.12 ≤ 0.1 0.4.0.5	\ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \	13.0-15.0 13.0-15.0 0.7	13.0-15.0 13.0-15.0 13.0-15.0 13.0-15.0 13.0-15.0 0.7 13.0-15.0	1.0-1.5 2.5-3.5 13.0-15.0	Ti (C-0.02).5-0.60 W 2.0-2.75 Mo 0.25-0.4

Table 30 continued

		-	Cher	Chemical composition, %	ition, %	
Grade of steel of alloy	Сагроп	Silicon	Silicon Manganese Chromium	Chromium	Nickel	Other elements
4X15H7Γ7Φ2MC (ЭИ388)	0.38-0.47	0.9-1.4	6.0-8.0	14.0-16.0	6.0-8.0	Mo 0.65-0.95
0X14H28B3T3IOP (ЭИ786)	80.0≥	9.0≥	9.0≥	13.0-15.0	13.0-15.0 26.0-29.0	Ti 2.4-3.2
				-		Al 0.5-1.2 W 2.8-3.5 B < 0.09
1X16H13M2B (ЭИ680)	0.06-0.12	8.0≯	≤1.0	15.0-17.0	15.0-17.0 12.5-14.5	Mo 2.0-2.5
X16H15M35 (ЭИ847)	60.0	8.0.₹	9.0≥	15.0-17.0	15.0-17.0 14.0-16.0	Mo 2.5-3.0
X17F9AH4 (ЭИ878) X17H13M2T (ЭИ448)	≤0.12 ≤0.1	∧ № 0.8 № 0.8	8.0-10.5	8.0-10.5 16.0-18.0 1.0-2.0 16.0-18.0	16.0-18.0 3.5-4.5 16.0-18.0 12.0-14.0	Nitrogen 0.15-0.25
X 17H13M3T (ЭИ432)	≤0.1	8.0.8	1.0-2.0	16.0-18.0	16.0-18.0 12.0-14.0	Mo 1.8-2.5 Ti 0.3-0.6 Mo 2.6.4
0X17H16M3T (ЭИ580)	≥0.08	8.0≯	1.0-2.0	16.0-18.0 15.0-17.0	15.0-17.0	Mo 9.0-4.0
X17AF14 (ЭП213)	≤0.15	6 0.8	13.5- 15.5		0.6	Nitrogen 0.3-0.4
00X18H10 (ЭИ842) 0X18H10 (0X18H9)	V V V	8.0 V/V	1.0-2.0 $1.0-2.0$	17.0-19.0	9.0-11.0	1 1
X18H9 (1X18H9)	€0.12	8.0 √	1.0-2.0	17.0-19.0		1
2X 18H9 X 18H10F (ЭИ453)	0.13-0.21	8.0 V V	1.0-2.0	17.0-19.0 17.0-19.0	9.0-10.0	Se or Te 0.18-0.35
0X 18H10T (3M914)	8 0.0 8	8.0 √	1.0-2.0	17.0-19.0		Ti ≤ 0.6
X18H10T (1X18H9T) X18H9T (1X18H9T)	V 0.12 V 0.12	& 8. 0 0 V V	1.0-2.0	17.0-19.0 17.0-19.0	8.0-9.5	Ti (C-0.02)·5-0.7 Ti (C-0.02)·5-0.7
0X18H11 (3M684)	≥0.06	₹0.8	1.0-2.0	17.0-19.0		

Table 30 continued

Tadle 30 continued

Table 30 continued

	Other elements	Ti 1.8-2.3 Al 1.7-2.3 W 5.0-7.0 Mo 2.0-4.0 V 0.1-0.5 B € 0.02	Fe \$\left\{ \tilde{\cappa}\} \cdot 0.02 \\ \tilde{\cappa}\} \tilde{\cappa}\] 1.8-2.3 \\ \tilde{\cappa}\} All 0.5-1.0 \\ \tilde{\cappa}\} 1.0-1 5.	Fe 63.0 Ti 1.9-2.8 Al 1.0-1.7 W 9 0.3 5	Mo 4.0-6.0 Nb 0.5-1.3 B ≤ 0.01 Ce ≤ 0.02 Fe ≤ 5.0 Ti 2.2-2.8 W 4.0-5.0 Mo 4.0-5.0 B ≤ 0.01 Ce ≤ 0.01 Al 1.0-1.5 Fe ≤ 4.0
ition, %	Nickel	The balance	Ditto	Ditto	Ditto
Chemical composition, %	Chromium	13.0-16.0 The bal- ance	15.0-18.0	16.0-19.0	17.0-20.0
Cher	Мапдапезе	≪0.5	≥ 1.0	€0.5	€0.5
	Silicon	9.0≽	≥0.8	€0.6	≥0.6
	Carbon	≤0.12	≈0.08	≤0.12	№ 0.08
	Grade of steel or alloy	хнтовмтю (эмыт)	XH80TBЮ (ЭИ607)	хн70мвтюв (эи598)	XH67BMTiO (ЭИ445P)

Table 30 continued

			Cher	Chemical composition. %	tion, %	
Grade of steel or alloy	Carbon	Silicon	Manganese	Chromium	Nickel	Other elements
хн75мБТЮ (ЭИ602)	<0.1	€0.8	≥ 0.4	19.0-22.0	The baf- ance	Ti 0.35-0.75 Al 0.35-0.75 Mo 1.8-2.3 Nb 0.9-1.3
XH78T (ЭИ435)	≤0.12	₹0.8	<0.7	19.0-22.0	Difto	Fe ≤ 8.0 Ti 0.15-0.35 Al ≤ 0.15
ХН77ТЮ (ЭИ437А)	≥0.06	€0.6	≤0.4	19.0-22.0	Ditto	Fe = 0.0 Ti 2.3-2.7 Al 0.55-0.95
ХН77ТЮР (ЭИІЗ7Б)	90.0≶	≥0.6	<0.4	19.0-22.0	Ditto	Fe $\leqslant 4.0$ Ce $\leqslant 0.01$ Ti $2.3.2.7$ Al $0.55-0.95$ Fe $\leqslant 4.0$
хн6ов (ЭИ868)	V 0.1	8.0≥	<0.5	23.5-26.5	Ditto	D ≤ 0.01 Cc ≤ 0.01 Ti 0.3-0.7 Al ≤ 0.5 W 13.0-16.0
ХН70Ю (ЭИ652)	€0.1	6.0 ≥	<0.3	26.0-29.0	Difto	Fe < 4.0 Al 2.6-3.5 Fe < 1.0 Ba < 0.1
XH70 (ЭИ442)	<0.07	₹0.8	≥0.5	28.0-31.0	Ditto	Ce ≤ 0.03 A1 ≤ 0.15 Fe ≤ 5.0

Table 31

Hardness Numbers of Stainless Steels Supplied in Annealed State

0X13 1X13 2X13 3X13 4X13 417	Brinell hard Impression diameter, mm 4.5-5.5 4 4.5-5.3 4.2-5.2 4.2-5.0 4.3-5.0	In Hardness number Hardness number 116-179 121-187 126-197 131-207 143-229 143-229
12	3.6 min	225 max 286 max

Table 32

Chemical Composition of Alloys Used for the Manufacture of Electric Heating Resistors

				Ö	Chemical composition, %	ition, %		
Grade	Carbon	arbon Manga-	Silicon	Chromium	Nickel	Aluminium	Titanlum	Iron
		тахітип	tt ED					
X 13 104	0.15	0.4	1.0	16.0-19.0	9.0≫		ı	The balance
0X 23IO5	90.0	0.5	0.7	21.5-24.5	Ditto		1	Ditto
0X23lO5A	0.02	0.3	9.0	21.5-23.5	Ditto		1	Ditto
0X27FO5A	0.05	0.3	9.0	26.0-28.0	Ditto		1	Ditto
X 15H60	0.15	0.7	0.4-1.3	15.0-18.0	55.0-61.0	<0.2		Ditto
X 20 H80	0.15	0.7	0.4-1.3	20.0-23.0	75.0-78.0		ı	0:1
X 20H80T3	0.08	0.5	1.0 тах	19.0-23.0	The balance		2.0-2.9	<2.5

Chemical Composition and Hardness Numbers of Carbon Tool Steels

	Chem	ical composi	tion, %	Brinell hard	ness number
Grade	Carbon	Manganese	Other elements	Impression diameter, mm, minimum	Hardness number, maximum
У7 У8 У8Г У9 У10 У11 У12 У13	0.65-0.74 0.75-0.84 0.80-0.90 0.85-0.94 0.95-1.04 1.05-1.14 1.15-1.24 1.25-1.35	0.20-0.40 0.20-0.40 0.35-0.60 0.15-0.35 0.15-0.35 0.15-0.35 0.15-0.35	0.15-0.35 Sulphur 0.03 max Phosphorus	4.4 4.4 4.35 4.3 4.2 4.2	187 187 187 192 197 207 207 217

Note. High-quality steel differs from quality steel, the amount of carbon being equal for both, by lower sulphur and phosphorus contents, 0.02% and 0.03 as a maximum, respectively, and by a slightly different content of manganese and silicon. High-quality steel is designated V7A, V8A, VBFA, etc.

Table 34

Chemical Composition and Hardness Numbers of Alloy Tool Steels

		Chemical c	omposition	%		hardness n ber
Grade	Carbon	Chromium	Tungsten	Other elements	Impres- sion dia, mm	Hardness number
7X3 8X3	0.60-0.75 0.76-0.85	3.2-3.8 3.2-3.8		<u>-</u>		229-187 255-207

		Chemical o	composition	1. %		hardness iber
Grade	Carbon	Chromium	Tungsten	Other elements	Impression dia,	Hardness number
65 X 9 X X X 09 X 05 X 12 Φ	0.80-0.95 0.95-1.10 0.95-1.10 1.25-1.40	1.3-1.6 0.75-1.05 0.4-0.6 11.5-13.0	_ _ _ _		4.1-4.5 4.0-4.4 4.0-4.5 3.9-4.3 3.7-4.1 4.1-4.5	229-187 229-179
χr	1.3-1.5	1.3-1.6	_	Manganese		
4XC	0.35-0.45	1.3-1.6	_	0.45-0.70 Silicon	3.9-4.3	241-197
6XC	0.6-0.7	0.95-1.25		1.2-1.6 Silicon	4.2-4.6	207-170
	1		_	0.6-1.0	4.0-4.4	229-187
XLC	0.95-1.1	1.4-1.8	_	Silicon 0.5-1.0	3.8-4.2	255-207
	!	,		Manganese 0.8-1.2		
9XC	0.85-0.95	0.95-1.25	-	Silicon		
8ХФ	0.75-0.85	0.5-0.8	_	1.2-1.6 Vanadium	3.9-4.3	241-197
85ХФ	0.80-0.90	0 45-0 70		0.15-0.30 Vanadium	4.2-4.6	207-170
		Í	_	0.15-0.30	4.2-4.6	207-170
4XB2C	0.35-0.44	1.0-1.3	2.0-2.5	Silicon 0.6-0.9	4.1-4.5	217-179
5XB2C	0.45-0.54	1.0-1.3	2.0-2.5	Silicon		
6XB2C	0.55-0.65	1.0-1.3	2.2-2.7	0.5-0.8 Silicon	3.8-4.2	
5ХВГ	0.55-0.70	0.5-0.8	0.5-0.8	0.5-0.8 Manganese	3.6-4.0	285-229
9ХВГ	0.85-0.95	0.5-0.8	0.5-0.8	0.9-1.2	4.1-4.5	217-179
]			Manganese 0,9-1.2	3.9-4.3	241-197
ХВГ	0.90-1.05	0.9-1.2	1.2-1.6	Manganese 0.8-1.1	3.8-4.2	255-207

Table 34 continued

Grade		Chemical c	omposition	, %		hardness ber
Grade	Carbon	Chromium	Tungsten	Other elements	Impres- sion dia, mm	Hardness number
3X2B8 4X8B2 XB5	0.30-0.40 0.35-0.45 1.25-1.50	7.0-9.0	7.5-9.0 2.0-3.0 4.5-5.5	— Vanadium	3.8-4.2	255-207 255-207
X12M	1.45-1.70	11.0-12.5	_	0.15-0.30 Molybdenum 0.4-0.6		285-229 255-207
5XHM	0.5-0.6	0.5-0.8	-	Vanadium 0.15-0.30 Molybdenum 0.15-0.30 Nickel 1.4-1.8	3.9-4.3	241-197
5ХГМ	0.5-0.6	0.6-0.9	_	Molybdenum 0.15-0.30 Manganese	3.9-4.3	241-197
5XHT	0.5-0.6	0.6-0.9	_	1.2-1.6 Nickel 1.4-1.8 Titanium	3.95-4.35	235-192
5XHC	0.5-0.6	1.3-1.6	_	0.1-0.2 Nickel 0.8-1.3 Silicon	3.8-4.2	255-207
5XHCB	0.5-0.6	1.3-1.6	0.4-0.6	0.6-0.9 Nickel 0.8-1.3 Silicon	3.8-4.2	255-207
5XHB	0.5-0.6	0.5-0.8	0.6-1.0	0.6-0.9 Nickel	2040	044 40=
45 X H T	0.43-0.5	1.2-1.4	_	1.4-1.8 Nickel		241-197
45ХНВ Х12Ф1	0.43-0.5 1.45-1.70	0.5-0.8 11.0-12.5	0.6-1.0 —	1.5-2.0 Titanium 0.03-0.12 Nickel 1.5-2.0 Vanadium	3.9 min	
				0.7-0.9	3.8-4.2	255-207

Chemical Composition and Hardness Numbers of High-speed Steels

Grade		5	Chemical composition, %	sition, %			Hardness number In annealed condition	number In condition
	Carbon	Chromium	Tungsten	Vanadium	Molybde- num	Cobalt	Impression diameter,	Hardness number, HB
Ь9	0.85-0.95	3.8-4.4	8.5-10.0	2.0-2.6	₹0.3	1	3.8-4.2	255-207
P9 Ф 2	1.4-1.5	3.8-4.4	9.01-0.5	4.3-5.1	₹0.4	1	3.55	≥ 293
P10K5Φ5	1.45-1.55	4.0-4.6	10.0-11.5	4.3-5.1	€0.3	5.0-6.0	3.55	₹ 293
P9K10	0.9-1.0	3.8-4.4	9.01-0.6	2.0-2.6	₹0.3	9.5-10.5	3.55	< 293
P14Φ4	1.2-1.3	4.0-4.6	13.0-14.5	3.8-4.1	≤0.4	1	1	
P18	0.7-0.8	3.8-4.4	17.5-19.0	1.0-1.4	€0.3	1	3.8-4.2	255-207
P18Ф2	0.85-0.95	3.8-4.4	17.5-19.0	1.8-2.4	₹0.5	l	3.55	≤ 293
P18K5Ф2	0.85-0.95	3.8-4.4	17.5-19.0	1.8-2.4	₹0.5	5.0-6.0	3.55	≥ 295
P18K5	0.65-0.77	3.6-4.5	17.0-18.5	1.0-1.4	0.3-0.6	4.5-5.5	3.7-4.1	269-217
P18K10	0.65-0.77	3.6-4.5	17.0-18.5	1.0-1.4	0.3-0.6	9.5-10.5	3.7-4.1	269-217
				_				
	_	_	_					

Critical Points of Steels upon Heating

	point, °C	Ac ₃	820	810	9//	820	820	840	838	820	775	800	850	830	830	780	880	830	006	098	820	840	830	820	800	800	780	
	Critical point.	Acı	735	735	715	750	763	. 750	165	740	750	720	730	745	740	<u> </u>	750	750	710	750	750	750	750	720	710	700	700	
caring		OI 4 UE	30XM	35XM	35X2MA	33XC	37XC	40XC	20XF	40XF	50XF	35XF2	18XLT	40XLT	18XFM	38XFM	27CF	35CF	27CLT	20XFC	25X LC	30XIC	35X FC	25H	30H	13H5A	21H5A	_
n nodn ei	ooint, °C	Ac,	794	780	992	150	1 120	Stools		820	840	815	815	815	800	77.1	771	840	820	820	840	800	790	930	815	860	840	_
2316 10 61	Critical point,	Acı	713	710	713	710		tructural		735	740	740	740	740	740	721	721	775	750	220	200	750	750	730	740	740	735	
	Grade	200	35 <u>Г</u> 2	40F2	45 <u>F</u> 2	5012	J153	Allon S.		15X	20X	30X	35X	38XA	40X	45X	20X	55C2	60C2	15XФ	20XФ	40XΦA	50X Φ A	15M	30M	15XMA	20X M	_
	oint, °C	Aca	Steels	2	874	874	863	854	040	813	810	800	780	770	774	992	752	743	863	854	820	790		765	750	. 830	802	-
	Critical point,	Acı	Structural		732	732	735	735	00/	732	730	730	725	725	727	727	727	730	735	720	720	720	720	726	724	720	715	_
	Grade		Carbon S		80	Q.	15	88	25	2	32	40	45	20	25	09	65	0/	151	.50 <u>r</u>	301	40ľ	. 20I	. 109		10Γ2	3012	

Table 36 continued

1 0	Critical	Critical point, °C		Critical point, °C	oint, °C		Critical	Critical point, °C
orane	Acı	Aca	Grade	Ac,	Acs	Grade	Acı	Ac,
ļ.								
30H3A	200	780	High.	High-speed Steels				•
15HM	725	800					Alloy Tool Steels	teels
40HM	715	276		810-840	1	-		
20XH	715	810	P18	820-840	-	X05	740	!
40XH	715	780		_		×	750	1
50XH	715	760	Ball-h	Rall-hearing Stop	_	X6	750]
12X H2	720	840		9	. ;	X.	750	!
12X H3	715	820	9 7 11	730-740		7X3	770	i
20X H3A	710	262	WX9.	730-755	890 Ac.	XĽ	750	[
30XH3	710	780	CIVII	730-765		9XC	785	!
37XH3A	710	760	01.8.10	740-760		Bl Be	750	1
12X2H4	715	820	Č			82 74 74	765	1
35XIOA	780	006	Stati	Stainless Steels		9XB1	750	1
38X MIOA	780	006	1X13	730	850	XBI YBI	755	1
35X M • A	740	840	2X13	820	920	ABO	00/	ı
13XHBA	710	820	3X13	008		3X 2B8	088	1
18XHBA	710	800	4X13	820	1200	4 X 8 B Z	820	18
33XH3MA	720	200	X17	088		4AB2C	2 i	840
18X HWA	710	800	9X18	830-865	1	57.57 57.53	7/2	820
35XHM	730	810	4X9C2	006	026	MHYC	067	2 i
40XHMA	710	790	4X10C2M	850	920	9X1.M 4XC	770	860
WALLI VOL	27	070				X 12	810	}
Frag.	Fros Cutting Ctool	,	Carbon	Carbon Tool Steels	st:	X12M	810	1
3-30-7	on Burne	1	77	064	000	X 09	750	!
A12 A20	735	866	y8-y13	730	ē 1	Ф	735	1
271	2	OF O	_	-	=		_	_

	_
Steel grades	Uses
	Carbon Structural Steels
10, 15, 20, 25, 30, Ст. 1, Ст. 2, Ст. 3, Ст. 4, Ст. 5, 15Г, 20Г	Parts of machines, non-heat-treated, working under light loads, such as pins, screws, nuts, studs, flanges, couplings, rollers, caps, etc. Parts of machines requiring high surface hardness and subjected to carburising: gears, spindles, arbors, pivots, shafts, sliding blocks, etc. Measuring tools: carburised horse-
	shoe gauges and templets. Steels Cr. 30 and Cr. 5 are used to manufacture
35, 40, 45, 50, Ct. 6, 30Γ. 40Γ. 50Γ, 30Γ2, 40Γ2, 45Γ2, 50Γ2	carburised simple-shaped parts Parts of machines working under medium loads. These steels are used after normalising, as well as after a special heat-treatment process aimed at improving their machinability (quenching and high-temperature tempering): screw fastenings, tie bars, levers, spindles, gears, shafts, supports, disks, connecting rods, couplings, rods, etc., pliers, hand vice, screw-drivers, spanners, jaws for lathe chucks, track tools. Steels with increased manganese content possess higher hardenability and following heat treatment, are harder and tougher than carbon steels with similar carbon content. This must be taken into account when choosing
55, 60, 65, 70, C _T . 7, 60Γ, 65Γ, 70Γ	steels Collets, coil and plate springs, spring washers, expanding rings, vehicle springs, construction and harness-mak- ing hand tools, Manganese steels are best as far as springiness is concerned
	Alloy Structural Steels
15X, 20X, 20X3, 15XФ, 20XФ, 15XГ, 20XГ, 18XГТ, 18XГМ, 13Н2А	Carburised extra-strength parts of ma- chines working at high speeds and under heavy loads: gears, shafts, arbors, worms, spindles, plungers, pushers, cams, piston pins, pivots, crosspieces of agricultural machinery, etc. Slim and intricate- shaped carburised measuring tools

Steel grades

Uses

15XH, 20XH, 12XM, 15XM, 20XM, 12XH2, 12XH3, 12X2H4-12X2H3MA, 13H5A, 13XHBA, 13XHMA, 18XHBA, 18XHMA, 20XH3A, 20X2H4A, 18X2H4MA

38XA, 40X, 45X, 50X, 40XF, 40XH, 45XH, 40XMФA, 50XМФA, 35XFC, 35XМФA, 30XM, 35XM, 35X2MA, 25X2MФA, 30M, 35XF2, 37XH3A, 40XHMA

45Χ Η ΜΦΑ

34X M, 34X H1M, 34X H2M, 34X H3M, 34X2H3M, 34X H3MФ, 35X MA, 33X H3MA

30X, 35X

33XC, 37XC, 40XC

27CF, 35CF

Large heavily loaded parts working at high speeds and subject to impacts. Following carburising and quenching they acquire a wear-resistant case and a tough and resilient core: gears, shafts, worm-shafts, jaw clutches, piston pins, connecting rods, parts of vibroshears, spindles of automatic lathes, etc. The latter five grades can also be used after heat-treatment

Heavy loaded heat-treated parts working under increased bending loads and low speeds, as well as cyanided parts and those subject to quenching and low-temperature tempering, working at high speeds and under heavy non-impact loads: gears, shafts, arbors, spindles, couplings, water-pump rotors, critical fastenings. The more the steel is alloyed, the more it is suitable for the manufacture of large critical machine parts

The heaviest loaded critical parts: gears, spindles, shafts, clutch pawls for engaging presses, etc. Fancy-shaped parts which are to be minimum in size

Large heavily loaded parts: spindles, torsion shafts, etc.

Turbine disks and shafts, and turbine rotors

Axles, arbors, rockers, gears, fastenings, autocrane gears, etc.

Extra-strength and tough parts with high torsional resistance: torsion shafts, cranks, axles, chain links, etc.

Axles, shafts, connecting rods, cranks, buckets of excavating machines and other parts requiring high toughness and wear resistance

Steel grades	Uses
25XFC, 30XFC	Arbors, gears, main rods, rods, critical fastenings, critical welded and stamped
35X IOA, 38X MIOA	parts, excavator buckets Highly corrosion-resistant nitrated parts working in abrasive media: spindles,
25Н, 30Н	shafts, rams, plungers, measuring tools Extra-hard critical parts having high resistance to impacts: crankshafts, axles, connecting rods, gears, rotors, etc.
	Ball-bearing Steels
ШХ6	Balls up to 13.5 mm, rollers up to
шх9	Balls from 13.5 to 22.5 mm, rollers from 10 to 15 mm
ШХ15	Balls from 22.5 to 50 mm, rollers from 15 to 35 mm
ШХ 15СГ	Rings with wall thickness from 20 to 30 mm Balls above 50 mm, rollers above 35 mm, rings with wall thickness above 20-30 mm Flat wire used for flexible rollers
ШХ10	Wear-resistant Steels
Γ13	Machine parts working under heavy impact loads: track shoes, and other excavator parts, crusher jaws and jaw plates, bucket lips, buckets, tracks of excavating machines, railroad switches and frogs
	Magnet Steels
EX EX3, E7B6 EX5K5, EX9K15M	Rotors of internal-combustion engines Magnets in telephones, electric meters, etc. Powerful magnets for navigational de- vices, etc.
	Electric Sheet Steels
91, 92	Armature and pole shoes of d-c electrical machines

Steel grades	Uses
Э3, Э 4	Stators and rotors of a-c electrical ma- chines, magnetic circuits of transform- ers, electromagnets for a-c apparatus (measuring instruments)
	Low-carbon Electric Steels
Э, ЭА, ЭАА	Magnetic circuits of relays, screens of audio-frequency apparatus, etc.
	High-permeability Alloys
45H, 50H	Cores of power transformers, choke coils, relays and elements of magnetic circuits operating at high inductances, mainly with no or negligible magnetising
50НП, 65НП	Cores of magnetic amplifiers, choke coils, commutating devices, elements of computers, etc.
38HXC, 42HC, 50HXC 79HM, 80HXC, 79HMA	Cores of impulse transformers and audio- and/or high-frequency communications systems, operating with no or negli- gible magnetising Cores of miniature transformers, choke
Torini, Corrido, Torrido	coils, relays as well as magnetic screens
0710 1710 0710	Corrosion-resisting Steels
OX13, 1X13, 2X13, 4X13, X14, 2X13H4F9, X14F14H, X14F14H3T, 1X13H3	Intended for operation in weak corrosive media below 30°C (aqueous solutions of salts, dilute nitric and various organic acids, food media). Steels are adequately resistant to corrosion by fresh water, steam, air
9X18, 1X17H2, 2X17H2, X17, 0X17T, X25T, X25H91, X17H7K0, 0X21H5F, X28 1X21H5T, 00X18H10, 0X18H10, X18H9, 2X18H9, X18H10E, 0X18H10T, X18H10T, 0X18H10T, 0X18H12T, X18H12T, 0X18H12E, X28AH, X18H9T, X17F9AH4, X17AF14, 0X10H20T2, X16H15M3B	Intended for operation in average corrosive media—nitric and organic (except acetic, formic, lactic, oxalic) acids, most of the solutions of organic and non-organic salts at various temperatures and in various concentrations

Steel grades	Uses
0X21H6M2T, X17H13M2T, 0X17H16M3T	For operation in highly corrosive media, organic acids in particular: formic, acetic, lactic, oxalic (below 5%), etc., as well as phosphoric (up to 32% P ₂ O ₅), containing fluoric compounds; boric acid with admixture of sulphuric acid (up to 1%); hydrosilicofluoric acids (up to 10% concentration)—for temperatures below 40°C
0X23H28M2T	Recipients for dissolving: sulphuric acid in low concentrations (up to 20%) below 60°C; phosphoric acid, containing fluoric compounds, and for other highly corrosive media
0Х23Н28М3Д3Т	Recipients for sulphuric acids in all concentrations, below 80°C; phosphoric acid (32-50% P ₂ O ₅), containing fluoric compounds; hydrosilicofluoric acid in increased concentrations (up to 25%) below 70°C
	Carbon Tool Steel
Y7 and Y7A	Cold chisels, cape chisels, simple hand dies, snaps, screwdrivers, lathe centres, plate cutters, augers, lettering dies, centre punches, dinking dies, compound pliers, cutting pliers, hammers, forging punches, chisels shaping dies, pneumatic tools, coppersmith's and tinner's tools, woodworking planing cutters, chip axes, twist drills, joiner's tongs and mason's tools
у 8, У8А , У8Г	The same as for grade V7 steel plus metal cutters, collets, simple dies for cold punching, plane bits
У9, У9А	Blades for saws, cold punching dies, centres, woodworking tools—twist drills, changeable cutters, planing cutters, gouges, axes, chisels, planing bits, stone chisels
У10, У10А, У11, У11А, У12, У12А	Small simple-shape metal cutters working under light loads: round-nose tools,

Steel grades	U'scs
У13, У13А	metal slitting and end-milling cutters, drills, reamers, screw taps, threading dies, files, saw blades, scrapers, cold punching dies, hand wood saws, crosscut and band saws. Measuring tools. The choice of one of the above-mentioned steel grades is dependent upon the intended use of a tool*) Scrapers, razors, engraving tools, drawing tools for the working of hard stones
	Alloy Tool Steel
7X3, 8X3	Hot upsetting dies, dies and punches for hot bending, working below 500°C and
9X	under slight impacts Rolls for cold rolling, cogging rolls, lettering dies, drifts, cold upsetting dies
X09 X, ШX15	and punches Cold punching dies, measuring tools Round-nose tools, milling cutters, drills, reamers, threading dies, screw taps, measuring tools, cold punching dies, wood saws. Grade IIIX 15 steel is superior to grade X steel
X05	Scrapers, engraving tools, sharp surgical
X12, X12М, X12Ф1	instruments, razors, drawing tools Broaches, thread-rolling dies and knurls, drawing tools, knives for presses and cold metal slitting shears, intricate- shaped dies for cold pressing Grade X12 steel is intended for tools which require no great impact strength and which are not subject to heavy
xr	impacts Long and fancy-shaped tools for which, at hardening, slight deformation is allowable such as cutting tools work- ing under light loads: milling cutters, screw taps, threading dies, reamers,

^{*} Tools subject to slight impacts in the course of operations should be manufactured from steels lower in carbon. Tools which require extra hardness and which suffer no impacts should be manufactured from high-carbon steels.

Steel grades Uses	
	broaches, etc.; measuring tools, cold pressing dies, press moulds for plastics, woodworking cutters, jig bushings
B1	Drills, screw taps, reamers, saw blades, roller knives
XB5	Cutting tools for working of extra-hard
5ХВГ	metals at low cutting speeds Intricate-shaped dies for cold and hot
5 A DI	pressing, which tolerate slight warping only
9XBΓ	Cutting and measuring tools of compli- cated shapes. Dies of complicated con- figuration for cold pressing which tol- erate slight warping only. Long knives for cold cutting of metals
ХВГ	Cutting and measuring tools of long and complicated configuration which tolerate slight warping only
Φ	Screw taps, threading dies, embossing dies for thin pieces, dies for the manufacture of bolts, nuts and rivets
8ХФ	Knives for cold cutting of metals, dies for cold trimming, centre punches
85 X Φ	Woodworking tools: saws, cutters, knives, etc.; paper-mill knife-tools
4XC	Dies for hot pressing, knives for cold and hot cutting of metals, pneumatic tools
6XC	Knives for cold and hot cutting of metals, small dies for cold pressing, pneumatic tools
9XC	Cutting tools, mainly subject to grinding in the manufacturing process: drills, centre points, milling cutters, reamers, counterbores, screw taps, broaches. Dies for cold pressing
XIC	Measuring tools
4X B2C	Press moulds for low-temperature pressure casting, hot pressing dies, knives for hot cutting of metals, pneumatic tools
5X B2C	Dies for cold and hot pressing, knives for cold and hot cutting, press moulds for casting low-melting alloys, threading rolls

Steel grades	Uses	
6X B2C	Cold and hot pressing dies, threading rolls, woodworking tools	
3X2B8	Press moulds for pressure casting of aluminium and copper alloys, hot pressing dies and knives for not cutting of metals working under heavy-duty and high-temperature conditions	
4X8B2	Press moulds for casting non-ferrous alloys, dies for hot pressing working under heavy-duty and high-temperature conditions	
5XHM, 5XFM, 5XHT, 5XHC, 5XHB, 5XHCB	Forging dies	
P9, P18	High-efficient cutting tools which retain their cutting ability at a temperature of up to 600°C. Grade P9 is more difficult to grind than the P18 grade Efficiency, wear resistance, and red hardness of steels listed below with their particular features and uses are higher than those of the P9 and P18	
Р9Ф5	grades Chiefly used for the manufacture of finishing tools. A difficult-to-grind ma- terial	
P9K10	Heavy-duty cutting tools. Drills for working of heat-resistant alloys and hard-to-work materials This steel has a tendency to decarburise	
Р14Ф4	Cutting tools for the working of extra- hard materials. A difficult-to-grind material	
Р18Ф2	Cutting tools for the working of materials of different hardness numbers, and of stainless and heat-resistant alloys	
Р18К5Ф2	Takes grinding satisfactorily Cutting tools for the working of hard- to-work materials	
P18K5, P18K10	Satisfactory grindability High-speed cutting tools. Retain hardness at temperatures up to about 670° C	

Heat-resisting and Non-scaling Steels *

Grade	Use	Heat-re- sisting	Non- scaling
		up to, °C	
X5 [.]	Pipes		650
X15М, X5ВФ	Parts of oil-refinery equipment	600, v.l.d.	650
X6CM	Pipes, parts of pumps, valves	660, 1.d.	700
1Х8ВФ	Pipes for furnaces, apparatus and mains of oil-refining plants		
4X9C2.	Exhaust valves of internal com-	500, 1.d. 650. l.d.	650
4X10C2M (ЭИ107)	bustion engines, pipes of re- cuperators, heat exchangers, fire bars	050, 1.a.	850
1Х12Н2ВМФ (ЭИ961)	Compressor disks, blades and various heavy duty elements	600, s.d.	750
1XB, 2X13	Parts of machines and hard- ware requiring high plasticity: turbine blades, valves, bolts, etc.	500, l.d.	750
0X 13	Ditto	_	
3X13, 4X13	Machine parts requiring high hardness: cutting, measuring and surgical instruments, springs, hardware		
3X13H7C2 (ЭИ72)	Valves for automobile engines	_	950
1 X 13 H3	Heavy duty parts working in fresh and/or sea water	: -	_
1X17H2 (ЭИ26-8)	High-strength steel for blades of axial compressors	. -	: :::::
9Х18 (ЭИ229)	Bearings for oil-refinery equip- ment and various extra-hard wear-resistant parts		1 1 5 1 5 1 5 1 5 1 5 1 5 1 5 1 5 1 5 1
Х6С10 (ЭИ428)	Parts of boiler installations	_ :	700
IX11МФ	Blades of steam boilers	550, l.d.	750
IX12ВНМФ (ЭИ802)	Rotors, disks, blades, bolts	570, l.d.	750
<u> </u>	si pamban lisa a protes 📗	·	

^{*} The terms "heat-resisting" and "non-scaling" when referring to steel are rather loosely used and it is necessary to finalise their meaning.

Heat-resisting steels are those that retain much of their strength at elevated

Non-scaling steels are those that are highly resistant to corrosion at elevated temperatures.—Tr.

Grade	Use	Heat-re- sisting	Non- scaling
		up to,	°C
2X12BMБФР (ЭИ993)	Rotors, disks, blades, bolts	600, l.d.	750
1 X 12B2MФ (ЭИ756)	Rotors and disks of steam tur- bines, blades	600, l.d.	_
1X12СЮ (ЭИ404) X14 (ЭИ241)	Valves of engines and various other parts Parts turned in automatic	_	950
X17	lathes—screws, nuts, threaded parts Equipment of nitric acid plants,		_
0Х17Т (ЭИ645)	hardware. Not recommended for welded structures The same as for the X17 steel	-	900
X 25T (ЭИ439)	and also recommended for welded structures Structures (including welded	_	900
	structures), working in more corrosive media than the X17 steel. Thermocouple shields, pipes for pyrolisis installations, etc.		1050
Х 18СЮ (ЭИ484)	Pipes for pyrolisis installations,		
X 28 (ЭИ349) X 28АН (ЭИ657)	various equipment, etc. Same uses as for steel X25T, as well as for media containing acetic acid. Pipes for pyrolisis installations, heat	-	1050
2X13H4Г9 (ЭИ100), X18H9, 2X18H9,	exchangers Parts manufactured from sheets and strips welded by point welding. When welded by other methods, intercrystal-	_	1150
X18H10E (ЭИ453), X14ГМН, X17НГ14,	line corrosion is possible. Furnace fittings, etc.	-	800
X17Г9АН4 X15Н9Ю (ЭИ904), X17Н7Ю (ЭИ973)	Extra-strength steel working in the air		-
			÷

Grade	U s e	Heat-re- sisting	Non- scaling
		up to, °С	
2X 17H2	The same as above, but in corrosive media		10.00
0Х2ЭНМ14С2 (ЭИ 7 32)	Pipes	_	1050
X20H14 C2 (ЭИ211)	Furnace conveyors, carburising boxes	_	1050
0X21H5T (9И53), 1X21H5T (9И811), X18HЮT, X18H9T,	Manufacture of welded apparatus for various branches of industry. Furnace fittings, parts of exhaust manifolds	-	800
X18H12T, X14T14H3T 9X18H10T (ЭИ914), 0X18H11 (ЭИ684), 00X18H10 (ЭИ842), 0X18H12T, X18H12T,	Welded apparatus operating in higher corrosive media than those manufactured from steel X18H10T. Parts of furnace equipment	-	800
0X18H12B 0X21H6M2T (ЭИ54), X17H13M2T (ЭИ448), X17H13M3T (ЭИ432), 0X17H16M3T	Parts operating in boiling phosphoric, formic, lactic, acetic acids and other highly corrosive media	_	-
(ЭИ580) 0X 23H28M2T (ЭИ628)	Welded structures operating in hot phosphoric and sulphuric acids, below 60°C	_	_
0X 23H 28M3Д3Т (ЭИ943)	hot sulphuric acids (below 80°C), resistant also to hy-		
4X 18H25C2	drofluoric acids Furnace conveyors and other	_	1100

Grade	Use	Heat-re- sisting	Non- scaling
" Glade "	Use .	up to.	
0X23H18, X23H18(ЭИ417)	Pipes and parts for methane conversion and pyrolisis in-		
V.051114D74D	stallations, for gas mains, combustion chambers	1000, d.	1050
X 25H16Г7AP (ЭИ835) X 25H20C2	Parts of gas mains, manufac- tured from sheet steels Boiler brackets and supports,	950, s.d.	1100
(ЭИ283) 1X25H25TP (ЭИ813)	pipes for pyrolisis installations Parts of different systems		1200 1100
Х Н38ВТЮ (ЭИ787) Х Н60Ю	Turbine and compressor disks and blades Parts of gas mains, and equip-	750, s.d.	900
(ЭИ559A), XH70IO (ЭИ652)	ment	850-1100, s.d.	1200
(ЭИ052) X H 78 Т (ЭИ435) X H 70 (ЭИ442) 4 X 12 H 8 Г 8 М Ф Б (ЭИ481)	Ditto Equipment Turbine disks	650, s.d.	1100 1150 7 50
X 12H22T3MP (ЭИЗЗ)	Turbine disks	75 0, s.d.	850
0X I4H28B3T3ЮР (ЭИ786)	Ditto	750, s.d.	900
X 12H20T3P (ЭИ695)	Turbine parts	700, s.d.	850
1X [4H16Б (ЭИ694), 1X 14H16БР	Superheater pipes and extra- high pressure pipelines	650, l.d.	850
(ЭИ694Р), 1Х14Н18В2Б		650, l.d.	8 5 0
(ЭИ695), 1Х14Н18В2БР		700, 1.d.	850
(ЭИ695Р) 1Х14Н18В2БР1	Turbine rotors, disks and	700, 1.d.	850
(ЭИ726) 4X 14H14B2M (ЭИ69)	blades Engine valves, pipeline fittings	700, 1.d. 650, 1.d.	850 850
4X15H7Г7Ф2МС (ЭИ388) 1X16H13M2Б	Blades of gas turbines, fasten-	650, s.d.	800
(ЭИ680)	Forgings for disks and rotors, blades	600, v.l.d.	850
	,	i	

Grade	Use	Heat-re- sisting	Non- scaling
		up to, °	c
X 16H 15M3Б (ЭИ847)	Superheater pipes and high- pressure pipelines	350, v.l.d.	850
3X19H9MBB	Rotors and disks, bolts	600, v.l.d.	800
(ЭИ572)	1	000, 7	1
X Ĥ3 5B T´	Gas-turbine blades, disks, ro-)
(ЭИ612)	tors, fastenings	700, l.d.	
X Н35ВМТ (ЭИ692)	Turbine blades, fastenings	700, 1.d.	900
(301692) X H35BTP	Gas-turbine parts manufactured		
(ЭИ725)	from thick sheets	750, v.l.d.	900
X H38BT.	Parts manufactured from sheet	,	
(ЭИ703)	steel and working at medium		
VIIIOD MIOT	pressures	600-950, s.d.	1050
х H70ВМЮТ (ЭИ765)	Blades, fastenings	750, v.l.d. 800, l.d.	1000
хн70ВМТЮ	Turbine blades	850, s.d.	1000
(ЭИ617)	1		
х Н80ТБЮ	Turbine blades, fastenings	700, v.l.d.	1050
(ЭИ607) У 170МВ ТЮБ	Turbine blades	950 0 4	1000
Х Н70МВТЮБ (ЭИ598)	Turbine blades	850, s.d.	1000
X H67MBTIO	Turbine blades	800, 1.d.	1
(ЭИ445P)	J	850, s.d.	1000
х Н75МБТЮ	Turbine parts manufactured	200 250 1	1050
(ЭИ602)	from sheet steel Turbine disks and blades	800-950, s.d.	1050 1050
ХН77ТЮ (ЭИ437А)	Turbine disks and blades	750, s.d.	1090
XH77TЮP	Ditto	750, s.d.	1050
(ЭИ437)			
XĤ60B	Turbine parts manufactured		
(ЭИ868)	from sheet steel	800-1000, s.d.	1200
		3.4.	1200
	1	1]
			}
		1	l

Notes. 1. High-temperature resistance is defined as the temperature of the beginning of intensive scaling.
2. Short duration of service of an element (s.d.) denotes a life span from 100 to 1,000 hours, while long duration (l.d.) implies an interval from 1,000 to 10,000 hours; very long duration (v.l.d.) signifies a considerably longer period, extending usually from 50,000 to 100,000 hours.

Chapter IV

HEAT-TREATMENT OF STEEL

1. HEATING OF STEEL

A heat-treatment procedure consists in heating metals and alloys to a certain temperature, holding and cooling them at various rates with the aim of altering their structure and properties.

Each of the heat-treatment processes comprises the following opera-

tions:

1. Heating to a prescribed temperature.

2. Holding for heat saturation and completion of structural changes.

Cooling at a requisite rate.

The rate of heating depends on chemical composition, shape and size

of the work.

A. When using flame or electric furnaces the following work may be charged into a furnace preheated to a given temperature: (1) articles from low-carbon steels, regardless of their structure, shape and size; (2) small- and medium-size articles from other grades of steels which have been annealed or normalised.

Articles with sharp angles, large-size articles, as well as any articles subject to repeated hardening should be heated up gradually, after be-

ing charged to a cooled furnace.

B. When using salt or lead baths, the following articles may be immersed in a bath preheated to a given temperature: (1) articles from low-carbon steels, regardless of their structure, shape or size; (2) articles from other grades of steel having simple configuration, and with no abrupt changes in section (rolls, rollers, tools with welded bits, etc.).

The heating of all other articles should comprise a preheating procedure to about 500-800°C. Preheating can be carried out in separate furnaces as well as in bath-type furnaces hy repeated (depending on size) immersion of the article in molten salts (see Table 37) or lead for 2-3 seconds. Two preheatings, up to 600 and 850°C, are necessary for articles from high-speed steels having a highly complicated configuration.

Approximate duration of heating in various furnaces may be found

by referring to Table 38.

The duration of heating is influenced by many factors, e.g., the number of articles charged into the furnace, the useful volume of the furnace, the arrangement of articles in the furnace (in bulk or in such a way that each piece is flushed by the hot gases), charging tempera-

fure, temperature uniformity, looseness of furnace door, etc. Practically, the duration of heating is to be determined on the spot with due consideration of the all above-mentioned factors.

Heating temperature, holding time, and cooling rate are dependent upon the steel grade and the heat-treatment process employed.

Table 37
Salt Mixtures Most Frequently Used for Heating in Salt Baths on Hardening

Mixture components	Chemical formula	Weight,	Melting point, °C	Recommended temperature ranges, °C
Common salt Soda ash	NaCl Na ₂ CO ₃	50 50	560	590-900
Common salt Calcium chloride	NaCl CaCl ₂	50 50	595	630-850
Common salt Barium chloride	NaCl BaCl ₂	22.5 77.5	635	665-870
Common salt Potassium chloride	NaCl KCl	44 56	660	72 0-900
Common salt	NaCl	100	800	830-1100
Barium chloride	BaCl ₂	100	962	1100-1350

Table 38

Approximate Heating Time for Articles in Various Types of Furnaces

	Furnace	Heating time for 1 mm of dia or thickness of article, so		m of diameter rticle, sec
	temperature,	Round	Square	Rectangular
	°C	section	section	section
Electric furnace	800	40-50	50-60	60-75
Oil furnace	800	35-40	45-50	55-60
Salt bath	800	12-15	15-18	18-22
Lead bath	800	6-8	8-10	10-12
Salt bath	1300	6-8	8-10	10-12

2. OXIDATION AND DECARBURISATION ON HEATING

The furnace gases affect the heated steel by oxidising and decarburising it. The combination of the oxygen of the furnace gases with iron results in the formation of scale, while its combination with steel carbon causes decarburisation.

The best method to protect the work from oxidation or decarburisation is the use of muffle furnace provided with controlled or protective

atmosphere.

Controlled atmosphere composed of CO2, CO, H2, and N2 is mostly

used in the practice of heat-treatment shops.

Controlled atmospheres may be produced by incomplete combustion of coke-oven, generator, natural and various other gases, purified from CO2 and thoroughly dried to remove water vapours.

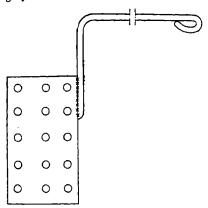


Fig. 5. Perforated cylinder used for deoxidising salt baths

If no installations are available for the production of controlled atmosphere, the articles to be heat-treated are packed into containers filled with insulating solid material consisting of: (a) used carburiser; (b) cast-iron chips, fresh or with the addition of 50% of the used ones; (c) burned asbestos, etc.

Pure charcoal is not an effective means of protection of high-carbon

steels against decarburisation.

Tools from high-speed and high-chromium steels are protected against decarburisation by preliminary immersion in a saturated solution of

Upon heating, the borax melts and spreads itself in a thin layer over the surface of the article, thus safeguarding the latter from reaction with the oxidising medium.

To protect heated parts against oxidation, some charcoal is placed

in the furnace near the charging door.

Smaller parts are protected against scaling by immersion in the solution prepared as follows: I part of calcium chloride is dissolved in 25 parts of water; after boiling the solution is allowed to cool, then

2 parts of ground fluorite are added to it.

To avoid decarburisation, salt baths should be deoxidised 2-3 times per shift by adding, to the molten salt, crushed 75% ferrosilicon (200-300 g per one CN3-75 type bath), boric acid, and 40 to 50 g of potassium ferrocyanide or potassium ferricyanide per bath. The deoxidation of a barium chloride bath is effected at 1300-1320°C. With the deoxidising agent charged, the bath is allowed to stand for 15-25 minutes, after which the slag is skimmed and the temperature brought down to the required level.

Deoxidation of salt baths with charcoal is widely gaining recognition. A perforated cylinder with a welded handle (Fig. 5) is filled with charcoal and immersed in the molten salt. As the cylinder with charcoal is immersed, big tongues of flame appear and then fade away gradually (within 15-20 minutes) and disappear. This indicates the end of the deoxidising process. It is recommended to deoxidise the baths in

the above manner 2-3 times per shift.

Salt baths are checked for deoxidation with the aid of ordinary safety razor blades. A blade heated for 3-5 min. in a properly deoxidised bath and quenched in water should rather break than bend.

The deoxidation of alkaline baths is effected by charging potassium

ferrocvanide on the bath surface in the amounts equal to 0.2-0.3% of the alkali weight. The bath should be mixed for 15-20 minutes, after which the slag is skimmed.

A layer of charcoal 15-20 mm thick is charged on the surface of lead baths to protect the latter against oxidation.

Files are safeguarded against decarburisation by luting (Table 39).

Table 39 Lutes Used to Protect Files Against Decarburisation on Hardening

Material	Compos	sition, %
-	No 1	No. 2
Crushed horn	30 3 38 12 17	45 4.5 45 — 1.5

A Leningrad plant uses abrasive powder No. 100 as a protective lute. The files are flushed with joiner's glue, coated with the abrasive powder to a layer 1.0-1.5 mm thick, and dried. A simple reliable protective means is a prehardening pickling of files for 10-15 minutes in an acid solution composed of (% by volume): sulphuric acid (concentrated) — 7%, nitric acid (concentrated) — 7%, water—86%. After pickling, the files are dried near the furnace.

3. ANNEALING

Annealing is used to decrease hardness, to relieve internal stresses, to correct structure, to eliminate strain hardening, to improve machinability. Data given in Table 40 may be useful to determine the type of annealing necessary.

Recrystallisation Annealing of Cold-Worked Steels

Grade of steel	Heating temperature, °C	Cooling medium
У7-У13, У8, Ф. В1, 08-70, 15Г-70Г, 10Г2-50Г2, 70С2ХА, ЕХЗ	680	Air
ШХ6-ШХ15, ШХ15СГ, 40Х, 38ХА, 20Х3, 40ХФА, 38ХМЮА, 30ХГСА, X, 9X, X05, 7Х3, 8Х3, 9ХС, ХГС, 4ХС, ХГ, 4ХВ2С, 5ХВ2С, 6ХВ2С, ХВГ, ХВ5	700	Air
P9, P18	760-780	Water
X12, X12M, X12Φ1, 9X18	730-750	Water
X9C2, X10C2M, X17	850	Water
1X13, 2X13, 3X13, 4X13	720	Air
0X18H10, 1X18H9, 1X18H9T, 2X13H4F9, X18H11B, 2X18H9, X15H60, X20H80, X20H80M3, 4X14H14B2M, 4X18H25C2	850	Water
12XH3A, 12X2H4A, 20XH3A, 30XH3A, 37XH3A, 40XHMA	660	Air
18XHBA, 25XHBA, 20XH4ΦA	640	Air
X13Ю4, 1X17Ю5, 0X17Ю5, 1X25Ю5, 0X25Ю5, X25, X27, X28, X25T	700	Water

Note: Holding at recrystallisation temperature on open heating can last up to one hour.

A Rough Guide for the Choice of Annealing Treatments

Type of annealing	Steels subject to the process	Heating temperature, °C	Cooling rate	Purpose
Full annealing	Hypoeutectoid, eu- tectoid, small- and medium-size steel castings	Ac ₃ + 20-30, Ac ₁ + 20-30	Down to 500-600° C at a rate of: 2. Stress relieving 2) 50-100° C per hour for arbon steels; 2) 20-60° C per hour for alloy steels	1. Softening 2. Stress relieving 3. Structure improve- ment
Process anneal- ing	Hypoeutectoid	Between Ac ₁ and Ac ₃	Ditto	1. Softening 2. Stress relieving
Spheroidising (globular pearlite anneal-ing)	Hypereutectoid	$Ac_1 + 10-20$	Down to 500-600° C at a rate of 20-30° C per hour	Down to 500-600° C at a rate of 20-30° C per 2. Improvement of machin-ability (cutting) 3. Improvement of cold broaching 4. Preparation of structure for subsequent hardening
Isothermal annealing	Chiefly for alloy steels	1. Ac ₁ + 20-30; 2. Ac ₁ - 20-30	1. Ac ₁ +20-30; Rapid cooling down to The same 2. Ac ₁ -20-30 Ar ₁ -20-30°C, holding an the said temperature followed by air cooling	The same as for full annealing

Purpose	To eliminate coarse cast structure and segrega- tion	1. Softening 2. Stress relieving 3. Improvement of machinability	Regeneration of structure after cold working
Cooling rate	With the lurnace	With the furnace or in 2. Stress relieving the air 3. Improvement of machinability	Ref. to Table, § 3
Heating temperature, °C	Ac ₃ + 150-250	Ac ₁ —15-30	Ref.
Steels subject to the process	Large steel castings Ac, + 150-250 and ingots	High tempering Hypereutectoid and low-tempera- fure anneal- ing)	All grades of steels following cold working
Type of annealing	Interdiffusion annealing (homogeni- sation)	High tempering (low-tempera-ture anneal-ing)	Recrystallisa- tion annealing

Notes. 1. Spheroidising is also carried out as a step or cyclic annealing process according to the diagram presented in Fig. 6.

2. Rapid cooling from the temperature of initial heating down to the temperature of sich remains a fine temperature of sich remains to a furnace heated to a predetermined temperature or by cooling in the lumace with its door open.

3. High temperature of softening operation for chrome-nickel-tungsten of step (pendulum)

4. NORMALISING

Normalising is used to correct the structure of overheated steel, to relieve internal stresses and to improve machinability of structural carbon and low-alloy steels.

For normalising, the articles are heated to 30-50°C above the Ac,

point and then cooled in still air.

When it is necessary to increase the depth of hardening of carbon tool steels, they are also normalised.

Hardness numbers of carbon steels in annealed and normalised conditions are presented in Table 41.

Table 41

Brinell Hardness Numbers of Steels in Annealed and Normalised Conditions

	Structural steels		
Condition	Low-	Medium	High-
	carbon	carbon	carbon
Annealed	125	160	185
	140	190	230

- 5. HARDENING

Hardening is used to impart maximum hardness to steel. Prior to hardening of hypoeutectoid steel, the latter is heated to 20-30°C above Ac_3 , while eutectoid and hypereutectoid steels are heated to 20-30°C above Ac_1 and then, after holding, rapidly cooled. The work is cooled in water, oil or air depending on its chemical composition, size, and shape.

Cooling of work on hardening. The rate of cooling on hardening should be sufficient to ensure the transformation of austenite to martensite. In practice, the required rate of cooling is attained by cooling carbon steels in water; alloy steels, depending on their composition, are cooled in water, oil, kerosene or between plates, while certain grades

of high-alloy steel are air-cooled (Table 42).

Table 42 Cooling Rates of Steels in Various Quenching Media (according to S. S. Steinberg)

Cooling media	Cooling rate per second at	
	650-550° C	300-200° C
Water at 18° C Water at 26° C Water at 50° C Water at 74° C Solution of 10% of sodium hydrate in water at 18° C Solution of 10% of common salt in water at 18° C Solution of 10% of soda in water at 18° C Solution of 10% of sulphuric acid in water at 18° C Soap water Mineral oil Kerosene Copper plates Iron plates	600 500 100 30 1200 1100 800 750 30 100-150 160-180 60 35	270 270 270 200 300 300 270 300 200 200-50 40-60 30 15

Table 43 Oils Used for Isothermal and Step Hardening and Tempering

Oil	Flash point, °C	Oil	Flash point, °C
Spindle oil 2	170 180 180	Cylinder oil 2 Viscosine 3	215 240 290 300 320

Oil retains its hardening ability from 20 to 150°C.
Copper and iron plates are better when hollow and water-cooled.
When using massive plates, their surface should be coated with a thin film of oil to add to their cooling ability.

When hardening work from high-carbon steels and of very complicated configuration, 50% aqueous solution of caustic soda (sp.gr. 1.36) at 20-60°C is used as a hardening medium. The quenching in the above solution is to be effected under the hood.

When employing isothermal or step hardening, hot oil or low-melt-

ing salts are to be used (Tables 43, 44).

Fig. 7 shows the manner in which the work should be immersed in the cooling medium to prevent warping.

Table 44
Compositions of Salts Most Frequently Used for Hardening and Tempering

Constituents	Formula	Weight,	Melting point, °C	Temperature recommended for use, °C
Potassium nitrate	KNO ₃ NaNo ₂ KNO ₃ NaNO ₃ NaNO ₃ KNO ₃	56 44 50 50 100	153 } 220 317 337	175-500 245-500 325-500 350-500

When considering hardening, the following methods are to be distinguished:

1. Hardening in one cooling medium. The heated work is cooled down to below 150-100°C in one medium. This is the chief procedure for hardening oil-cooled alloy steels and carburised carbon steels cooled in water: this method is also used for hardening simple-shaped work from

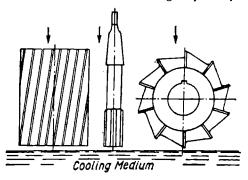


Fig. 7. Direction of immersion of tools in a cooling medium.

medium- and high-carbon steels. Carbon-steel articles up to 6-7 mm in

diameter are oil-hardened.

2. Hardening in two cooling media. The heated work is cooled in water down to 200-250°C, then it is quickly transferred to an oil bath for further cooling. Approximate duration of cooling in water prior to transfer is about 1-1.5 sec for every 5-6 mm of diameter or thickness. This is the main hardening procedure for carbon tool steels.

Table 45 Effect of Salt Temperature on Cooling Time of Heated Steel Cylinders

<u> </u>		Salt temperature, °	C
Diameter of	205	260	315
cylinder, mm	Time necessary	to bring the cyling temperature, min	ider to the salt
25 50 70	5.0 8.0 13.5	4.0 7.0 12.5	3.5 6.0 11.5

3. Jet hardening. Cooling is by a water jet or shower, this method

being used for work with through or blind holes.

4. Isothermal hardening. The heated work is cooled in hot oil, molten salt or metal (lead, tin) at 20-30°C above the beginning of the martensitic transformation for the grade of steel considered (Table 45). Once the transformation of austenite is over, the work is air-cooled. As a result of isothermal transformation, acicular troostite is formed. Tempering, following isothermal hardening, is optional.

This method is used to harden work of complex shape with a view

to avoid cracks and warpage.

5. Step hardening. 1) The heated work is cooled in a hot medium to 20-30°C above the martensitic transformation point to even the temperature throughout the cross-section, following which it is aircooled. After the work is removed from the heated medium, it can be straightened before the austenite transformation is over. This method is used for the hardening of complex-shape articles from carbon tool steel up to 8-10 mm in diameter.

2) The heated work is cooled, first in a hot medium at 150-180°C, to balance its temperature, then in the air. This method prohibits the straightening operation but the number of rejected pieces due to cracks and warpage, as compared to the usual hardening procedure, is consid-

erably reduced.

6. Tempering combined with hardening. The working edge of the heated work is water-cooled, then removed and tempered to the colour required. This method is mainly used for percussive tools made of carbon steel.

7. Hardening through air-cooling. The work, heated above the required temperature, is air-quenched down to normal hardening temperature, after which it is hardened. This method is widely employed for carburised work when the latter is hardened directly from the case-hardening temperature (chiefly used for dies manufactured from grade 5XHT steel, etc.).

8. Bright hardening. The work, heated to hardening temperature is cooled in molten alkalis, after which it is flushed, first in hot water, then with a heavy jet of cold water and, finally, immersed in aqueous solution of 1.5% sodium nitrite and 0.3% soda ash to prevent corrosion. The bright-hardening technique, while displaying all the merits of the isothermal and step hardening procedures, is superior to them on the following points: the work remains bright, warping is minimised and higher mechanical properties are obtained.

Salt baths used for heating prior to bright hardening should confain no barium chloride as its presence spoils the alkali bath. The best composition is 100% KCl or a mixture of 50% NaCl and 50% KCl. Table 46 gives chemical compositions of alkali baths and their opera-

tion temperature charts.

Table 46
Chemical Composition of Alkalis Used for Bright Hardening

Bath composition	%	Melting point,	Temperature range of use, °C
Potassium hydrate	75 25 6	130	150~250
Potassium hydrate Sodium hydrate	63 37	159	180-350
Sodium hydrate	100	322	350-700

Baths working above 250°C are deoxidised, when necessary, by additions of potassium ferrocyanide (0.2-0.3% of alkali weight). Parts treated in non-deoxidised baths blacken. Baths working below 250°C do not require deoxidising. In case the bath hickens, small amounts of 30-50% aqueous solution of potassium hydrate are added, at below 200°C, in minute portions, with the aid of a long-handled ladle, in order to fluidise the bath and to enhance its hardening ability.

Hardening power of low-temperature alkali baths depends on the amount of water added to it. Maximum hardness is displayed by articles processed in baths with 10% of water. A greater amount of water

causes soft spots. No eruptions are observed when water is added. The amount of water in the bath is determined by measuring the hardness of samples hardened under similar conditions. Fig. 8 presents hardness curves of samples from grade 45 steel 25 mm in diameter and 50 mm in length.

The alkali bath should be agitated by an impeller at 800 to 1,400 r.p.m. or by a rotating worm at 600 to 800 r.p.m. The bath

should be cleaned periodically from foam and sediments.

When bright-hardened parts rust as a result of inadequate flushing, the defect may be corrected without change of the part's size, by heat-

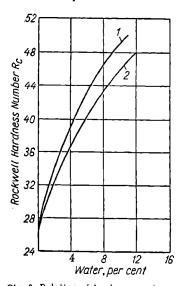


Fig. 8. Relation of hardness number of a sample from steel Cr. 45 to the amount of water introduced into a low-temperature alkall bath:

1—bath temperature equal to 200°C;

2—bath temperature equal to 250°C

ing it in a special solution at 70-80°C for 30-40 min (to remove rust) with subsequent thorough washing by a strong water jet and immersion in a solution of sodium nitrite. The solution is prepared as follows: 100 grams of chromic anhydride and 110 grams of orthophosphoric acid (sp.gr. 1.6) are dissolved in 200 cm³ of water, after which water is added to obtain 1 litre of solution.

9. Sub-zero hardening (sub-zero treatment). The structure of hardened carbon and alloy steels always includes some amount of austenite. In articles tempered above 200-250°C, this austenite turns to tempered martensite upon heating. In articles tempered below 200°C, the austenite is "frozen dead" and remains unaltered in the steel.

Austenite is remarkably stable in some high-alloy steels, and on tempering, its transformation to martensite is incomplete. Maximum austenite decay can be achieved through additional cooling to below zero after hardening. Table 47 lists the sub-zero treatment temperatures for some grades of steels. Sub-

zero treatment should be carried out immediately after the article has cooled down to room temperature, since maturing at room temperature makes austenite very stable. Time interval from hardening to sub-zero treatment should not exceed 1-3 hrs while it should be still shorter for carbon tool steels. Sub-zero treatment should be carried out prior to tempering. It has been found that sub-zero treatment does not improve the cutting ability of high-speed steels. Rapid cooling should be avoided in order to prevent cracks. Liquid nitrogen, liquid oxygen, solid carbon dioxide, freon, etc., may be used as cooling media.

Sub-zero Treatment Temperatures for Some Grades of Steels to Enhance Austenite Transformation

Grade of steel	Treatment temperature, °C	Grade of steel	Treatment temperature, °C
У8 У10 У12 ХГ ХВГ	0 0 20 50 80	IIIX15 X12Ф1 18XHBA (carburised) 12X2H4A (carburised)	-30 -70 -85 * -85 *

^{*} For the carburised layer.

10. Surface hardening. It is effected by heating the work quickly in an electrolyte, oxygen-gas flame, by contact method or high-frequency current.

The electrolyte used is a 5-10% aqueous solution of soda ash or pot-

ash. Direct current at 180 V, as a minimum, is used.

There are three methods of heating in an electrolyte:

(a) butt heating used to heat the tip of an article;

(b) surface-heating method used to heat the work surfaces;

(c) heating in which the work is gradually passed through an electrolyte. The lower part of the work is insulated by a bushing and is not heated.

11. Surface hardening by oxy-gas flame (flame hardening). This procedure consists of local heating of the part to be hardened by means of an oxy-gas flame, and of subsequent cooling. The depth of hardening ranges from 1 to 6 mm, and cooling is ensured by a sprayer.

There exist the following methods of surface hardening by oxy-gas

flame:

1. Stationary. The work and the burner are stationary, the former being cooled after the withdrawal of the burner.

2. Rotary. The work rotates at 100-200 r.p.m. The burner is stationary. The work is cooled following the extinguishing of the burner.

3. Flat-translatory. The work moves in a straight line, while the burner is stationary, or vice versa. Cooling is continuous, being applied at a distance of 10 to 20 mm from the burner.

- 1. Parts with negligible hardened areas, such as teeth of chain gear, cams, lathe centres, short end-cutting tools, etc.
- 2. Cylinder-shaped parts up to 20 mm in diameter, e. g., necks, journals of arbors and axles, low-module gears, etc.
- 3. Flat parts of considerable length: frame guides, shears, etc.

- 4. Rotary-translatory. The part rotates slowly. The burner is stationary. Cooling is continuous, being applied at a distance of 10 to 20 min from the burner.
- 5. Helical-translatory. The work rotates slowly. The burner moves in a straight line. Cooling is continuous, being applied at a distance of 10 to 20 mm from the burner.
- 6. Combined. The ring-shaped burner moves in a straight line along the axis of the part which rotates rapidly inside the burner. Cooling is continuous, being applied at a distance of 10 to 20 mm from the burner.

4. Cylinder-shaped parts with diameters exceeding 200 mm: wheel bands, mounting rings. crane wheels, large-diameter rollers, etc. A tempered streak borders on the hardened area.

5. Cylinder- and helix-shaped parts: worms, screws, etc. An annealed streak is formed at the point where the helices meet.

6. Slim cylinder-shaped parts: shafts, spindles, rods, etc.

Gas burners used for prehardening heating should follow the profile of the work to be hardened (Fig. 9). This ensures uniform heating and

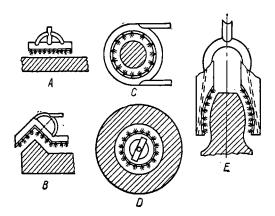


Fig. 9. Configuration of hardening inductors: a - flat rectilinear configuration; b - angle configuration; c-annular configuration for surface hardening of cylinder-shaped parts; d-annular configuration for hardening of internal surfaces; e - modular configuration

completion of the hardening process in one run. Cooling is ensured by water dispersed by a sprayer. Water consumption is 0.4-0.5 litres/min per 1 cm² of the surface being cooled. The less the distance from the burner to the hardened work and the less the speed, the greater the hardened depth.

Stationary or rotary hardening procedures for alloy steels are to be followed by oil-cooling. Large parts are hardened in special hardening machines which ensure the requisite direction and speed of the work and the burner. Hardening in special machines permits automation, use of control instruments, etc.

12. Hardening by high-frequency current. When a high-frequency current flows through a solenoid-shaped inductor, a magnetic field is created in the coil. As a result of a great number of magnetising cycles per second, current is generated in the work located in the magnetic field; the current heats the work up very quickly, the depth of heat penetration being a function of the current frequency.

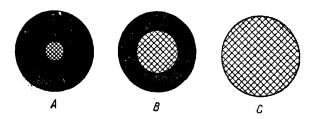


Fig. 10. Diagram illustrating distribution of currents in a conductor at various frequencies:
a-at 50 cps; b-at 2,500 cps; c-at 250,000 cps

Heat-treatment induction heating makes use of audio- (from 1,000 to 10,000 cps) and radio-frequencies (from 100,000 to 1,000,000 cps). The depth of heat penetration diminishes, as current frequency increases. Fig. 10 presents a diagram showing the effect of current frequency on heat penetration.

In practice, radio-frequency current, supplied by radio-tube oscillators, is employed to heat articles up to 2-3 mm in depth, while a deeper heating, including heat soaking, requires audio-frequency current,

supplied by motor generators.

The best material for inductors is red copper. Brass should be used only when red copper is not available. Usually, the inductors are made of square or rectangular red-copper tubes which carry water for cooling.

Table 48 gives data on tube profiles. Inductors for fancy-shaped work are made sectional, cast or forged. The thickness of the sides of a water-cooled inductor ranges from 0.5 to 2 mm, while that of an uncooled one reaches 7-10 mm.

To prevent the inductor from coming into contact with the work, if is wound with asbestos cord impregnated with soluble glass.

Initial	liameter, mm	* Cross	Initial d	iameter, mm	Cross
External	Internal	section, mm	External	Internal	section, mm
4 5 6 7 8 9 9 10 10 11 11 12 12	3 4 4.5 5 6.5 7 7 7 8.5 or 7 8.5 or 7 9 or 8 9 or 8 9 or 8 10 or 9 10 or 9	2.5×5.5 3.5×4 4.5×5 4×7 5×7.5 4×10 5×9 7×7 5×10.5 7.5×8 5×12 6×11 8.5×8.5 7×12 8×11 9×9	13 13 14 14 14 15 15 16 16 17 18 19 19 20 20	11 or 10 11 or 10 12 or 11 12 or 11 13 or 12 13 or 12 14 or 13 14 or 13 16 or 14 16 17 or 16 17 or 16	$\begin{array}{c} 8\times12.5\\ 10\times10.5\\ 8\times14\\ 10\times12\\ 11\times11\\ 8.5\times15\\ 10.5\times13\\ 10\times5\\ 12.5\times12.5\\ 10\times16.5\\ 12\times16.5\\ 12\times16.5\\ 10\times20\\ 15\times15\\ 11.5\times20\\ 15\times16.5\\ \end{array}$

The inductors should be brazed with hard solder.
The following methods of induction hardening are in general use:

- 1. Simultaneous hardening of the whole area to be treated.
- 2. Step hardening of separate sections of the work. Each section is heated separately. When treating pieces with large surface areas, the heating is carried out by a continuous-successive method.
- 3. Continuous-successive method of hardening (Fig. 11).
- 1. Used for hardening disk-shaped pieces.
- 2. Used for hardening gear teeth, axle journals, etc.
- 3. Used for hardening parts of great length. Cylinder-shaped parts are to be rotated in order to obtain a uniform hardened layer; speed of rotation is from 50 to 200 r.p.m.

To prevent uneven heating, particular care should be taken to centre the work properly in the inductor. The clearance between the work and the inductor should remain within 2-6 mm. Inadequate spacing may cause contact between the work and the inductor, or the puncture of the air gap between them, and overheat the external layer, particularly when vacuum-tube oscillators are used. Larger clearance is required for brazing or for deeper hardening. The spacing of turns in a

multiturn inductor should be minimum to ensure uniform heating. Practically this gap should average 2 mm.

Induction heating considerably raises critical points of steel and

widens hardening temperature ranges.

Fig. 12 presents temperature diagrams for induction heating of some grades of steel in initial annealed state.

To avoid overheating and even melting of the sharp edges of the work, it is recommended to fill the grooves with brass or copper inserts

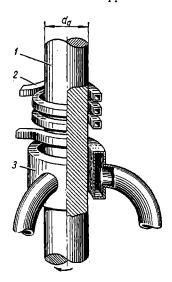


Fig. 11. Relative positions of inductor and sprayer when hardening a shaft: 1-shaft; 2-inductor; 3-sprayer

or to use an adequately shaped inductor in which the sharp edge of the work would remain at considerably greater distance from the sides of the inductor than the rest of the work.

Parts from carbon steels, heated by means of high-frequency current, are cooled in water, those from medium-alloy structural steels—in aqueous solutions, while those from high-alloy steels—in oil. The latter is practicable only in case of simultaneous hardening of the whole work.

When the continuous-successive method of hardening is employed, cooling is effected by a sprayer in which the orifices, 1-2 mm in diameter, are spaced at an angle of 20-30° to the axis of the work and oriented away from the inductor. This prevents early cooling of the heated work.

When the simultaneous method of hardening is used and the work is cooled by water or aqueous solutions, spraying is ensured by the inductor itself (Fig. 13), the internal section of which is provided with a number of orifices. After the work has been heated, current is switched off and the inductor tube is supplied with water or aqueous solution under pressure, which quenches the work.

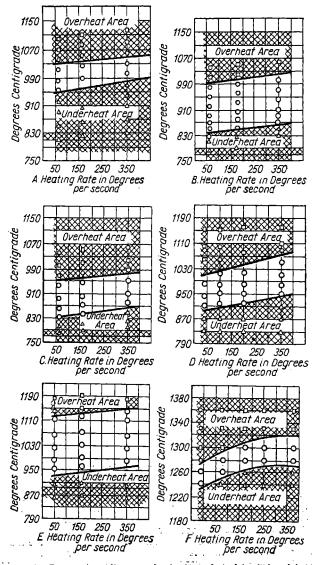


Fig. 12. Temperature diagrams for heating of steel by induced heat:
a-for steel Cr. 50; b-for steel V8; c-for steel V12; d-for steel 9XC; efor steel 30ΧΓCA; i-for steel P9

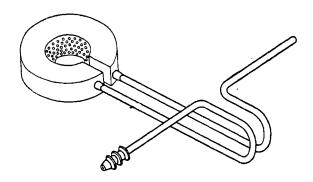


Fig. 13. Inductor for simultaneous hardening of articles

6. TEMPERING

Tempering is employed to lessen or eliminate entirely internal stresses, to soften the hardened steel and to increase its ductility.

Tempering consists in heating the hardened steel to below Ac₁, in maintaining the work at the said temperature and in subsequent cooling. For tempering, the work is heated in oil, saltpetre or alkali baths, as well as in furnaces with air atmosphere. The total holding time for work in such a furnace should not be less than 30-40 minutes and should average 2-3 min per each millimetre of the least section.

When the work is charged in bulk in large numbers, the holding time should be increased 1.5-2 times. After tempering, the work is usually air-cooled.

To avoid temper brittleness in some grades of steels, the latter are cooled in oil in the 450-650°C range; these steels include chromium, chrome-nickel, chrome-silicon, chrome-manganese, chrome-silicon-manganese, chrome-nickel-vanadium, chrome-aluminium steels.

Colour tempering or tempering with surface hues as temperature indicators is carried out in the 220-330°C range (Tables 49 and 50), with subsequent dipping in oil or water.

Colour tempering of selected areas of the work is done chiefly in lead baths which ensure rapid heating necessary to prevent tempering of the rest of the work.

The above-mentioned tempering process may be effected much quicker in an inductor. The rate of heating and the necessary tempering colour are dependent upon the distance between the work and the inductor. To facilitate and stabilise the operation, the inductor is encased in an asbestos sheath (Fig. 14) which serves as a mounting for the work to be tempered. The thickness of the sheath permits adjusting the duration and degree of tempering.

Tempering Colours

Tempering colours	Temperature, °C	Approximate thickness of oxidised layer, microns
Pale yellow	220	0.45
Straw	240	0.50
Purple brown		0.65
Blue	300 315	0.72
Grey	330-350	

Table 50

Tool diameter or thickness, mm	Furnace type	Holding time at the tempering temperature, hrs
Up to 20	Oil or saltpetre baths,	1.0
21-40	electric shaft furnace,	1.5
41-60	type ПН-31, ПН-32 or	2.0
Above 60	ПН-34 furnaces	2.5

Approximate Holding Time at Tempering Temperatures

At present, tempering, in some plants, is carried out in induced heat installations at the expense of incomplete cooling following hardening; this is the so-called "self-tempering". The work heated for hardening in an induced heat installation is cooled for a definite length of time, determined by calculation or, in most cases, experimentally. At a given moment cooling is automatically cut out and the work heats itself up to the tempering temperature at the expense of accumulated heat. This method is applicable in plants with mass or large-scale production, provided they are equipped with up-to-date automatic machines.

To avoid warping and cracking, the work, after grinding, is subjected to low-temperature tempering, also called ageing, at 120-160°C.

The heat-treatment process consisting in hardening the work with subsequent high-temperature tempering at 500°C and above is called improving*.

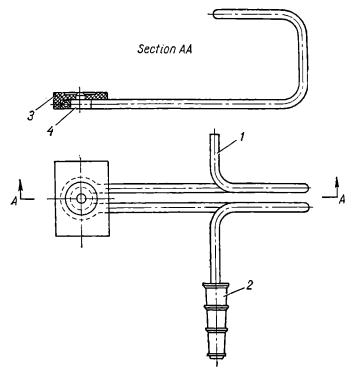


Fig. 14. Inductor for tempering of parts:
 1 - inductor; 2 - nipple; 3 - asbestos sheath; 4 - asbestos plug for fastening and centering the sheath

The improving is used to reduce the coarseness of structure as well as to attain the best combination of strength and toughness in structural steels, and chiefly in the alloy steels.

Table 51 lists defects which may be met with in heat-treatment processes.

^{*} The word improving (a literal translation from the Russian) is used to denote a special heat-treatment procedure aimed at bettering the properties of steel, -Tr.

Defects in Heat-treatment of Steel

Chief measures to prevent and correct the defect	
Chief cause	
Detection	
Defect	

1. Annealing and Normalising Defects of Structural Steels

Repeat annealing or normalising at normal temperatures	Repeat annealing or normalising at normal temperature. In case of considerable overheat	ing use double annearing, the first one at a temperature by 50-150°C above, the normal	ž	Repeat annealing at the required cooling rate
Underheat. Annealing Repeat annealing or temperature below that normalising at normal required or insufficient temperatures	0		Overheat. Heating of steel in an oxidising atmosphere at high temperatures, in the proximity of the melting	Excessive cooling rate
Tensile test	Coarse-grained fracture. Fracture inspection. Ten- Low ductility and, sile and impact tests particularly, impact strength		Very coarse and bright Fracture inspection. Mi- fracture crostructural analysis	Hardness test
Low ductilify, the frac- Tensile test ture being relatively fine-grained	Coarse-grained fracture. Low ductility and, particularly, impact strength		Very coarse and bright fracture	High hardness

	Repeat annealing at a requisite temperature and follow strictly the prescribed cooling schedule, or temper at high temperature	Normalising or hardening with subsequent tempering at 670-700°C, holding time not less than 2 hrs		Irreparable defect. Preventive measures: 1. Whenever possible use step-hardening, as well as intermittent hardening in two quenching media 2. Avoid manufacturing parts with acute angles and sharp changes in section; when this is unavoidable, use only alloy steel 3. No water should be present in the oil quenching tank
Tool and High-speed Steels	Underheat Excessive cooling rate in normal annealing or insufficient holding in isothermal annealing	Heating above Ac	3. Defects in the Hardening of Steel	Visual inspection and Stresses caused by changes crack detector test; kerin volume arising from sosne test and black-ing-in ing-in temperatures below as intermittent hardening, as well temperatures below as intermittent hardening as intermittent hardening in two quenching media are should be present in the oil quenching tank
2. Defects in Annealing of Tool and High-speed Steels	Hardness test	Microsection inspection under a microscope	3. Defects in the F	Visual inspection and crack detector test; kerosene test and blacking-in
•	High hardness	Carbide lattice		Cracking

Table 51 continued

2			
Defect	Detection	Chief cause	Chief measures to prevent and correct the defect
Cracking	Visual inspection and crack detector test: kerosene test and blacking-in	Stresses caused by changes in volume arising from transformation of austenite to martensite at temperatures below 1250° C	4. Heat slowly the hard- ened work when hard- ening procedure is re- peated 5. Grooves, holes near edges, sharp transitions in section should be
Insufficient hardness	Hardness test	Low hardening tempera- ture, insufficient hold- ing or low cooling rate	insulated with asbestos The work should be nor- malised or annealed and hardened again accord- ing to normal schedule
Increased brittleness. Coarse-grained fracture	Visual inspection of fracture. Impact test	Excessive hardening temperature or too prolonged holding	To be corrected by the same method as is used for insufficient hardness defect
Soit spots	Hardness tests at various points	Ineffective cooling, local decarburisation, scaling, non-uniform grain size, contamination of steel by abnormally high amounts of slag, mutual contact of parts on cooling	ັດ
	_		

If the depth of penetration is greater than the grinding allowance, the defect is irreparable. To prevent the defect, the part should be heated in a controlled atmosphere, and, if the latter is not available, in cases containing castrion turnings (50% fresh, 50% used), charcoal with 5% soda, etc. To avoid decarburisation, cushed ferrosilicon (1.0-1.5% of the weight of the salt), borax, boric acid, potassium ferrocyanide (25-30grams), are added to the salt), are to the salt, and the salt, and the salt, are added to the salt, are the salt, are added to the salt, are added to the salt, are the salt, a	baths two-three times per shift or the bath is deoxidised with charcoal. The defect is mostly irreparable. To prevent it, hardening temperature and cooling rates should be lowered and finegrained and alloy steel used.
Combination of furnace atmosphere oxygen with the iron of the work (oxidation) or with its carbon (decarburisation)	Structural transformation in the 650-500° C range and below 300° C, caus- ing distortion
Oxidation is detected by inspection, while decarburisation by hardness tests and microstructure analysis	Checking for size
Oxidation and decarbu- risation	Deformation (change in dimensions)
o pu	(ch <i>z</i>
ion a	eformation dimensions)
Oxidation	Deform dime
4*	9

Table 51 continued

Defect	Detection	Chief cause	Chief measures to prevent and correct the defect
Warping (change in shape)	Checking for beat (in centres) or by thickness gauge on a face plate	Incorrect immersion of work in quenching media, internal stresses prior to heating, etc.	Reparable defect which can be corrected through: 1) cold or hot straightening; 2) grinding if the warping does not exceed the allowance. Preventive measures
Point or groove-like pit- ting of the surface	Visual inspection	Uneven scaling. High content of sulphuric salts and chemical	in the martensite range, proper dipping in quenching media, uniform heating, and checking for warpage prior to hardening Prevention: 1) scrupulous calks used for heating of the proper seals used for heating in the properties of
		action of chloride safts in saft baths. Contact with molten lead spilled on the furnace hearth	2) deoxidation of salt baths; 3) covering of lead bath surface with charcoal; 4) elimination of oxidising atmosphere in the furnace; 5) strict control of hearth contamination

Correction: 1) heat to 1140-1160° C, hold for 3-8 min, cool to 800-720° C, hold for 15-30 min, cool in air; 2) repeat the above operation; 3) conduct normal amealing; 4) apply hardening and tempering (this procedure is in need of further investigation)		Deoxidise alkali bath additionally with potassium ferrocyanide Deoxidise additionally the salt bath	Bring the bath to proper consistency by adding the required salt
Annealing at high temper- ature. Repeated hard- ening omitting inter- mediate annealing	4. Bright Hardening Defects*	Alkali bath is insuffi- ciently deoxidised Salt bath is deoxidised insufficiently	Salt bath is too fluid
Visual inspection	4. Bright Hard	Darkening or oxide tinting 1. Heat thoroughly a bright work in an alkali bath and dip in water 2. Heat work in a salt bath 150-200°C below bath 150-200°C below in salt bath 150-200°C below in salt bath 150-200°C below in salt bath	ture national tempera- ture and cool in alkali 3. Check by chemical analysis to make sure the salt composition meets the appropriate technological require- ments
Coarse-grained flaky fracture characteristic of high-speed steel articles over 12 mm in diameter		Darkening or oxide tinting of the work	

* Column "Defection" lists methods of detecting causes of the blackening of the work in molten alkali in the bright hardening processes.

Table 51 continued

Defect	Detection	Chief cause	Chief measures to prevent and correct the deject
Sooty deposits on the work	Sooty deposits on the yeis for absence of salt- work Sooty deposits on the alkali bath bath petre in the alkali bath pet	Presence of saltpetre in the alkali bath	Make up a new alkali bath and prevent salt- petre from getting into
Required hardness is not attained	Visual inspection of surface	Visual inspection of sur- Addition to the alkali Heat the alkali at the face hath of an excessive operating or somewhat amount of potassium higher temperature for ferrovaride used as a few house	Heat the alkali at the operating or somewhat higher temperature for
	Hardness testing with machines	devices and devices as a device as a devic	Adequate mixing of the bath in operation
		2. Lack of water in low- Check the bath for water temperature alkali baths by thermal analysis and adjust its amount to that required	Check the bath for water by thermal analysis and adjust its amount to that required

5. Defects in Steel Tempering

Low temperature or insuf- Repeat tempering accord-ficient holding time ing to normal schedule
Low temperature or insuf- ficient holding time
Hardness test
Under-fempering, In- Hardness test creased hardness and low ductility. A drop in hardness for high-speed steel

	₹	nig to normal schedule Prevention: 1) cool in water or oil after tem- pering at 450-600°C; 2) use steel containing molybdenum, titanium and niobium	Correction: repeated tempering followed by oil cooling, the temperature being by 20-30°C above that of the initial tempering	
	Tempering at temperature above that required	Carbides, oxides, nitrides and phosphides separate out at the grain boun- daries		
_	Hardness and tensile tests	Izod impact test		
_	Over-tempering. Low hardness and low tensile strength and elastic.	Temper brittleness, Low resilience after temper spring at 450-600°C and slow cooling		

Chapter V

CASE-HARDENING

The combined chemical and heating treatments called case-hardening consist in the heating of the work up to a given temperature in a specially chosen chemically active medium, e. g., in coal with additions of salts, in a medium producing atomic nitrogen, in chemically active molten salts, etc. As a result of the interaction with the surrounding medium, the chemical composition of the outer layer of the work is altered.

The work subjected to case-hardening acquires surface hardness, high resistance to corrosion, heat resistance, wear resistance or the ability to harden, depending upon the nature of the treatment. Pertinent details on the different case-hardening processes can be summarised as in the following paragraphs.

1. CARBURISING

Carburising consists in the saturation of the surface layers of the steel with carbon. The carburised work, following hardening, acquires an extra-high surface hardness while retaining a tough core.

Carburising is used for work from carbon and alloy steels with up to 0.25% carbon. Steels with up to 0.3-0.4% C may be used for large simple-shaped parts as well as for parts requiring high core strength.

In shop practice carburising is being used for tool steels (chiefly alloy chromium steels) hardened in oil. Cold stamping dies of regular shape, extruding dies, etc., are subjected to carburising which raises considerably the wear-resistance of the above tools.

Carburised tools are not to be ground, as in this case the most valuable wear-resistant layer will be removed. The carburising procedure follows the usual pattern yielding a 1-mm case. Hardening should be carried out carefully, with all precautions taken against warping and decarburisation. Hardening temperature averages 800-820°C, while tempering is accomplished at lower temperatures.

Steels are carburised in solid, gas or liquid media, called carburis-

ers.

Carburising by solid carburising compounds (pack carburising). Table 52 lists the chemical composition of carburisers used in shop practice. Charcoal, chiefly oak or birch by its nature, is crushed to pieces 3 to 10 mm in size and impregnated with a solution of carbonate

salts, after which it is dried at 100-150°C. Sometimes plain mixing of charcoal with powdered carbonate salts proves to be satisfactory.

Table 52
Composition of Carburisers for Pack Carburising
(% by weight)

_							
No.	Charcoal	Barium carbonate	Soda ash	Cal- cuim carbon- ate	Coke	Turf coke	Fuel oil
1 2 3 4 5 6 7 8 9*	74-78 65 87 85-90 90 — 60 45 98	12-15 10 — 10 — 40 12 2	1.0-1.5 10 10-15 10-15 	3.5	20 43 	85-90 ————————————————————————————————————	4.5-5.0 3 — — — — —

^{*} This carburiser is used for boron-bearing steel.

The moisture content of the carburiser should not exceed 5.7%. A carburising mixture consists of 20-30% and 70-80% of fresh and used carburiser, respectively. Parts subject to carburising should be dry and free from scale, rust, dirt, grease, etc. The parts are packed into boxes or containers made of heat-resisting steel, and, when the necessary steel is not available, boxes manufactured from plain structural

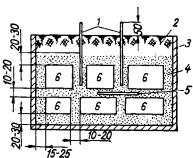


Fig. 15. Arrangement of parts in containers: 1 - outside "witnesses": 2 - luting; 3 - container; 4 - carburiser; 5 - internal "wifness"; 6 - parts

steel are used; the wall thickness of the box or container should be equal to 4-6 mm. The shape of the box should follow, as much as possible, that of the carburised work. The packing of the work should conform to the dimensions indicated in Fig. 15. The width of the box should not exceed 250 mm in order to facilitate its handling and to hasten heating.

The boxes are luted with a mixture consisting of two parts of refractory clay and one part of river sand, diluted to paste-like consistency.

The boxes are charged into a furnace heated to the required temperature. The boxes should be heated through at 780-800°C, after which the temperature is raised to 900-950°C, this being taken as the beginning of the carburising process.

Table 53

Relation of Holding Time to Layer Thickness in Carburising by Solid Carburisers at 900-950° C

		Thickness of carburised layer, mm							
The least cross- section of the	0.25	0.5	0.7	0.9	1.1	1.2	1.3	1.4	
box, mm	Total holding time in furnace, hours								
100 150 200 250	3.0 3.5 4.5 5.5	4.0 4.5 5.5 6.5	5.0 5.5 6.5 7.5	6.0 6.5 7.5 8.5	7.0 7.5 8.5 9.5	7.5 8.5 9.5 10.5	8.0 9.5 10.5 11.5	8.5 10.5 11.5 12.5	

Too rapid a heating to $900-950^{\circ}C$ of cold boxes causes a difference in carbon saturation of parts situated near the walls and in the centre of the box.

Witness samples for determining the depth of the case are 6 to 10 mm in diameter. These samples are manufactured from the same steel as the carburised parts, or from grade Cr. 15-20 steel.

After the supposed ending of the carburising process, the outer witness sample is extracted from the container, hardened and the fracture is etched for I minute in a compound composed of 100 cm³ of methylated alcohol, I cm³ of hydrochloric acid and 2 g of copper chloride. Copper separates out at the non-carburised core. The thickness of the case can also be found by tempering of the fractured hardened witness until tempering colours are observed; in this instance the case and the core are coloured in different tints.

After the case-hardening process, the containers are cooled on the shop floor.

Paste carburising. Table 54 presents the composition of pastes used for carburising. The prepared powder of one of the pastes indicated is diluted in a 15% aqueous solution of molasses or glue, to a consistency which permits colouring.

The surface to be carburised or the whole work is coated by a layer 3-4 mm thick, the coating being accomplished by immersion or by the use of a brush. This done, the parts are dried and packed into boxes provided with sand seal (ref. to Fig. 22). The paste carburising process is carried out at 920-930°C, the approximate speed of case-hardening being 1 mm per hour.

When the carburising is completed, the parts are air-cooled or quenched immediately after they are taken out of the container.

Gas carburising. In gas carburising, the parts are heated up in a gas atmosphere containing carbon.

The following gases are used for carburising: 1) natural gases, 2) ar-

tificial gases (producer gas, pyrolysis gas, etc.).

An artificial gas produced by dissociation (pyrolysis) of oil products has gained widest application.

Table 54 Chemical Composition of Pastes Used for Carburising

,			Pas	ste num	ber			
Paste component	Stalite		3	4	5	6	7	
Paste component	ī	2		7				
	Composition in % by weight							
Ivory black or coke Barium carbonate Sodium or potassium carbonale Cyanide compound "FUNTX" Sodium or potassium oxalate Nickel formiate or cobalt oxalate Ferrochrome (for carbon steel). Sand	30-60 20-40 5-10 — — 5-10	30-60 20-40 5-10 5-10 5-10 —	30-60 30-60 — 5-10 — 15 —	35 15 20 — — — 15 —	45 20 20 — — — — — 15	40 15 20 — — — 5 20	50 40 	

Note. The composition of the cyanide compound "ГИПХ" is given in the footnoie to Table 64.

Carburising is effected chiefly in shaft furnaces of the "II" series. When the carburising is carried out in shaft furnaces, the pyrolysis of oil products occurs directly in the furnace retort, as the carburi-ser is fed by drops. Kerosene, benzene, pyrobenzene, spindle oil, and synthol filtered from solids are used as carburisers.

Table 55

Amounts of Black, Coke and Gas Obtained in the Pyrolysis of Some
Carburisers

Carburiser	Amount of gas (litres) from 1 cm ³ of carburiser	Amount of black and coke (grams) per 1 cm ³ of carburiser
Benzene	0.73	0.60 0.54 0.39 0.28

A new carburiser for gas carburising, the so-called synthol, which produces less black and coke and increases the speed of the process, has currently found wide application in shop practice. Table 55 presents data on the amounts of black, coke and gas generated in the pyrolysis of some carburisers. Maximum effectiveness has been obtained with the use of synthol in continuous furnaces. Synthol is currently used in a great number of plants.

Table 56

Recommended Kerosene Feeding Rate for Carburising in Shaft
Furnaces of Various Sizes

			Furna	ice type		
	Ц25	Ц35	Ц60	Ц75	Ц90	Ц105
			Drops p	er minut	e	
During the heating with temperature reaching 800° C During the heating with temperature reaching 900° C	20-40 60-70	40-50 70-80	70-80 110-130		120-140 200-220	l

Table 56 presents the kerosene feeding rates recommended for shaft furnaces of various sizes during the carburising process. These data are compiled from published literature, and practical results obtained in a number of plants. The author considers it expedient to cut off the carburiser feed one hour beforethetermination of the gas carburising process in order to give the carbon enough time to diffuse into the inner areas of the work and to avoid oversaturation of the outer layers. This is of utmost importance when a thicker case is required.

As the carbon absorption rate on the work surface decreases with holding time, some investigators believe that kerosene should be fed at maximum rate at the beginning of the carburising process and that its supply should then be gradually diminished (Table 57).

Table 57

Kerosene Feed Rates Recommended for Carburising in Shaft Furnaces
(Data by Chirikov V. T.)

Furnace type	Kerosene i	fed during the al period	Kerosene fed (drops per min) during the
	Drops per min	Feed duration, hrs	remaining period
Ц25, Ц35 Ц60, Ц75 Ц90, Ц105	70-80 100-120 200-300	2-3 2-3 2-3	20 30-40 50-70

Note. Furnaces may be fed either with artificial or natural gases, instead of the iquid fuel.

The following sequence of technological operations for gas carburising process in shaft furnaces is recommended:

1. Charge the work into the furnace heated to 850-950°C.

- 2. Witness samples for process control should be charged at the same time as the work.
 - 3. Press the cover tightly on the retort.
 - 4. Feed the kerosene according to Table 56.
- 5. Set fire to the waste gases; in a normal process the flame should smoke slightly.
- 6. The termination of the process is determined with the aid of witness samples; to this end, the process is interrupted, the samples extracted and hardened (Table 58).

At present the gas carburising process has been automated by controlling the carburising medium for its dew point.

Table 58

Approximate Holding Time for Gas Carburising at 900-950° C in "U"-Type Furnaces

Ti	me, hrs		2-3	4-5	6-7	8-9	9-10
Layer	depth,	mm .	0.3-0.5	0.6-0.7	0.8-1.0	1.1-1.2	1.2-1.4

Note. Time is counted off from the moment the temperature of 900°C has been attained.

The composition of the waste gases should be within the following limits: carbon dioxide—0.5%, oxygen—0.5%, unsaturated hydrocarbons—10%. Surplus pressure varies from 8 to 25 mm water gauge.

Liquid carburising. This process is practised in a number of plants, in baths, where the carburising component is carborundum (silicon carbide). The usual composition of the bath is: soda ash (Na_2CO_3) —80%, common salt (NaCl)—10-12%, carborundum (SiC)—8-10%. This compound is currently supplemented with ammonium chloride (NH_4Cl) , which speeds up the carburising process, and produces active nitrogen, which in conjunction with carbon saturates the surface of the work.

A liquid carburising process such as it is practiced at the Kharkov

Sewing Machine Plant is described below.

The composition of the bath is: soda ash 70-74%, common salt 9-12%, ammonium chloride 8-9%, carborundum 9-10%. The carborundum used is black, coarse-grained, the size of grains being 24-50 mesh (mesh is the number of openings per linear inch).

Soda and salt are melted first, then ammonium chloride is added. As soon as the bath heats up to 850-870°C, carborundum is charged.

Ammonium chloride and carborundum, to avoid vaporisation and rapid burning out, are charged into the bath in paper wire-corded packets, sunk to the bath bottom with a poker, perforated bell or some other object and held there till melting, after which the bath is brought to the carburising temperature through continuous mixing. The process is sometimes accompanied by such intensive foaming that the bath temperature is to be temporarily lowered.

To protect the bath against oxidation, a layer of slag 10-15 mm thick is left over, while the remainder is skimmed. The amount of salt should not exceed $^2/_s$ of the bath height, the remaining volume being

intended to counteract foaming.

The work subject to carburising should be dry and clean. The temperature of the process averages 850-900°C. Holding time is a function of the case thickness (Table 59). Hardening is carried out either in oil or water depending upon the grade of the steel.

Table 59

Approximate Time of Carburising in a Liquid Carburiser as a Function of Case Thickness

Case thickness, mm	Duration of the process, hrs	Case thickness. mm	Duration of the process, hrs
0.4	1	1.2	4
0.6	1.5	1.4	5
0.8	2	1.6	6.5
1.0	3	1.8	8

Note. The bath should be refreshed every 5-8 hrs of operation by adding carborundum (2-5% of the total salt weight).

A complete replenishment of salt is carried out every 3-4 weeks of continuous bath operation, as the bath thickens and loses carburising ability.

The bath is controlled by observing the following practical symptoms: 1) considerable flaming and foaming (runs against the normal functioning of the bath); 2) the work should possess a uniform silvery colour (dark spots are signs of bath exhaustion).

Carborundum consumption averages 10-12 kg per 1 ton of work. High-temperature carburising. High-temperature carburising, at 1050°C and above, is currently finding increasing application. Pack carburising at the above temperatures is effected in furnaces with silit (silicon carbide) resistors. After carburising, normalising and hardening or double hardening are carried out.

High-temperature gas carburising is a current practice at the Likhachev Plant, where induction heating in a special installation is

being applied.

Protection of work surface not required to be carburised.

1. Positive allowance which is machined off after carburising.

2. Luting (Table 60).

3. Copper-plating (Table 61).

To protect holes against carburising, a mixture of quartz sand and scale (from the hardening tanks) is used, or the apertures are plugged with metallic plugs.

Heat-treatment schedules for carburised work are presented in Ta-

ble 62.

Lutes Used for Protection against Carburising

Lute composition		
Component	%	Method of manufacture
Copper monochloride Minium	7 0 30	The powders are mixed thoroughly and diluted in alcohol rosin varnish (250 g of varnish per litre of ethyl alcohol). The lute is spread over the surface of the part in a uniform layer with the aid of a brush. Layer thickness is equal to 0.5-1 mm
Silica	3	The paste consists of both the liquid (20-25% by volume) and solid (75-80% by volume) phases. The liquid phase comprises 80% soluble glass and 20% water. Prior to use, both phases are mixed and then spread over the part with a brush. After drying (after 30-40 min) the part is given a second coat

Table 60

Lute composition		Walland of manufacture
Component	%	Method of manufacture
Sand	40 44 10 3 3	The compound is diluted in soluble glass and applied uniformly over the work. The clay should be well ground
Talcum	50 25 25	The clay is well ground. The compound is diluted in soluble glass to paste-like consistency
Fireclay	90-95 5-10	The compound is diluted in water to paste-like consistency

Table 61
Relation of Copper Coating Thickness to Depth of Carburising

Required carburising depth, mm	Below 0.8	0.8-1.2	Above 1.2
Thickness of copper coating, mm	0.02	0.03-0.04	0.05-0.07

Table 62 Heat-treatment Schedules for Carburised Articles

Heat-treatment schedule	Application	Notes
Hardening from 860-900° C Hardening from 760-800° C Tempering at 160-180° C	For natural coarse- grained steel in- clined to overheating Whenever extra-high quality is required for carburised parts	
Hardening from 800-850° C Tempering at 160-180° C	For natural fine- grained steel For non-critical parts from natural coarse- grained carbon steel	

Heat-treatment schedule	Application	Notes
1. a) Tempering at 640-650°C b) Hardening from 850-860°C c) Tempering at 160-180°C d) Sub-zero treatment e) Tempering at 160-180°C	For high-alloy steel (grades 18XHBA, 12X2H4, 12XH3, etc.)	tempering and
2. a) Rapid partial cooling in a molten salt or oil from carburising temperature to 250-550°C b) Transfer to a furnace heated to 550°C, and a 4-8 hr holding c) Air cooling d) Tempering at 630-680°C for 8-10 hrs with cooling in air or furnace e) Hardening		The heat-treatment method was proposed by V.T. Chirikov to ensure optimum structure both of the case and the core
f) Low-temperature tempering. Hardening from carburising temperature with air-blast cooling to 740-840° C. Tempering at 160-180° C. Sub-zero treatment. Tempering at 160-180° C		

2. NITRIDING

Nitriding (nitrogen-hardening) consists in the saturation of surface layer of steel or cast iron with nitrogen. There exist two varieties of nitriding:

1. Strength-nitriding aimed at increasing hardness, wear resistance and fatigue endurance (Table 63).

2. Anticorrosive nitriding aimed at increasing the resistance to cor-

rosion in water (fresh) and in humid atmosphere (Fig. 16).

Strength-nitriding. Nitriding is chiefly applied to special grades of steel (35 X IOA and 38 X M IOA), containing aluminium, with a view to raise their hardness. In addition to these, alloy tool and stainless steels are also nitrided. Maximum hardness, after nitriding, is displayed by articles made from grade 35 X IOA and 38 X M IOA steels (1000-1150 H_V).

Plain alloy structural steels—chrome-nickel, chrome-nickel-molybdenum, etc.—are nitrided to increase fatigue endurance. Surface hard-

ness of these steels, after nitriding, is 600-700 H_{ν} .

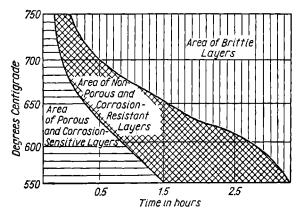


Fig. 16. Diagram for computing optimum schedules of anticorrosion nitriding

Cast irons, subject to nitriding, should contain some aluminium and chromium. Hardness of cast irons following nitriding is $1000 \ H_{V}$.

Prior to nitriding, steel parts are hardened and tempered at 550-650°C, while those from cast iron are hardened and tempered at 600-650°C.

Protection against nitriding. Areas not to be nitrided are to be insulated by one of the following methods: (1) tinning; (2) nickel plating; (3) luting with compounds composed of three parts of powdered tin, one part of powdered lead and one part of powdered chromium oxide. A well ground mixture is diluted in a solution of zinc chloride. After luting, the part should be dried.

Strength-nitriding checking. 1) Hardness is checked by instruments with diamond tips, at low loads (Table 8). 2) Brittleness is characterised by the distortion of the diamond indentation (shearing of the indentation edges). 3) The layer depth is determined by etching a ground

Strength-nitriding Schedules

				Silving	STATE THE THE STATE OF THE STAT	S Contract	3				
			lst step			2nd step		. 8	3rd step		
Nitriding schedule	Version	Тетрега- Сиге, °C	Dissociation degree, %	гіте, һтз	Tempera - O°, 49111.1	Dissociation degree, %	erd , smiT	-sraqmeT D°, suri	Dissociation degree, %	end ,emiT	Thickness of nitrided layer, mm
One-step	- 2	480-520	20-25	20-25 Up to 90 30-50 36-65	l	1	ı	ı	1	I	0.5-0.7
Two-step	- 6	500-510	Below 25 30	18-20	550-570	35-55	20-24	I	1	1	0.5-0.7
Three-step	1	500-520	20-35	12-15	550-570	40-60	12-15	500-520	40-60	12-15	0.5-0.8

butt of a sample with a 4% solution of aqueous nitric acid, as well as by the nature of the fracture and the microsection.

Anticorrosion nitriding. Articles from carbon and alloy steels and cast iron may be subjected to anticorrosion nitriding. Prior to nitrid-

ing, the parts should be thoroughly degreased.

Nitriding of work from high-carbon steels requiring high hardness should be followed by hardening at adequate, for the given steel, temperature.

The degree of dissociation should be maintained within the range of

1 ne de 35-70% .

Anticorrosion nitriding checking is effected through dipping or wetting the part with 10% aqueous solution of blue vitriol (copper sulfate) for 1-2 min. Copper is deposited at pores, fractures, and other defective points.

Uniform heating throughout the muffle furnace is essential for successful nitriding. The volume of the retort should be filled with parts

to maximum capacity.

3. CYANIDING

Cyaniding consists in the simultaneous saturation of the surface of the steel both with carbon and nitrogen. Cyaniding is carried out in liquid, gas or solid media.

Two types of cyaniding are distinguished: (a) high-temperature cyaniding aiming to increase hardness, wear resistance, and fatigue endurance of parts from structural steels; (b) low-temperature cyaniding intended to increase hardness and red hardness of tools from high-speed steel.

1. High-temperature liquid cyaniding. To prepare the bath for operation, first neutral salts (soda ash, etc.) are melted, then cyanides are

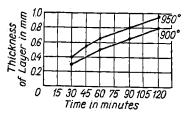


Fig. 17. Effect of duration of cyaniding at 900° and 950°C on the depth of cyanided layer

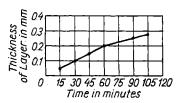


Fig. 18. Effect of duration of cyaniding of steel 20X at 850°C on the depth of cyanided layer

added. In baths containing compounds 6 and 7 (Table 64) excessive foaming occurs when cyanide compound is charged; the excess foam is removed, a thin layer being left to protect the bath from exhaustion. The melt surface of a deep cyaniding bath (compounds 4 and 5) is covered with a layer of graphite, 3-4 mm thick, for the same reasons.

The working temperature averages 830-860°C for common cyaniding and 900-950°C for deep cyaniding (Figs. 17 and 18).

The refreshing of baths with sodium cyanide is effected by adding high-percentage sodium cyanide at a rate of 0.5-1% of the weight of the salts in the bath per hour; for baths containing the " Γ И Π Х" cyanide compound, the additions should average 2-4% of the weight of the

salts in the bath every 2 hours of operation.

The author practises high-temperature cyaniding, in conjunction with bright hardening, in a bath containing 38-40% sodium chloride, 38-40% potassium chloride and 20-24% potassium ferrocyanide. The latter should be well dried and added by small portions, while mixing the bath. On charging, foam and slag are formed, and they should be removed.

Table 64
Composition of Baths for High-temperature Cyaniding

	Composition of baths, %									
Compound No.	Sodium cyanide	"ГИПХ" cyanide compound	Soda ash	Common salt	Calcium chloride	Barium chloride				
1 2 3 4 5 6 7	25 40 45 50 6 —	 9 9	15-20 30 35-40 — — —	55-60 30 15-20 15 14 26 37		35 80				

Note. The "ΓΜΠΧ" cyanide compound consists of: 43-49% Ca(CN)₃; 2-3% CaCN₂; 30-35% NaCl; 14-16% CaO: 4-5% C.

After requisite holding in the bath, the work is cooled in a 50% aqueous solution of caustic soda. Hardened parts are clean and bright. The depth of cyaniding equals 0.15 mm, the holding time for the abovementioned compound averaging 15 min.

2. High-temperature gas cyaniding. Gas cyaniding is carried out in a mixture of ammonium and carbonaceous gases used for gas carburising. The cyaniding mixture comprises: ammonium 20-30%, carburis-

ing gas 70-80%.

Benzene, pyrobenzene, kerosene and synthol are used as materials for producing carburising gas. They are fed dropwise direct into the retorts

of shaft furnaces.

The working temperature of gas cyaniding should average 840-860°C. For parts of simple shapes, the temperature may be raised to 900°C.

Approximate depth of cyaniding after 1 hour of operation at 850°C

is 0.12-0 16 mm (Fig. 19).

The hardening of the work, following liquid and gas cyaniding, should be effected as soon as the work is withdrawn from the furnace

at the fermination of cyaniding. On high-temperature cyaniding, the work is air-blast cooled. After hardening, the parts are tempered at 160-200°C. Small non-critical parts from low-carbon steels are not tempered at all.

Parts to be machined after cyaniding are air-cooled, then tempered at 630-650°C; after machining they are hardened and then tempered

again.

Areas not to be cyanided are copper-plated, the thickness of the coat being 0.018-0.025 mm; the copper-plated layer should be compact.

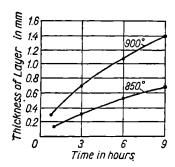


Fig. 19. Variation in the depth of cyanided layer in relation to the duration of gas cyaniding at various temperatures

The cyanided parts are checked in the shop by testing hardness with the aid of a file, diamond pyramid or cone, depending on cyaniding depth.

3. Low-temperature liquid cyaniding. Tools are cyanided after final machining, sharpening, cleaning, and degreasing. The temperature of the process averages 540-560°C (Tables 65 and 66).

 ${\it Table~65}$ Composition of Baths for Low-temperature Cyaniding

Sodium cyanide	Potassium ferrocyanide	Soda ash	Sodium or potassium hydrate	Melting point, °C
50-55 85-90 —	<u>-</u> 80-90	23-30 10-15 —	15-20 10-20	515 490-500

Relation of Holding Time to Cyaniding Depth in Low-temperature Cyaniding of High-speed Steels

Holding	Layer thicknes below-me	s in mm, the bath co	ntaining the yanide
time, min	90%	50%	30%
5 15 30 45 60 120 360	0.008 0.020 0.035 0.037 0.045 0.055 0.080	0.006 0.018 0.030 0.035 0.043 0.055 0.075	0.006 0.015 0.030 0.035 0.040 0.052 0.070

Tools smaller than those indicated in Table 67 are not cyanided because of high brittleness. Optimum cyaniding depth for cutting tools is 0.02-0.035 mm. Prior to cyaniding, the tools should be preheated in a separate furnace or near the crucibles. Long, slim tools should be

suspended throughout the heating and cooling.

One of the Kiev plants performs low-temperature cyaniding in a bath with a slightly poisonous compound of the following composition: 70% sodium hydrate and 30% potassium ferrocyanide. Potassium ferrocyanide is added to the bath in the same way as for high-temperature bath. The bath is analysed for CN and CNO once every two days. Normal cyaniding requires at least 5% CNO. Potassium ferrocyanide is added as the bath exhausts itself. The above-mentioned plant cyanides dies for pressure casting; for more information refer to page 185.

4. Low-temperature gas cyaniding. The composition of the gas cyaniding medium and the proportions of the gases are the same as for

high-temperature cyaniding.

After being held at 540-560°C the tools are air-cooled or cooled in a furnace down to 200°C. In the latter case the tools acquire a silvery

tint (Table 68).

One of the plants in Leningrad conducts low-temperature cyaniding of tools from high-speed steel in a "IL25" furnace, the kerosene feed amounting to 60-80 drops per minute and that of ammonium to 2 litres per minute. Aluminium-silicon alloy turnings are used as catalyst. Taking over the practice of the above-mentioned plant the author introduced gas cyaniding of stamping dies from grade 3X2B8 steel in a shopmade electric furnace with a retort 200 mm in diameter and 600 mm high. The feeding rates are: kerosene—10-15 drops per minute, ammonium—1 2-1.4 litres per minute. Positive mixing of the medium is essential. Fig. 20 presents a schematic view of a cyaniding furnace.

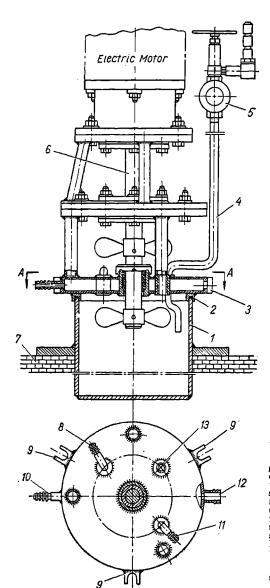


Fig. 20. Gas cyaniding installation:

1 — muffle; 2 — rubber gasket; 3 — water-cooled muffle Ild; 4 — tube for feeding kerosene into the furnace; 5 — kerosene dropper; 6 — impeller shaft; 7 — furnace lining; 8 — ammonia feed sleeve; 9 — shackle for fastening lid to crucible; 10 — water inlet sleeve; 11 — gas outlet sleeve; 12 — water outlet sleeve; 13 — kerosene feed orlice

Table 67

Recommended Holding Time for High-speed Steel Tools Cyanided in Liquid Media with Minimum Content of 50% Sodium Cyanide

Tools	Diameter or thickness, mm	Holding time at 560°C, min
Drills, reamers, and counterbores	3-5 6-10 10-15 15-20 20-30	6 8 10 12 15
Taps	Above 30 5-8 8-12 12-20 20-30	16-23 5 8 10 12
Broaches	Above 30 5-10 10-15 15-20 20-30 Above 30	14-18 8 12 14 16 20-25
Threading cutters with non-honed teeth	25-35 35-50	12 15
Threading cutters with honed teeth	Above 50 25-35 35-50	18 10 12
Tangential threading dies Hob cutters with non-honed teeth Hob and slot cutters with honed teeth Cylindrical profile cutters and end cut-	Above 50 50-60 Above 60 50-60 Above 60	15 8-10 30 40-50 16 20-26
ters with honed teeth	Up to 30 30-40 40-60 Above 60	10-12 16 20 25-30
Disk milling cutters	1-2 2-5 5-10 10-15 Above 15	6 8 12 15 18-23
Forming and tangential cutters Cutters for tooth-cutting machine Turning and shaping tools Gear cutters		12-30 10-12 12-40 15

Note. Holding above the recommended limits results in a brittle cyanided layer.

5. Low-temperature solid cyaniding. The cyaniding mixture consists of the following well-mixed components: 60-80% charcoal (size of grains approximates 3-6 mm) and 20-40% powdered potassium ferrocyanide. The tools are packed into iron containers in the same manner as for carburising (Fig. 15).

The temperature of the process is 550-560°C. Cooling is effected with-

out unloading the containers, in the air, down to 100-200°C.

Some of the defects which are met with in case-hardening of steel are listed in Table 69.

4. CALORISING

Calorising or aliting is employed to increase the resistance to scaling of steel articles working at high temperatures, and consists in the saturation of steel surface layer with aluminium. Calorising can be carried out in solid, liquid or gas media, but at the present time solid calorising treatment is most widely used. In this method, clean, degreased articles are packed into steel containers filled with thoroughly mixed powder-like mixture, consisting of 99.5% ferro-aluminium alloy and 0.5% ammonium chloride.

Articles are packed in the same manner as for the pack-carburising procedure. Areas not to be calorised are covered with asbestos or luted with fireclay. Emphasis should be placed upon thorough luting of the containers to avoid oxidation of the calorising mixture. A calorising compound consists of used and fresh mixtures (80% and 20%, respectively). The temperature of the process is within 850-1000°C. Calorising rate is approximately 0.05-0.08 mm per hour depending upon the temperature of the process.

The calorised layer is highly brittle which cannot be tolerated for a number of critical parts. These calorised parts are annealed at 900-1000°C for 3-6 hours to bring down the aluminium content of the sur-

face layer at the expense of its diffusion into the inner areas.

In thin parts, this may lead to a through calorising of the article and cause brittleness throughout the whole section. Such defects should be prevented by adequate choice of temperature and duration of annealing.

5. SULPHIDING (SULPHATING)

Sulphiding is used to raise wear resistance of machine parts and certain tools manufactured from high-speed steel by saturating their sur-

face layers with sulphur.

Sulphiding is brought about through treatment in liquid or solid media. In the liquid sulphiding process degreased dry articles are placed in baths containing: 1) sodium chloride—17%; 2) calcium chloride—38%; 3) barium chloride—25%; 4) iron sulphide—13.2%; 5) sodium sulphate—3.4%, and 6) potassium ferrocyanide—3.4%.

The former three salts are melted first, then, gradually, the remaining components of the bath are fused. After the complete fusing of the salts and subsequent cleaning of the bath, the latter's surface is covered with graphite. Sulphiding is effected within the 560-750°C temper-

Table 68

Recommended Holding Time for Tools from High-speed Steels in Gas and Solid Cyaniding

	Diameter or thickness.	Holding time in hours at 550-560° C		
Tool	mm	In gas medium	In solid medium	
Drills, counterbores, reamers Taps	Up to 15 15-25 25-50 Up to 15	1.0-1.5 1.5-2.0 2.0-3.0 0.5-1.0	2.0-2.5 2.5-3.0 3.0-4.0	
Threading cutters:	15-25 25-50	1.0-1.5 1.5-2.0	_	
a) honed teethb) non-honed teeth	25-50 Above 50 25-50 Above 50	1.0-1.5 1.5-2.0 1.5-2.0 2.0-2.5	1.5-2.0 2.0-3.0 2.0-2.5 2.5-3.0	
Threading chasers and dies: a) screw pitch up to 2 mm b) screw pitch above 2 mm		1.0-1.5	1.5-2.0	
Worm and slitting cutters a) honed teeth	50-75 Above 75	1.0-1.5 1.5-2.0	2.0-2.5 2.5-3.0	
b) non-honed teeth Cylindrical, profile and end cutters	50-75 Above 75 Up to 50 50-75 Above 75	1.5-2.0 2.0-2.5 1.0-1.5 1.5-2.0 2.0-2.5	2.5-3.0 3.0-4.0 2.0-2.5 2.5-3.0 3.0-4.0	
Disk cutters	Up to 10 Above 10	1.0-1.5	2.0-3.0 3.0-4.0	
Round-nose cutters	Up to 5	1.0-1.5	2.0-2.5	
Tangential cutters	5-15 10×10 25×25	1.5-2.0 1.5 2.0	2.5-3.0 3.0 3.5	

Notes. 1. Holding time is counted off from the moment the tool is heated through.

2. Holding time in excess of that recommended results in a brittle cyanided layer.

Processes
Case-hardening
Ξ,
Defects

	Derects III	Detects in case-naturaling Processes	
Defect	Detection	Cause	Prevention and correction
	1. Defects	1. Defects in Carburising of Steel	
Pitting Vitreous bulges on the surface of articles	Visual inspection Visual inspection	Over 3-6% of sulphate in the carburiser Presence of sand in the carburiser	Prevention: Keep sulphate content below 3-6% Prevention: Keep sand out when storing or unpacking
Uneven carburising depth	carburising Fracture inspection	Poor mixing of carburise. Uneven heat soaking of containers	carouniser. Pass the carburiser through a sieve Prevention: Thorough mixing of the carburiser; use of a solution of carbonates
Excessive carburising depth. Increased carbon content in the carburised layer	Fracture inspection Microscope inspection of microsection	Elevated temperature and holding time on carburising. Strong carburiser used	in the manufacture of the carburiser. Uniform heat soaking of the containers Correction for excess carbon: Hardening in oil or normalising at 900-910° C for carbon steels and 850-860°C for alloy steels with sub-
Low carbon content in the carburised layer	Hardness test. Microsection inspection	Carburising at low temperature. Weak solid carburiser or insufficient gas or kerosene feed in gas carburising	sequent hardening at 760° C Correction: Repeated carbu- rising at normal tempera- ture and with normal car- buriser Prevention: Use adequately strong carburiser

Correction: Repeated carburising at normal temperature and with normal car-	buriser. Prevention: Use stronger carburiser Correction: Repeated short-term carburising at normal temperature. Prevention: Cooling of the boxes	in the air Prevention: Strict conformance to technology chart. Use natural coarse-grained	Correction: Heating in containers, filled with charcoal plus 3-5% soda ash, up to 920-940°C and holding at this temperature for	2-4 hours Correction: 1. Sub-zero treatment. 2. Improving with subsequent hardening from 750° C. 3. Hardening with air-blast cooling to 650-	Correction: Accurate proportioning of carburising material fed to the furnace	
Ca	kerosene feed in gas car- burising Too slow cooling of contain- ers in the furnace after carburising (particularly when coarse-lumped car-	buriser is used) Dirty, greasy work surface, shrinkage of carburiser, abnormality of steel	Use of a strong carburiser	High carbon content in carburised layer; too rapid quenching	Excessive gas or kerosene feed to the furnace	
Fracture and microsection inspection	Hardness test. Micro- section inspection	Hardness test	Visual inspection	Hardness test	Visual inspection	
Insufficient carburising depth	Surface decarburisa- tion of carburised layer	Non-uniform carburis- ing	Scaling-off of hardened carburised layer	Increased amounts of Hardness test residual austenite. Low hardness. Defect is characteristic of high-alloy steels	Considerable sooty deposits on gas carburising	•

Table 69 continued

Prevention and correction		- Pi		<u>a</u> <u>a</u>
Cause	2. Defects in Nitriding of Steel	Warping and distor- Checking for size and stresses caused by the differ- heat core core	Presence of distinct boundary between the nitrided layer and the core	Slag inclusions in the part surface Excessive saturation of the skin by nitrides as a result of coarse-grained structure and surface decarburisation on preliminary heat-treatment
Detection	2. Defects	Checking for size and beat	Visual inspection	Bulges on the surface Visual inspection Brittleness (surface Visual inspection. Diapiting) of the nimond pyramid hardided layer ing of indentations
Defect		Warping and distortion	Cracking (scaling-off of the nitrided lay- er)	Bulges on the surface Visual inspection pitting) of the nitrided layer trided layer ing of indental

	Prevention: Phosphating prior to nitriding and careful preparation of the surface of the articles	Correction: Repeated nitriding with shorter holding time. For small-sized threaded parts this defect is irreparable		Prevention: Strict checking of neutral salts for sulphates Prevention: Oil hardening with air-blast cooling down to 600-650° C. Correction: Sub-zero treatment	Correction: Heating a salt- petre bath at 550-560° C with a 30-min holding	
	iamond pyramid Local tinned areas on the hardness test nitrided surface	Insufficient holding time, great dissociation degree of ammonium, low temperature of the process	3. Defects in Cyaniding of Steel	Over 0.7-0.8% of sulphates in neutral bath salts Inadequate cooling rate	Prolonged holding and heating in high-percentage baths. High ammonium content (above 40%) in gas cyaniding	
_	Д	Testing by 1-min immersion in a 10% aqueous solution of blue vitriol	3. Defects	Visual inspection Hardness test	Inspection of the indentation after the diamond pyramid hardness test	
_	Uneven hardness. Characteristic of parts subjected to local	Invitating Insufficient or porous layer. Characteristic of anticorrosion ni- triding		Pitting on high-tem- Visual inspection perature cyaniding Residual austenite in the layer, low hardness after hardening. Characteristic of al-	loy steels Brittleness of cyanided layer of articles from high-speed steel	
						10

ature range, but in no case above the tempering temperature of the article. Approximate sulphiding rate is 0.1 mm per hour at 560°C. The duration of sulphiding of high-speed steel tools, depending upon tool size, is within the limits of 1 to 3 hours. In solid sulphiding process, the articles are packed in steel containers filled with a compound having the following composition: 1) ground iron sulphide—30-40%;

2) graphite-50-60%; 3) potassium ferrocyanide-10%.

The heating temperature is the same as for liquid sulphiding. Holding time in solid sulphiding is approximately three times greater than in the liquid process. After sulphiding, the parts are flushed in hot water and heated in an oil bath at 110-120°C in order to protect them against corrosion. The hardness of the articles after sulphiding remains practically unaltered.

6. CLEANING AND PICKLING OF ARTICLES AFTER HEAT-TREATMENT

Articles are cleaned from salts, oil, and dirt in an aqueous solution containing up to 3% caustic soda or 10% soda ash. The temperature of the solution is approximately 80-90° C. Flushing takes 5 min and longer. The solution is replenished at least every ten days.

The articles are descaled in sand- or shot-blasting installations and

in pickling baths.

Sand-blasting makes use of dry river sand, with grains sized to 1-2 mm, the necessary air pressure being within 5-6 atmospheres.

Sand-blasting may be replaced, for the sake of improving the working conditions, by hydro-sandblasting, in which the articles are dressed by a mixture of 50% sand and 50% water. This mixture is atomised by high-pressure air supplied to the mixer outlet.

Hydraulic cleaning using a 150-atmosphere water jet may also be

used to dress larger articles.

Shot-blasting uses shots 0.5-2.0 mm in diameter.

Ferrous metals are pickled in a 5-18% aqueous solution of sulphuric acid or 7-20% aqueous solution of hydrochloric acid. To avoid overpickling, the solution carries 1% of the "KC" or 0.5% of the "K" organic additions (by volume of the pickling solution). The addition is effective for 100-150 hours. Pickling time ranges from 30 to 60 min, the temperature averaging 40-90°C for the sulphuric acid and 30-60°C for the hydrochloric acid. The temperature should be increased as acid concentration decreases. The minimum permissible acid content of the working bath is 3-4%.

Stainless steels are pickled at 40-50°C in a bath composed of: 47% hydrochloric acid, 5% nitric acid and 48% water. They are then immersed in a 5% solution of nitric acid and held over for 3-5 min. Pickled articles are flushed in warm water, then in a 0.5% solution of caustic soda, and dried at 120-150°C to eliminate pickling brittleness.

Overpickling may occur when chemical descaling is applied; because of this, parts finished to size or designed with negligible machining allowance are to be electrochemically pickled in special shops.

7. ANTICORROSION PROTECTION FOR ARTICLES AFTER HEAT-TREATMENT

After heat-treatment associated with the use of salts, alkalis, water, and various other materials, which may cause corrosion of articles on long-term storage, the latter are given the anticorrosion treatment involving a 5-min dipping of cleaned, flushed and dried articles in a 20-30% aqueous solution of sodium nitrite; this done, the articles are wrapped in paper impregnated with the same solution.

In such a wrapping, the parts can be stored for a long period of time.

Chapter VI

HEAT-TREATMENT TECHNOLOGY

1. APPROXIMATE HEAT-TREATMENT SCHEDULES AND MECHANICAL PROPERTIES OF VARIOUS GRADES OF STEELS

Plain carbon steels (grades from Cr. 1 through Cr. 7) are heat-treated and case-hardened according to charts specified for structural steels with similar carbon content.

Free-cutting steel is heat-treated and case-hardened according to the

same charts as above.

High-quality structural carbon steel, depending upon carbon content and application, can be normalised, annealed, hardened or carburised. In shop practice, steels with up to 0.5% carbon are normalised, while those with a higher carbon content are annealed to improve machinability. Cooling rate is 50-150°C per hour down to 500-600°C with subsequent air-cooling.

Annealing is to be carried out in the lower range of temperatures indicated in Table 70, while the upper temperature range is preferred

for normalising.

One must bear in mind that steels of the medium-carbon grades (grades 45, 50, etc.) are inclined to cracking when hardened in water, and they should be quenched with intermediate transfer to oil bath.

Steels with high manganese content have a tendency to overheat. and because of this their heating and holding time should be as short as possible.

After tempering at 400-600°C, manganese steel should be cooled

rapidly so as to prevent temper brittleness.

Steel castings. Large steel castings which acquire quite a coarsegrained non-uniform structure as a result of slow cooling are, first, homogenised at Ac. +150-250°C, then annealed at normal temperature.

Medium- and small-size steel castings which are currently produced in great numbers by precision casting are, as a rule, annealed or normalised at $Ac_1+20-50^{\circ}C$ for hypocutectoid steels and at $Ac_1+20-60^{\circ}C$ 30°C for eutectoid and hypereutectoid steels. Holding time on annealing and normalising should be increased by 1.5 times as compared to that specified for forgings and rolled blanks.

Castings can be, depending upon the specifications, subjected to improving (heavy duty machine parts) or hardened with subsequent low-temperature tempering (tools, etc.) according to usual schedules.

Steel castings subjected to any kind of heat-treatment are less ductile than forgings and rolled stock.

Approximate Heat-treatment Schedules and Mechanical Properties of Quality Structural Carbon Steels

	For Far-		e.	Mechanical properties			
Grade	Heating to perature hardening, normalisin or full anne ing, °C	Cooling medium	Tempering temperatu °C	Hard H _B	iness R _C	Tensile strength, kg/mm²	Elonga- tion, %

Carbon Steels with Normal Manganese Content Steels Subject to Carburising

08 10 15 20	900-960 900-940 890-930 880-920	Air Air Air Air	 	131 137 143 156		32 34 37 41	33 31 31 27 25
20	880-920	Air	_	150	_	41	25

Steels Subject to Improving

25	860-900	Water	200-300	ı -	33-27	1	I —
30	850-890	Water	200 300		35-30		_
			600			55	20 .
35	840-880	Water	300-400	[-	50-41	 	
			400-500	i —	41-31	l — I	_
			500-600	_	31-23	l —	
40	820-860	Water	200-300		52-48		
-]	300-400		48-41] _ '	
			400 - 500		41-33		
			500-600		33 · 22	<u> </u>	
45	810-840	Water	200-300	:	54-50	l _	_
50			300-400	_	50-41	l	_
			400-500		41-33		
			500-600		33-24	l	_ `
			600-700		24-15	l	
55	790 - 830	Water	430-450		42-33		
60	785-820	Water,	400	321	_		
		oil	550-620	241-207			
65	785-815	Water,	300-400	2.7 20.	52-45		
-		oil -	400-500	_	45-37	l	<u> </u>
		"	500-600		37-28	l	:
70	770-815	Water,	400	_	46-39		_ ~
75	010	oil	610-670	260 230	10-03		
80	770-800	Water,	375-400	200-200	49-40	!	
85		oil	560-600		33-26	i i	
			200-000	-		-	· _
ı	5 .	Ė		ا . ا			ļ

or g		<u> </u>	Me	chanical pr	operties	
Grade Heating te perature for hardening, normalisin or fullanne ing, °C	Cooling medium	Tempering temperatu °C	Haro H _B	iness R _C	Tensile strength, kg/mm²	Elonga- tion, %

Carbon Steels with Increased Manganese Content Steels Subject to Carburising

15F !	880-920	1 Air	i —	163	I —	40	24
20Γ	860-900	Air		197		43	22
25Γ	850-890	Air		197		43	22

Steels Subject to Improving

		••					
30Γ	840-880) Water	200-220	_	54-52	1 -	i —
			400-500	_	38-33	l —	_
	:		600	196	i —	70	15
40Γ	820-860	Water	600	235		80	15
50r	800-840	Water,	200	_	5 0		
		oil	550-600	295-246		85	8
60Г	790-820	Water,	200-220	_	60-56		
		oil	380-420	_	46-40		
		l :	500-600	302-269	l —	85	9
65Γ	790-815	Water.	150-200	_	60-58	_	_
		oil '	200-300		58-54		_
			300-400	 .	54-47		
			400-500		47-39		_
)		500-600	-	39-30		
70F	780-800	Water,	200-220		62-58		_
		oil ´	400-450	_	46-40	_	_
	,] [
]				
,		J.	,				

Notes. 1. After carburising, steels are hardened and annealed, according to usual schedules.

2. Mechanical properties of steels mentioned in the table refer to samples 60-80 mm in cross-section.

It should be borne in mind that steel castings, supplied to the heat-treatment shops, show a decarburised outer layer, the decarburisation being sometimes very high. Decarburisation in the casting bay is

caused by scales present in the moulding sand. When decarburisation is slight, the defect can be corrected by annealing in a carburising medium. Castings are rejected if there is considerable decarburisation.

Table 71

Approximate Heat-treatment Schedule and Hardness Number of Steel Used for Plough Shares

Grade of steel	Heating temperature on hardening, normalising or full annealing, °C	Cooling medium	Tempering tempera- ture, °C	Rockwell hardness number, R_C
Л53	800-820	Water	180-250	58-55
			250-300	55-52
			300-350	52-50
			350-400	50-44
	l			l.

Structural alloy steels are either carburised or improved, depending on their carbon content. Alloy steels have lower heat conductivity, and because of this fancy-shaped parts should be heated slower than those from carbon steel.

Table 72 presents methods for softening alloy steel. Some grades of complex alloy chrome-nickel steels cannot be annealed and are softened only by means of high tempering with prolonged holding time. The parts are cooled down to 500°C within the furnace, and then in the air.

It is a customary practice in some plants to substitute high tempering for annealing for other grades of steel as well, as tempering is simpler than annealing, and gives good results. When one is not certain of the correctness of the forging procedure used for treating blanks, the latter should be normalised prior to tempering. Approximate heat-treatment methods for alloy steels are given in Table 73. Hardened parts from alloy steels containing chromium or manganese should be cooled in oil or air, after tempering, to prevent temper brittleness.

be cooled in oil or air, after tempering, to prevent temper brittleness. Spring steel is generally not ground, and special measures are, therefore, to be taken to prevent heat-treatment decarburisation. This applies especially to silicon steels which decarburise much quicker than other grades of steel. When hot-rolled, as well as heat-treated, spring steel is to be softened, high tempering at 660-700°C with minimum holding should be used. A tentative method for heat-treatment of spring steel is presented in Table 74.

 ${\it Table~72}$ Softening Procedures for Alloy Structural Steels

Grade	Operation	Heating temperature, °C	Cooling rate conditions
			10. 500.0
30X, 35X, 38X, 40X, 45X, 50X	Annealing	800-860	40-50°C per hr
15ΧΦ, 20ΧΦ, 40ΧΦ, 50ΧΦ	Annealing	840-860	50-60° per hr
33ΧC, 37ΧC, 40ΧC 40ΧΓ, 35ΧΓ2	Annealing	860-880	40-60° per hr 40-60° per hr
40ΧΓ, 35Χ Γ 2	Annealing	840-850	40-60° per hr
20XFC	Isothermal	880-900	Isothermal
	annealing	· ·	holding at
	l		_ 660-680° C
25XTC	Isothermal	870-890	Ditto
30XTC	annealing	000 000	Ditto
30X1 C	Isothermal annealing	860-880	וונס
35 Х ГС	Isothermal	850-870	Ditto
35 AT C	annealing	000-010	Ditto
27СГ, 35СГ	Annealing	860-880	50-60° per hr
25HA, 30HA	Normalising,	840-860	Air
,,	tempering	620-640	Air
13H2A, 13H5A, 21H5A	Tempering	620-640	Holding, 5hrs
25H3, 30H3A	Tempering	650-670	Ditto
15XMA	Annealing	880-900	30-40° per hr
20 X M	Annealing	860-880	Ditto
30 X M	Annealing	840-860	Ditto
35 X M	Annealing	830 - 850	Ditto
38ХЮА, 38ХМЮА	Annealing	860-880	60-80° per hr
35 X МФ А 20 X Н	Annealing	860-880	40-50° per hr 30-40° per hr 30-40° per hr
40XH, 45XH	Annealing	840-860	30-40° per hr
50XH	Annealing Annealing	800-820 790-810	Ditto
		880-900	Ditto
12XH2, 12XH3, 12X2H4	Normalising, tempering	600-640	_
20 X H3, 30 X H3, 37 X H3A	Normalising,	640-650	Prolonged
,	tempering	415 555	holding
20 X Н4ФА	Annealing	850 - 860	30-40° per hr
13XHBA, 18XHBA,	Tempering	650-680	Prolonged
18XHMA, 25XHBA			holding
12X2H3MA, 12X2H4MA, 20XHM, 33XH3MA,	Tempering	650-670	Holding for
20XHM, 33XH3MA,			5-6 hrs
40X H.M.A	1 1:	050 070	40 500
	Annealing	850-870	40-50° per hr
45XHBФA		000 000	•
45ХНВФА 18ХГТ	Normalising, tempering	900-930 670	'

Approximate Heat-treatment Schedules and Mechanical Properties of Alloy Structural Steels

						Mechanical properties	properties	
		Heating tem- perature for	Cooling	Tempering	Hardness	ness	:	
	Grade	nardening and normalising, °C	medium	temperature, °C	H_B	RC	lensile strength, kg/mm²	Elonga tion, %
{		200	- -	000		04		
	351.2	008-018	110	200-300	ļ	20-48	i	1
				300-400	1	48-38	1	1
				400-200	1	38-34]	1
				550-620	241-217	ı	75-85	15-18
	$40\Gamma 2$	810-840	Oi!	250-600	300-230	ł	ı	!
	45 Г 2	800-840	Oil	300 - 400		49-43	1	1
				400-200	1	43-33	1	ł
				200-600	325-262	I	1	ı
	$50\Gamma 2$	790-820	Oil	200-600	321-269	1	1	Up to 9
	15X	780-820	Oil or	200	200	l	20	10-11
j_ 54	20X	780-820	water Oil or	200	220	i	80	10
- NI	15XP	780-810	water Oil or	200	1	ı	75	12
	30X	840-870	water	540-380	907-999	I	75-80	14-18
119	35X	840-865	i	180-200		45-50	3 1	: 1
			;	480-520	1	28-31	ł	1
, .	35 X P	860	Oil	260	1	1	95	12
7. 18	38X A	830-860	Oil	180-200	1	45-52	1	}
1			-	500-560	285-375	l	130-100	ſ
		_		099-099	200-285	1	100-70	1

Table 73 continued

		Elongation, %	1	1	}	1	ı	12	1 '	20 6	n (က	x 0	12	6	6	6	1	10	=	9	<u>2</u> «	- <u></u>	:	
properties		Tensile strength, kg/mm²	1	1	1	1	ľ	00 1	1	100	င္တ	105	110	90	100	001	150	1	100	001	18	86	86	3 1	
Mechanical properties	ness	RC	54-52	52-45	45-36	36-30	30-27	1	25-52	1	1	1	1	1	1	l	1	53-45	!	I	53-45	1 8	1 1	55-52	
	Hardness	H_{B}	1	ı	i	ı	J	1	1 8	280-302	230-280	18	300	220	1	1	1	1	272-302	1	1 200	602-992	> 269	1	_
	Tempering	temperature, °C	200-300	300-400	400-200	200-600	600-650	540	200-220	500-580	080-090	220	200	180-220	180-220	200	200	200 - 250	250-600	250-600	220-250	950-950	620-640	250-280	
		medium	Oil					.	<u>.</u>		:	5	5	Oi1	Oil	Oil	Oil	Oil	į	Oil	Oil	Ċ	;	Oil	
	Heating tem- perature for	normalising, oC	825-860					860	8.50-850		0	840	815-845	800-820	800-820	870	820	088-098	1	850	820-860	900-930		900-930	
		Grade	40X					40XP	45X		11.25.27	45ÅŲ	200 200	JX02	18XFT	20XFP	30XLT	40XF		40XIP	35XF2	33XC		37XC	

7. 2. 8. 6. 6. 6. 6. 6. 6. 6. 6. 6. 6. 6. 6. 6.
90. 175. 175. 186. 186. 186. 186. 186. 186. 186. 186
55.52 42.50 1.1 1.1 1.2 1.2 1.3 1.3 1.3 1.3 1.3 1.3 1.3 1.3 1.3 1.3
241-269 241
600-650 240-260 600-650 230 475 550 600-620 650 550 650 650 650 650 650 65
Oil Water Water Air Oil
900-920 910-940 870-900 880-890 860-870 850-870 840-870 840-870 840-870 840-870 840-870 840-870 870-820 780-800 820-840 860-830
40XC 27CF 35CF 36F2C 15XM, 20XM 30XM 30XM 35X2MA 35X2MA 35X2MA 26XH 15XΦ, 20XΦ 40XH 45XH 12XH2 12XH3A 20XXH3A 20XXH3A 20XXH3A 20XXH3A 20XXH3A 20XXH3A 20XXH3A 20XXH3A

Table 73 continued

מחוב ום רחווווומבת			Elongation, %	6	15	7	69	120	10	12	o (12	10-12	12	12	12	=	=	12	12	01	- 10	12	0	01	
an an a	properties		Tensile strength, kg/mm²	100	160	150	110	96	100	110	165	110	100-120	120	6	120	95	100	105	100	150	110	8	120	100	
	Mechanical properties	ness	R_C	1	52-45	49-44	39-35	35-25		1	53-48	1	1	1	1	1	1	!	1	!	1	1	1	l	1	
		Hardness	H_{B}	≥ 292	- 100	120-120	i	 - - -		1	1	1 305	1	!	1	1	i	١	ı	1	1	1	1	l	1	
		Tempering	temperature,	530	200-220	390-420	550-600	590	230	580	200-220	610	280	550-620	069	550-620	180	180	170	180	180	200	220	200	530	
			medium	liO	io –	Oij		ë	55	Oi.1	Oii		ij	ïÖ	iö	ΙίΟ	Oil	i.o	Oil	Ϊ́Ο	Oil	Oil	0!1	Oil	Oii	
	Hooting tom	perature for	normalising.	810-840	810-840	860-880		850-860	860	870	830-850		980	820	820	820	820-850	820	770-810	098	840-860	880	820	870	860	
			Grade	30XH3A	37 X H3A	45X HMΦA		36 X H1 M A	30XHBA	38XHBA	40XHBA	40XHMA	30X2HBA	38X H3B Φ A	20X H4ΦA	38XH3MΦA	15XLHT	15X2FH2T	15X2FH2TPA	18XLH	25X2FHTA	30XFHA	38X LH	30X2FH2	18XCHPA	

10 10 6 6	1	i	1	1	1	16	1	10	15	10	15	. 10	10	
145 80 150 110	120-140	100-110	80-90	70-80	ı	9/	160-180	95	100	100	09	110	130	
50-46	1	1	1	ı	53-46	I	50-46	I	l	ı	1	38-26	44-33	
228 400 310	337-390	285-315	235-265	211-235	I	≥ 235	1	≥ 260	≥ 286		1	1	1	
200 500-520 200 500 225-250	480-520	540 - 580	600-640	640-680	200-250	640-660	200-300	650	029-009	640	180	180-200	180-200	
io o					Oil		Oil	Oil	liO	Oil	Oil	Oil	Oil	
870-900 890-910 890-910		_			088-098		890-910	930	930-920	930	098	830-850	810-830	
20XFCA 25XFCA 30XFCA					35X ΓCA		30XTHCA	38X IO	38X MIOA	38ХВФЮ	13H2XA	15ХГНР	20XTHP	

Table 73 continued

	2001	 			Mechanical properties	properties	
	perature for		Tempering	Hardness	ness	:	
Grade	nardening and normalising, oC	medium	temperature, °C	НВ	RC	Tensile strength, kg/mm²	Elongation, %
14XF2HP	820-840	Oil	180-200	1	44 - 33	110	10
14XF2CP	830-850	Oil	180-200	ı	44-33	110	10
15XHF2BA	880-920	Oil	200	ı	1	1	1
	810-830	li0	180-200	I	47-35	115-120	13
15X2F2CBA	880-920	Oil	200	ı	ŀ	ı	1
	830-850	Oil	180-200	I	47-35	115-120	12
35X2FCBA	880-890	Oil	200	555-477	1	200	80
			460	477-401	!	150	6
			009	341-286	1	105	13
			650	286-241	!	ı	ı

Note. The mechanical properties of carburised steels refer to the core, their hardness number being within $56.62 \, R_{C}$.

Table 74
Approximate Heat-treatment Schedules and Hardness Numbers of Spring Steels

Grade	Heating tem- peratures for hardening and normalising, °C	Cooling medium	Tempering temperature, °C	Hardness number, R _C	Remarks
55 65	790-830 785-815	Water Oil	430-450 300 400 500 600	33-42 52 45 37 28	
70	780-815	Oil	380		
75 25	780-815	Oil	380	39-46	
85 65 D	770-800	Oil Oil	375-400 380-430	40-49	
65Γ 55ΓC	790-815 790-820	Oil	390-430	42-47 39-46	
55C2	850-890	Oil	580 480-500	25-31 39-43	
60C2	840-870	Oil	400-510	. 43-50	
60C2A	840-870	Oil	400-425	40-49	
0002/1	820-840	Water	400-425	40-49	For wire up to 13 mm in dia For wire above
63C2A	860	Oil	400-510		13 mm
70C3A	850	Oil	400-510		in dia
50ΧΓ,	000	On	100-010		
50 Χ Γ A	840-870	Oi1	450-480	41-43	
60C2XA	860	Oil	420		
60C2XΦA	840	Oil	450		
65C2BA	840	Oil	450		
60C2H2A	840	Oil			Ì
55CΓ	880	Oi1	400-510		
60CLA	860	Oil	400-510		
50 X Φ	840-870	Oil	475		ĺ
50 Χ Φ A			370-420	42-50	
50 Х ГФА	850-880	Oil	550	39-43	1
58CH2A	875-900	Oi1	400-425	46-49	

Description	Mechanical properties		
Spring steel wire, heat-treated, from 1.2 to 5.5 mm thick Carbon spring steel wire, from 0.14 to 8 mm thick: grade I grade II grade III Steel spring band, heat-treated, from 0.08 to 1 mm thick: 1 (first grade of strength) 2 (second grade of strength) 3 (third grade of strength)	Tensile strength from 180 to 130 kg/mm²* Tensile strength: from 310 to 150 kg/mm² from 270 to 140 kg/mm² from 225 to 110 kg/mm² H _V 375-485 H _V 486-600 H _V over 600		

^{*} Depending on wire thickness.

Table 76

Approximate Heat-treatment Schedule for Ball-bearing Steel

	Annealing		Hardening		Tempering	
Grade	Heating tempera- ture, °C	Hardness number, H _B	Heating tempera- ture, °C	Minimum hardness number, R _C	Heating tempera- ture, °C	Hardness number,
ШХ6	780-800	170-207	800 - 825	62	150-170	60-62
	500 000				200-220	56-58
ШХ9	780-800	170-207	825-840	62	150-180	60-62
					200-260	60-55 55-50
ШХ10	820-830		840-850		400	50-45
ШХ15	780-800	170-207	835-855	62	150-200	64-61
					200-300	61-56
					300-400	56-49
				}	400-500	49-41
ШХ15СГ	780-800	170 007	015 005	1 00	500-600	41-28
m V 19C1	100-000	170- 2 07	815-835	62	170-200	61-65

 $\it Note.$ Oil serves as a cooling medium for ball-bearing steels, while kerosene is used for plain-shaped articles requiring high hardness.

Ball-bearing steel. Ball-bearing steel is heat-treated according to procedures recommended in Table 76. On annealing, the work, following heat soaking, is cooled to 500-550°C at a rate of 30°C per hour for grade IIIX15CF steel and below 40°C per hour for grades IIIX6, IIIX9, IIIX15, after which the blanks are cooled in the furnace with the power switched off. Isothermal annealing is also being applied, according to the following sequence of operations: heating to 800°C, holding till heat soaking is complete, cooling down to 700-720°C, holding for 3-4 hrs, cooling in the furnace down to 400-450°C, followed by cooling in the air.

If no correction of structure is required and the annealing aims only at softening, high tempering at 680-700°C for 3-4 hrs may then be

successfully applied.

When hardening large parts from the $\coprod X15$ steel, cooling should be effected in kerosene down to room temperature in order to ensure hardness above R_C =60. In this case all sharp changes of section and sharp grooves are to be covered with a mixture composed of asbestos and

clay, so as to prevent cracking.

Electric steels and alloys are annealed to improve their magnetic properties. Best results are obtained with vacuum annealing (Fig. 21). If no vacuum installation is available, the work is to be annealed in containers with solid protective media—white mountain sand, asbestos powder, aluminium oxide, etc. The protective mixture should be calcined prior to use in order to eliminate all moisture. Fig. 22 shows the packing of the containers. When a thin oxide skin is desirable, the parts are annealed in the same packets but with no protective mixture. Hydrogen annealing did not find wide application, because of explosion hazards and possible surface decarburisation.

Annealed parts are not to be distorted, dropped or hammered, as

this lowers their magnetic properties.

When cutting or stamping of permalloy is difficult, because of strain-hardening, the alloy should be annealed at 600-650°C (holding for 1-2 hrs and cooling at a rate of 100-200°C per hour).

Annealing schedules for electric steels and permalloys are given in

Table 77.

Magnetic steel is heat-treated according to schedules given in Table 78 in order to ensure the requisite magnetic properties. Holding time is counted off as soon as steel is heated to the required temperature.

A special post-rolling softening treatment intended to obtain a structure, ensuring best magnetic properties after hardening, is practised in some metallurgical plants. This must be taken into account by the users who should not re-anneal the stock as this will change the steel structure and entail (after hardening from normal temperatures) worsening of magnetic properties of steel.

When bending magnets, the work should be heated below 700°C for grades EX5K5 and E7B6, 750°C for grades EX and EX3 and 800°C for grade EX9K15M, in order not to impair magnetic properties

of steel.

Heat-treatment schedules for cast magnets are presented in Table 79. Stainless steel, on annealing, is to be cooled to 250-300°C within the furnace. Holding time on heating should be cut to a minimum so as

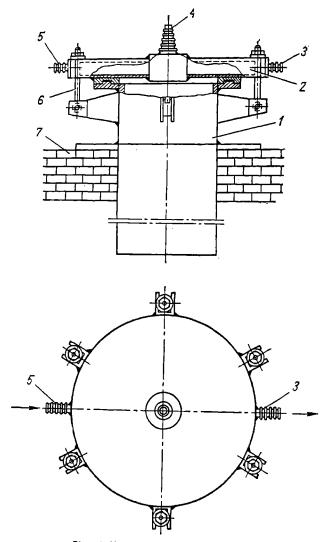


Fig. 21. Vacuum annealing installation:

1 - muffle; 2 - wafer-cooled muffle lid; 3 - water outlet nipple;
4 - air exhaust nipple: 5 - water inlet nipple; 6 - coupling bolts;
7 - furnace lining

to prevent grain growth. Depending on steel grade, the hardening procedure aims at toughening or increasing ductility and chemical resistance (tables 80 and 81).

It should be noted that hardening always increases the corrosion resistance of steel. Chromium stainless steels are characterised by tem-

per brittleness, and for this reason, they are to be cooled in water or oil

after tempering.

Manganese wear-resistant grade \(\Gamma\)13 steel is given a prolonged tempering at 600-650°C in order to improve its machinability.

Grade \$\Gamma 13 acquires high wear resistance after hardening in water at 1050-1100°C. Holding time on heating

is 2-3 hours.

Given below are approximate mechanical properties obtained after hardening:

Tensile strength $\sigma_b = 70 \cdot 100 \text{ kg/mm}^2$, elongation $\delta=25-45\%$,

specific resilience $a_k = 20-25 \text{kgm/cm}^2$, Brinell hardness number 197-212.

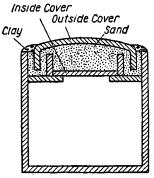


Fig. 22. Containers for oxidation-free annealing

After fempering at 400-450°C, the hardness of the Γ13 steel may reach as high as $H_R=400$. This steel has a tendency to decarburise. Because of its very low heat conductivity, the steel should be heated slowly up to 600°C, to prevent cracking.

Carbon tool steel. The work is cooled from annealing temperature at a rate of 20-50°C per hour down to 550°C, then the cooling is con-

tinued in the air.

This steel acquires best machinability and minimum hardness aft-

er sub-critical annealing (spheroidising).

Sub-critical annealing (spheroidising) is realised by heating and holding the said steel at 730-750°C, then cooling it down to 500-550°C at a rate of 20-30° C per hour, and, finally, cooling it in the air.

Carbon tool sfeel is "high" tempered at 680-700°C with prolonged holding and subsequent air-cooling, to lower hardness and remove stresses.

Tools are cooled, following hardening, chiefly in water and, subsequently, in oil. Small-sized tools are hardened in oil or kerosene (Ta-

ble 82).

Alloy tool steel. The work is first cooled from annealing temperature at a rate of 30-40°C per hour down to 550°C, and then in the air. In case of isothermal annealing, the steel can be air-cooled from isothermal holding temperature. When only decrease in hardness and stress removal are necessary, the work is high tempered at 650-680°C and then air-cooled.

Because of low heat conductivity, fancy-shaped tools should be heated slowly.

Table 77

Annealing Schedules for Electric Steels and Permalloys

				•	
Grade of steel or alloy	Purpose of annealing	Heating temperature,	Holding time, hrs	Cooling conditions	Equipment and materials required
	Improvement of magnetic properties and formation of oxide skin	830-870	en	50° C per hour. Parts are dis- charged from the container at 150° C	Furnace and heat-resisting steel box
Steels: 931, 941, 942, 945, 947	Improvement of magnetic properties without formation of oxide skin (bright annealing)	006-098	3-4	Ditto	Furnace. Heat-resisting steel box. Protective atmosphere
	Considerable improvement of magnetic properties due to conse-grained structure and removal of detrimental impurities from surface layer (no oxide skin after annealing)	1000-1050	10-20	60-70° C per Furnace. hour hydrog	Furnace. Vacuum or hydrogen annealing installation

Furnace. Vacuum or hydrogen annealing installation	Furnace	Furnace. Vacuum or hydrogen annealing installation	Furnace. Vacuum annealing installation or container from heaf-resistant steel, and protective mixture
Cooling in the furnace. Discharge at 100° C	Slow cooling within the furnace to 650°C, then in an air jet	Slow cooling within the furnace to 200° C, under vacuum	100-150° C per hour to room tem- perature
3-4	10	8-15	3-4
780-820	1170-1230	1000-1100	850-930
Considerable increase in magnetic properties (no oxide skin after annealing)	Improvement of magnetic properties and formation of oxide skin	Improvement of magnetic properties. No oxide skin after annealing	9, 3A, 3AA Improvement of magnetic properties
9310	Permalloys: iron-nickel	anoys or various grades	9, 3A, 3AA

Note. When no heat-resistant steel Is available, the container should be made of plain structural steel. When annealing with the formation of oxide skin is contemplated, the parts are packed into containers without sand, with the packing being loose so that the annealing metal should occupy 20-25% of the volume of the container.

Heat-treatment Schedules for Magnetic Steels

	Normalising	ising	Tempering	ring		Hardening)g	
Grade of steel	Heating temperature, °C	Holding time, min	Heating temperature, °C	Holding time, min	Preheating temperature, °C	Final heating temperature,	Holding time, min	Cooling
, H	001	ш			000	0 10	;	
4	1000	2	1	ļ	nna-nna	920	- 10-12 - 10-12	<u>.</u>
EX3	1050	'n	I	1	200-000	850	10-15	Ditto
E7B6	1200-1250	1	1	1	200-600	820-860	٠.	Ditto
EX5K5	1150-1200	1	ı	1	500-650	950	, <u>e</u>	Ditto
EX9K15M	1200-1240	_ 	200	ၕ	1	1030-1050	01	Ditto

Note. The magnets are aged for 5.8 hours in boiling water to stabilise their structure and magnetic properties.

Heat-treatment Schedules Recommended for Cast Permanent Magnets

Table 79

					ø
Grad	Grade of alloy	Hardening or	Cooling	Tempering	
New	PIO	temperature, °C	medium	temperature,	Comments
AH1 AH2	АЛНИІ	1200 1200	Boiling water Ditto	550	Normalising is applied to thin magnets
AH3	АЛНИЗ	_	Air		
AHK	АЛНИСи	_	Ditto	J	Slow cooling for thick magnets
AHKol	AJHMKo12	_	Ditto	J	0
AHK ₀ 2	АЛНИКо15	_	Ditto	1	
AHKo3	АЛНИК018	1300	Ditto	* 009	Cooling from 1300°C is carried out in a
					magnetic field of not less than 1,500
A 1 1 1.7	A 1111111	0001		000	oersteds, cooling rate not exceeding 5°C
AUN 04	WALHUKO MALHUKO	1900	01110	000	per second, down to 500° C*
	() / *** * * * * * * * * * * * * * * * *				

* The cooling rate is regulated, depending on volume and shape of the part, by wrapping it, prior to cooling, in asbestos paper.

Steel with high silicon content shows great tendency to decarburisation and, due to this, tools from grade 9XC and other similar steels, which are not subjected to grinding after heat-treatment, should be carefully protected against burning-out of carbon.

It is not recommended to cool the work within the furnace after

tempering (tables 83 and 84).

Table 80 Annealing Schedules Recommended for Corrosion-resistant and Non-scaling Steels

			Mechanical	properties
Grade of steel	Annealing tempe	or tempering rature, °C	Tensile strength, kg/mm², minimum	Elongation, %, minimum
0X13, 1X13	Ann.	860-880	40	21
	Temp.	740-800		
2X13	Ann.	860-880	50	20
	Temp.	740-800		
3X13	Ann.	860-880	5 0	15
	Temp.	740-800		
4X13	Ann.	860-880	56	15
	Temp.	740-800		
X 17, 0X 17T (ЭИ465)	Temp.	740-780	50	18
X25T, X28 (ЭИ439, ЭИ349)	Temp.	740-780	54	17
Х 15Н9Ю (ЭИ904)	Norm.	1040-1080	<110	20
X5	Norm.	850-870	40	24
4X9C2 (CX8)	Norm.	850-870	75	15
Х 6СЮ (ЭИ 428)	Temp.	750 - 800	45	20
1Х12СЮ (ЭИ404)	Norm.	800-850	50	15
	Norm.	800-850	50	20

Notes. 1. Abbreviations used: temp. - tempering; ann. - annealing; norm. normalising.
2. Either annealing or tempering is applied to steel grades 1X13 through

4X13.

Table 81

Hardness mumber, 51 - 5050-48 48-45 45-30 30-21 52-51 51-50 50-46 46-31 31-20 59-54 54-52 Mechanical properties Flonga-tion, % Hardening Schedules Recommended for Corrosion-resistant and Non-scaling Steels 1822 19989 9 Tensile strength, kg/mm² 8881 15888 83 1 Heating temperature, °C Tempering 1) 740-760 2) 550-600 375-500 Air, then from -30 Cooling medium Oil or air Oil or air to -70° C Ditto Ditto Ditto Ditto Ditto Hardening Heating temperature, °C 1000-1050 1000-1050 1000-1050 1000-1050 970-1040 1010-1050 1070-1130 925-975 1000-1050 1000 - 1050i 030-1070 4X10C2M (ЭИ107) 2X13H4Г9 (ЭИ100) X15H9Ю (ЭИ904) X17H7IO (ЭИ973) IX17H2 (3M268) Grade 0X13 1X13 2X13 3X13 4X13 9X 18

1 1

6

55

1

Air or water

1000 - 1150

0X 20H14C2 (3M732)

	I	1	1	1	1	1	1]	1	ł	1	1	1	ļ	í	1	1	ſ		1		!		ł	1]	1	I	!	I
_	35	20	20	20	32	32	40	40	35	40	40	45	40	40	40	40	40	40	25	35	According to specifications	1 40	2 2	G ?	35	17	32	35	32	32
_	09	9	09	65	20	65	20	52	20	45	48	20	28	22	20	25	22	22	65	20	ding to sp	1 20	2 6	9	දු	54	22	72	2	25
	(1	ı	I	1	ŀ	1	1	I	1	!	1	1	l	١	J	1	1	1	1	Ассол		1	1	Į	1	!	ĺ	l	1
_	Air or wafer	Air	Ditto	Ditto	Oil, water or air	Ditto	Ditto	Diffo	Ditto	Ditto	Ditto	Ditto	Ditto	Ditto	Ditto	Ditto	Ditto	Ditto	Ditto	Ditto	Ditto	21110	01110	Ditto	Ditto	Water or air	Water	Water or air	Water	Water or air
	1000-1150	950-1050	950-1050	950-1050	1100-1150	1000-1080	1050-1100	1050-1100	1050-1100	1050-1100	1050-1100	1050-1100	1050-1100	1050-1100	1050-1100	1050-1100	1050-1100	1050-1100	1100-1150	1100-1150	1100-1150	0011-0011	0011-0011	1100-1150	1100-1150	850-950	1050-1070	1050-1080	1050-1070	1050-1080
~	X 20H14C9 (3U211)	٠(۲	1X21H5T (ЭИ811)		Х23Н13 (ЭИЗ19)		-	X17H13M2T (ЭИ448)	` _ _	00X18H10 (3N842)	_	XIRHS	9X18H9	X18H9T	0X18H10T (9M914)	X18H10T	XI8HI2T	0X18H12E (3N402)	4X18H25C2	X23H18 (3N417)	≆`	UX23H28M3JJ31 (3N943)	∀,	X25H20C2 (3M283)	1X25H25TP (ЭИ8ІЗ)	X28AH * (3)(657)	X14F14H *	X17H13M3T * (ЭИ432)	X17AF14 * (3M213)	0X18H11 * (ЭH684)

* Data apply only to sheet steel; for other grades of steel, the heat treatment schedules and mechanical properties listed hold good for specimens made from heat treated blanks of quality steel.

Table 82

Approximate Heat-treatment Schedules and Hardness Numbers of Carbon Tool Steels

							•
	Annealing	aling		Hardening		Tempering	ering
Grade	Heating temperature.	Hardness number, HB, maximum	Heating temperature, °C	Cooling medium	Hardness number, R _C	Heating temperature, °C	Hardness number, R _C
y7, y7A	740-760	187	800-830	Water, then oil	61-63	160-200	63-60 60-54
						300-400 400-500 500-600	54-43 43-35 25 97
Y8, Y8A, Y8F, Y8FA	740-760	187	790-820	Ditto	62-64	160-200	64-60 60-55
; ;						300-400	55-45
	i	_				500-600	45-35 35-27
V9, V9A	740-760	192	780-810	Ditto	62-65	160-200 200-300	64-62 62-56
						300-400	56-46
	i i	ţ	0			500-600	37-28
\$ 10, \$ 10A	077-067	197	770-800	Ditto	62-65	160-200 200-300	64-62 62-56
						300-400	56-47
V11, V12,	750-770	207	760-790	Ditto	62-66	160-200	47-38 65-62
VIIA, VI2A						200-300	62-57
					_	300-400	57-49 49-38
V13, V13A	750-770	217	062-092	Ditto	62-66	160-200	65-62

High-speed steel. High-speed steel is annealed according to normal or isothermal procedures. In normal annealing the work is cooled at a rate of 30-40°C per hour. Isothermal annealing is conducted as follows: heating to 830-850°C, holding for 1.5-2 hours, rapid cooling to 720-750°C, holding for 3-5 hours, followed by air-cooling (Table 84). The steel is also high tempered at 720-750°C with subsequent aircooling, to lower its hardness.

High-speed steel possesses low heat conductivity and, because of this, the tools should be preheated to 800-850°C on hardening. Preheating may be omitted for regular-shaped and small-sized parts.

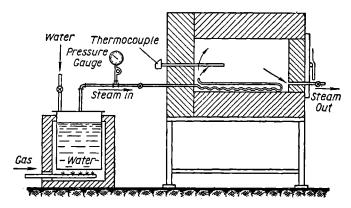


Fig. 23. Schematic view of a steam-treatment installation

The work is preheated either in special furnaces of by twofold or threefold dippings for 2-4 sec in the same bath wherein the final heating is carried out.

The work is hardened in oil, molten salt or air. Air-cooling results in low durability and can be recommended only for long thin tools with a view to diminish distortion (Table 85). Hardening in molten salts is effected at 250-500°C with subsequent air-cooling Tempering can be either single-stage with holding for 2-3 hours, or two-stage with 1-hour holding period for each stage.

Steam heat-treatment (tempering) of high-speed steel, recommended currently, consists of the following steps: 1) fully finished (including sharpening) tools are charged into an air-tight furnace, heated to 350-

370°C, after which steam is admitted into the furnace and the tools are held at the said temperature for 30 min; 2) the tools are then heated to 540-580°C (steam being introduced continuously) and held for 20-30 min at the above temperature. The work is first cooled in air down to 50-70°C, and then heated in oil to 40-50°C.

This treatment results in the formation of oxide film Fe₃O₄, dark blue in colour, which imparts appreciable corrosion resistance to the tools, as well as improves to some extent their service behaviour.

Table 83

*	Approximate F	feat-treatment	Approximate Heat-treatment Schedules and Hardness Numbers of Alloy Tool Steels	Hardness No	imbers of All	oy Tool Steel	s
	Anne	Annealing		Hardening		Tempering	ring
Grade	Heating temperature, °C	Hardness number, HB	Heating temperature, °C	Cooling medium	Hardness number, $R_{m{\mathcal{C}}}$	Heating temperature, °C	Hardness number, R_C
7X3, 8X3	800-820	229-187	820-860	lio	61-63	150-200 200-300 300-400 400-500 500-600	62-60 60-58 58-55 55-50 50-39
X6	780-800	217-179	820-850	Ditto	64-62	150-250	ſ
×	780-800	229-187	835-855	Ditto	65-62	150-200 200-300 300-400 400-500 500-550	64-61 61-55 55-49 49-41 41-35
60 X	780-800	229-179	825-850	Ditto	65-62	150-170 170-250 250-320	62-60 60-55 55-50
X 05	780-800	241-187	780-800 800-825	Water Oil	65-62 65-63	200-300 300-460 400-500	62-55 55-50 50-41

62-59 59-58 58-56 56-50 50-43	64-60	60-58 58-52 52-48	64-61 61-58 58-52 52-44 44-35	52 52-50 50-46 46-38 38-31	62-60 60-55 55-52 52-42 42-36
200-300 300-400 400-500 500-600	180-200	150-200 200-300 300-350	150-200 200-300 300-400 400-500 500-600	200-250 250-350 350-450 450-550 550-650	150-200 200-300 300-400 400-500 500-600
64-62	64-62 62-60	63-61 62-60	66-62	53-56	62 min
Oil or air jet	Water	Water Oil	110	Ditto	Ditto
950-1000	780-820 820-840	780-800 810-830	800 - 840	006-088	840-860
269-217	217-179	229-187	241-197	207 - 170	229-187
850-870	760-780	780-800	780-800	820-840	820-840
X12	Ð	Bl	XF	4XC	бХС

Table 83 continued

	Annealing	aling		Hardening		Tempering	ering
Hea	Heating temperature, oC	Hardness number, HB	Heating temperature, °C	Cooling medium	Hardness number, R_C	Heating temperature, c	Hardness number, R _C
790	790-810	241-197	850-880	lio	65-61	150-200 200-300 300-400 400-500 500-600	64-63 63-59 59-54 54-47 47-39
<u>8</u> 2	780-810	255-207	820-840	Ditto	62 тіп	1	1
908	800-820	207-170	800-850	Water	64 min	200-220	64-62
11	770-780	207-170	840-860	011	62-63	200-400	60-50
	1	1	820-840	Ditto	61-63	300-325	51-56
	800-820	217-179	900-098	Ditto	56-53	200-300 300-400 400-500 500-600	53-51 51-49 49-42 42-33
8	800-820	255-207	006-098	Ditto	56-54	150-200 200-300 300-400 400-450	54-52 52-48 48-42 42-36
	_	- -					

58-53 53-49 49-43	55-52 52-47 47-43 43-35	62-60 60-56	63-62 62-58 58-52 52-46 46-37	52-49 49-48 48-46 46-45 45-48 48-40	48-40	64-60 60-53 53-48 48-40
200-300 300-400 400-500 500-600	200-300 300-400 400-500 500-600	170-230 230-275	150-200 200-300 300-400 400-500 500-600	150-200 200-300 300-400 400-500 500-575 575-700	200-600	200-300 300-400 400-500 500-600
60-54	58-56	64-62	65-63	52-49	54-51	67-64
011	Ditto	Ditto	Ditto	Oil or air jet	Ditto	Water Oil
006-098	830-860	800-830	820-850	1075-1125	1025-1075	800-820 820-860
285-229	217-179	241-197	255-207	255-207	255-207	285-229
780-800	760-780	780-800	780 - 800	860-880	820-840	780-800
6XB2C	5XBF	9XBF	XBL	3X2B8	4X8B2	XB5

Table 83 continued

	Anne	Annealing		Hardening		Tempering	ering
Grade	Heating temperature,	Hardness number, HB	Heating temperature, C	Cooling medium	Hardness number, $R_{\mathcal{C}}$	Heating temperature,	Hardness number, $R_{\mathcal{C}}$
X12M	850-870	255-207	1000-1050	Oil or air jet	65-62	150-200 200-300 300-400 400-500 500-600	63-62 62-59 59-57 57-55 55-47
			1115-1130	Ditto	48-45	500-520 repeated 3-5 times	59-62
5X H M	790-820	241-197	820-860	li0	60-58	150-200 200-300 300-400 400-500 500-600	60-58 58-53 53-48 48-43 43-35
5XI'M	790-810	241-197	820-850	Ditto	58-53	200-300 300-400 400-500 500-600	57-52 52-46 46-40 40-34
• -							

47-40 40-33	63-59 59-57	62-61	46-40 40-34	41 - 35	47 - 41 41 - 34	46-41	46-41		
400-500	100-200 200-400	520-550 repeated 3-5 times	400-500	200-600	400-500	440-460	440-460		
58-53	64-62		59-55	59-55	59-55	59-55	1		
lio	Oil or air jet	Ditto	0:11	Ditto	Ditto	Ditto	Ditto		
830-860 with air- blast cooling to 720-760	1040-1080	1115-1150	850-870	850-870	840-860	063-028	870-890	_	
235-192	255-207		255-207	255-207	241-197	241 тах	241 max		
790-820	850-870		800-820	810-830	790-820	850-870	850-870		
5XHT	Х12Ф1		5XHC	5X HCB	5ХНВ	45XHT *	45XHB *		

* Data refer to cast die cubes.

To ensure adequate finish, the work is treated with superheated steam; to do this, steam is superheated in a tube, situated inside the furnace, prior to being admitted to the muffle.

Any type of boiler, of sufficient capacity, heated by any fuel can serve as a source of steam. Diagram of a steam-treatment installation

is presented in Fig. 23.

Recommended steel grades, tempering temperatures, and fool hardness numbers are presented in Table 86.

Table 84

Technological Schedules for Isothermal Annealing of Tool Steels

	Initial	heating	Isotherm	al holding	Hardness
Grade	Tempera- ture, °C	Holding, hrs	Tempera- ture, °C	Holding,	number,
X12 X12M X, ШX15, ШX12 9XC XΓ XBΓ B1 3X2B8 У11, У11A, У12, V12A У10, У10A У9, У9A P18 P9	\$850-870 770-790 780-810 790-810 770-790 750-770 750-770 750-770 \$830-850	1.5-2.5 1.5-2.5 1.5-2.5 1.5-2.5 1.5-2.5 1.5-2.5 1.5-2.5 1.5-2.5 1.5-2.5	720-750 670-720 680-720 700-730 700-730 670-700 710-740 640-680 620-660 600-650 720-750	3-5 3-4 3-4 3-4 3-4 3-4 1-2 1-2 1-2	288-255 217-255 197-228 207-241 217-255 197-228 187-228 187-228 187-207 179-197 170-187 217-255 228-255
P18K5, P18K10	850-870	1-2	730-750	4-6	217-269
		i			

In Table 86 hardening temperatures and cooling media are not indicated. These data are given in tables 70, 72, 81, 82, 84. Hardness numbers of tools are given in accordance with the Soviet Standards, while those of machine parts—according to specialised handbooks.

Table 85 Approximate Heat-treatment Schedules for High-speed Steels

	Annea	ling	Haı	dening		Tempe	ring
Grade	Heating tempera- ture, °C	Hardness number, H _B	Heating tempera- ture, °C	Cooling medium	Hardness number, R _C	Heating tempera- ture, °C	Hardness number, RC
			1000 1050	_		[
P9	830-860	207-255	1230-1250	icy ent	61-63	540-58 0	≥ 62
Р9Ф5	840-860	293	1240 - 1260 1220 - 1250	air. For endency ibsequent	62-64	575-585	64-66
Р10К5Ф5	840-860	293	1230-1250 1210-1240	0°C, a g a te th sub	62-64	575-585	65-67
P9K10	840-860	293	1230-1250	vin wi	62-64	575-585	65-66
P14Ф4	840-860	293	1250-1270	25 - 50 C	62-64	575-585	65-66
P18	830-860	207-255	1280-1300 1250-1290	alta e too at 2 in o	62-64	540-580	≥62
Р18Ф2	840-860	293	1280-1300 1260-1290	ricat ricat oil down	63-65	575-585	65-66
Р18К5Ф2	840-860	293	1280-1300 1260-1290	int int ack	63-65	575-585	64-66
P18K5 P18K10}	See Table 84	!	1280-1320	Oil, very to cr cooli	60-64	550-570	62-65
		1	1	1	1	İ	l

Notes. 1. On annealing the cooling rate is 20-25°C per hr.

Notes. 1. On annealing the cooling rate is 20-25°C per fir.

2. The hardening temperatures given as numerators refer to cutters and those given as denominators—to fancy-shaped tools; the lower temperatures apply to thin tools, whereas the upper range—to the larger tools.

3. Tempering procedures are repeated from 2 to 4 times for tools made of steel grades P905, P10 K505 and P1404, and 2-3 times for the remaining grades, the holding time lasting 1 hour in each case.

2. PRACTICAL HINTS ON HEAT-TREATMENT OF ARTICLES

Cutting tools. Long, slender tools should be suspended on heating. When using chamber furnaces, the tools mentioned above should be placed on special supports (Fig. 24) which ensure rectilinearity of the tool on heating. Welded tools should be heated below the welding seam. When heating large tools in electrode-salt baths, care should be taken to prevent contact between the tools and the electrodes so as to avoid sticking.

Upon air-cooling, tools should not be placed on the floor or plates, but suspended or driven into sand boxes by their shanks.

Toble 36

Toble 36

Toble 36

Temperatures and Hardness Numbers

Tune of too	Recommended	Temperature for tempering	Hardness number, R_C	er, R _C
		the working section	Working section a	Shank b
		Cutting Tools	s ₁	
II Tool with a welded bit	P18, P9	540-580	61-65	
	P18, P9, 3/1347 V11A, V12A, XBF 9XC	540-580 150-180 180-200	62-65 61-64 61-64	
Circular tool				
	9XBF, X, V10A.	540-580	62-65	
	VIIA 9XC	150-180 180-200	61-64 61-64	
Grooving cutter				

	30.45 30.45 30.45	30-45
Below 1 mm 60-63 Above 1 mm 61-64 Below 1 mm 58-62 Above 1 mm 60-64	62-65 61-64 61-64	Below 5 mm 60-64 Above 5 mm 62-65 Below 10 mm 59-63. Above 10 mm 61-64
540-580 180-240) 200-250 }	540-580 150-180 180-200	540-580 150-220 180-240
P18, P9, ЭИ347 У12A, X12M 9XC	P18, P9, ЭИЗ47 У10A, У11A, X, XF 9XC	V10A, V11A, V10A, V11A, V12A XBT, IIX12, X, 9XC
Slitting cutter	End milling cutter	Twist drill

Table 86 continued

			י ביינים	י בפונ פס פסונים ומכת
1 - 1 - 1 - 1 - 1 - 1 - 1 - 1 - 1 - 1 -	Temperature Recommended steel for tempering	Temperature for tempering	Hardness number, R_C	er, R_C
1ype of tool	grades	the working section	Working section a	Shank b
	P9, 3N347	540-580	Below 6 mm 61-63	30-45
	<i>y</i> 10A, <i>y</i> 11A, <i>y</i> 12A, X, XBF	150-220	Above 6 mm 62-65 Below 8 mm 59-63 Above 8 mm 60-64	30-45
Reamer. Counterbore	9XC	180-240		
	Р18, Р9, ЭИ347	540-580	Вејом 6 шт 61-63	Below 4 mm
9	У10A-У12A, Ф X XГ XВГ	160-220	Above 6 mm 62-65 59-62	From 4 to 8 mm 35-55
Тар	9XBr, IIIX12 9XC	200-270		Above 8 mm 30-45
	y 10A-Y12A, X,	180-220	58-62	
	AI, 9ABI, W	220-280	.	
Round threading die				

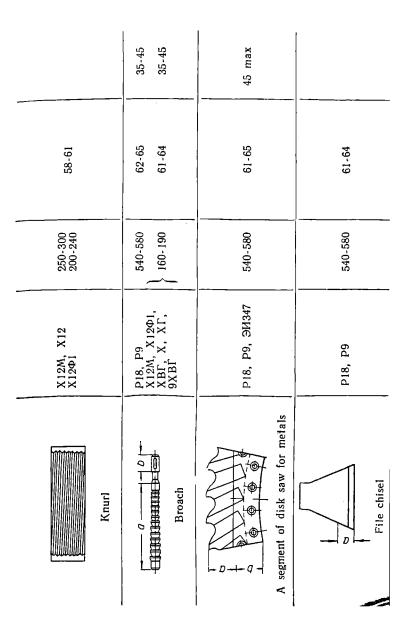


Table 86 continued

ļ			1	1
	r, RC	Sildlik	35 тах	40-45
		working section a	Should stick to a plate with hard-ness number above $R_C = 54$ Files for sharpening saws should stick to a plate whose hardness is above $R_C = 57$	58-61 58-61
	Temperature for tempering the working	section		180-220 220-250
	Recommended steel grades		V7.V13, ШХ6, ШХ9, ШХ12	y8-y12 9XC
	Type of tool		File	Hack saw blade

		-	Lower by 50 100 H _B as compared to the working section
	56-64		40-43 35-44 33-41 35-41 30-34 30-34 31-38
· sp	150-180	Dies	400-430 500-600 500-600 488-550 460-600 560-600 530-600
Measuring Tools	V10A-V12A, XF, XBF, XO9, IIIX15, 10, 15, 20, 15X, 20X, 15XF	Hot Stamping Dies	97A, 98A 5XHM 5XFM 5XHT 7X3 30XC 4XC, 35XFC
•	External gauges Internal gauges		Hot stamping die

Table 86 continued

Fitter's Tools

32-40	32-40	32-40	
52-57	52-57	48-54	42-50
250-320	250-320	300-360	300-380
y7, y8	<i>y</i> 7, <i>y</i> 8	<i>y</i> 7, <i>y</i> 8	45, 50
Chixel	Centre punch	Rivet set	Pliers

Table 86 continued

				manufacture of other
Type of fool	Recommended	Temperature for tempering	Hardness number. RC	r. R _C
TOO TO OAK		the working section	Working section a	Shank b
Hand vice	45, 50	300-380	42-50	
Combination pliers and cutting nippers	y7, y8	200-320	52-60	

40-45			
54-58	46-52	48-54	49-56
250-300	280-350 320-370	320-380 370-420 \ 400-450 }	270-350
<i>y</i> 7, <i>y</i> 8	50, 60 37, 3/8	Cr. 3, 15, 20 40, 50 40X	y7, y8
Stamp	Screwdriver	Spanner	Fitter's hammer

Table 86 continued

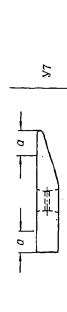
er, R _C	Shank b				30-40
Hardness number, R _C	Working section a		49-56	48-52	50-56
Temperature for tempering	the working section	Toots	270-350	200-250	270-340
	steel grades	Blacksmith's Tools	7.7	50	<i>y</i> 7
	Type of tool		Sledge hammer	Anvil chisel	Blacksmith's chisel

. 30-40			40-50
50-56	50-56		56-59
270-340	270-340	rools	230-270
27	7.8	Pneumatic Tools	y7, y8, 4XB2C
Blacksmith's punch	Blacksmith's flatter		Pneumatic chisel Pneumatic punch

Table 86 continued

Table 86 continued	r, R _C	Shank b	40-50	
Table 8	Hardness number, R _C	Working section a	53-56	
i i	Temperature for tempering	the working section	270-300	
	Recommended		<i>y</i> 7, <i>y</i> 8	
	Type of tool		Pneumatic rivet set	Drift

Coppersmith's and Tinsmith's Tools



Roofing hammer

45-50

330-380

		-	· ·
45-50	52-60		40-42
330-380	200-320	Fools	450-500
37	2.8	Woodworking Tools	85XΦ, 65X, IIIX6, Y8A
Beak iron	Hand shears		Circular saw

Table 86 continued

mber, R _C	Shank b		
Hardness number, R _C	Working section a	56-58	56-58
Temperature for tempering	the working section	250-275 240-280	250-275
Becommended	steel grades	V8-V10, 85XФ, 65X 6XB2C, XГ	y8, y10 85XΦ, 65X, 6XB2C, XΓ
	Type of tool	Twist drill End milling cutter	Gang milling cutter

		40-45
56-59	45-50	56-58
230- <i>2</i> 75 220-280	330-375	250-275
У7, У8, У9 85ХФ, 65Х, 6XB2С	У7, У8, У9 85ХФ, 65Х	y8, y9, 65X
Planing cutter	Scoring knife	Solid lathe gouge

Table 86 continued

		Temperature for tempering	Hardness number, R_C	r, R _C
Type of tool	steel grades	the working section	Working section a	Shank b
Chip axe	<i>y</i> 7, <i>y</i> 8, <i>y</i> 10	260-340	50-56	
$\begin{array}{c c} & -a & -b & - & - & - & - & - & - & - & - & $	y8, y9, 65X	320-350	50-52	40-45
Planing bit	V8, V9	200-320	53-60	

				40-45
53-58	44-48	43-50		58-62
250-320	360-420	330-400	tipping Tools	200-250
У8, У9, 65X	<i>y</i> 7. <i>y</i> 8	77	Parts of Fixtures and Gripping Tools.	V8, V10, 65T
Chisel	d	Joiner's tongs	Part	Collet

Table 86 continued

			י ממוני	ו ממוג מם כמונונותבם
Type of tool		Temperature for tempering	Hardness number, R_C	r, R _C
soot to add t	steel grades	the working section	Working section a	Shank b
Chuck jaw	45 40 X	300 - 350 350 - 400	45-50 45-50	35-42 35-42
Centre	y7-y10	250-300	55-58	55-58
	Track Tools	<i>S1</i>		
Spike hammer	Cr.6, Cr.7	300-350	45-50	

35-40	45-50	47-50	49-56
200-300	300-350	300-330	200 - 300
	Cr.6, Cr.7	Cr.6, Cr.7	Cr.6, Cr.7
Socket wrench	Tamping pick	End pick	Sleeper axe

Table 86 continued

Time of tool	Recommended	Temperature for tempering	Hardness number, R_C	r. R _C
ion o add		the working section	Working section a	Shank b
Pickaxe	C r.5	200-250	38-42	
	Harne somaker's Tools	Tools		
Saddler's knife	65, 70, 97, 98	320-360	49-53	
Cutter's knife	65, 70, 97, 98	280-320	53-56	

33-42	-		
40-46		42-52 45-50	43-51 48-52
350-420	ools .	300-400	350-420
45, 50	Bricklaying Tools	Y7-V10 Cr.5	50.70 Cr.5-Cr.7
Saddler's hammer		Blocking chisels, Planishing hammers, Trowels Plumbs Brick hammer, brick	axe, pickaxe Pointed and flat crowbars

Notes. 1. Cylindrical shanks of drills, reamers and cutters are not subjected to hardening.

2. Pneumatic chiesls such as cape chiesls, caulking chisels and file chiesls are treated in the same manner as pneumatic chiesls, while dollies and strikers are treated in the same fashion as pneumatic rivet sets.

3. Tinner's fools such as formers, embossing dies, scrapers, etc., are treated in the same manner as beak irons.

Small-sized cylinder-shaped tools, up to 6-8 mm in diameter, are hardened by the use of a smoothing iron (employed as a chill bar)

(Fig. 25).

Thin, slender tools made from high-speed steel are hardened between plates, which, when not positively cooled on the inside, should be greated prior to the hardening procedure.

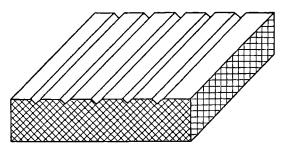


Fig. 24. Brick with grooves for heating of cylinder-shaped parts in horizontal position

Tools from grade 9XC and other silicon steels, which are not ground, should be carefully protected against decarburisation.

Tools hardened in water should be immersed therein very quickly so as to avoid cracking in that part of the tool which is dipped first.

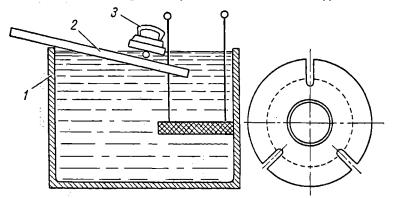


Fig. 25. Schematic view of "fron" hardening: 1 - bath; 2 - plate 3 - iron

Fig. 26. Cutter from carbon steel, slotted prior to hardening

Round cutters from carbon steel should be slotted by a 1-mm cutter to prevent cracking on hardening (Fig. 26).

Tools wish sections varying step by step should be quenched with the thick end down. Articles with blind holes should be cooled with the orifice oriented upwards or in a special installation with the aid of a water jet.

Tools with the cutting ends, when heated in salt baths, are heat treated as follows: for carbon and alloy steels—first the shank, then the cutting edge, while for high-speed steel the sequence is

reversed.

Prior to re-hardening the articles should be annealed, this being of utmost importance for high-speed steel articles in order to prevent coarse-grained flake-like fracturing.

When testing files for hardness by the use of a hardened steel plate, it may occur that the file slides on one side and sticks on the other,

i. e., one side is soft, the other hard.

In most cases, this failure occurs because of excessive hardness of

the pad on which the file is cut.

File dents when placed on the hard pad are dulled if the file is struck on the opposite side. If the pad is made of a material which takes on cold working (aluminium, etc.), it should be periodically annealed (the periodicity being set from practice). One of the Soviet research institutes recommends grade MT1 magnesium, which takes no cold working, as a pad material. An alloy consisting of 87% lead and 13% antimony is also used.

Annular cracks in welded tools. Cracks, situated at about 1 to 3 mm from the seam, can often be observed in high-speed welded tools.

These cracks are caused by rapid cooling of the tools from the welding temperatures and the subsequent air hardening of the seam area of the high-speed steel.

To prevent cracks, the blanks, after welding, should be charged into a furnace heated to 740-760°C. After charging, the furnace is cooled to 500-600°C and the blanks are then discharged.

Hot stamping dies. On hardening, small- and medium-sized dies can be charged into a furnace heated to requisite temperature, while large dies should be introduced into a furnace heated to 750-700°C.

When no spraying is available in the quenching tank, the die should be immersed with the face up. Dies from carbon steel are preliminarily normalised to increase the depth of hardening. For forging hammer dies the hardness of the shank should be by 50-100 $\,H_B$ less than that of the working section. The shanks are usually tempered through holding on the plates or at the opened furnace doors.

Dies should be tempered while still hot, immediately upon withdrawal from the quenching tank. When high tempering is applied, it is not recommended to place the dies immediately in the furnace heated

to tempering temperatures.

Pressure casting dies. The most durable material for pressure casting dies is grade 3X2B8 steel. It endures well the tempering at 600°C. Die performance shows that the higher the die surface hardness, the better is its wear resistance and the less the liquid metal sticks to

its sides. The best means of die case-hardening should be recognised

nitriding and cyaniding, as they cause minimum distortion.

Technology of casting die manufacture from grade 3×28 steel, which is cyanided, is as follows: (1) preliminary machining of the blank; (2) heat-treatment—hardening from $1075-1125^{\circ}$ C and tempering at $700-720^{\circ}$ C to 30-34 R_C ; (3) final machining; (4) gas, liquid or solid cyaniding (at $560-580^{\circ}$ C to a depth of about 0.08-0.10 mm); washing in kerosene and wiping dry. The hardness of cyanided die is above 60 R_C . Stresses appearing in the die during the operations, and casting cracks can be prevented by giving the dies periodical temperings after predetermined intervals of operation.

Cold stamping dies. Dies hardened in water should be protected against cracks at fastening apertures (Fig. 27) by 2-3-fold water dipping of the areas mentioned. To avoid warping when treating punches from

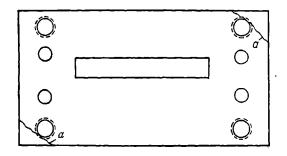


Fig. 27. Hardening cracks through auxiliar orifices of a $die-a_0$ cracks

carbon steel, the shanks should be cooled first, and then the whole article immersed with the face oriented upwards.

The hardness of the perforating punches should gradually shade off from R_C =54-58 on the working end to R_C =40-44 on the shank.

Measuring tools. While hardening measuring tools it should be borne in mind that if correct technology is not strictly followed, this leads to greater distortion and warping. Improving of gauges after roughing, as well as hardening in hot media, yields good results and

brings down distortion.

Measuring tools are tempered twice: the first time, after hardening, to remove internal stresses, and, the second time, after grinding, to relieve post-grinding stresses and to prevent distortion during storage (artificial ageing). The ageing temperature should not exceed that for the first tempering procedure. The duration of ageing is from 2 to 5 hrs depending on the size of the tool, from the moment the latter has been heated through. Tools of long, slender form should be hooked up on heating.

The following data may be used to determine the carburising depth in relation to tool thickness:

Tool thickness, mm 3-5 5-10 10-15 Up to 20 and over Carburising depth, mm 0.3-0.5 0.5-0.8 0.6-0.9 0.8-1.0

Woodworking tools. Disk saws are oil-hardened while being placed vertically. As soon as the saw cools down to about 400-300°C, it is extracted from the tank and clamped between two plates till it cools down entirely. The saws are tempered when gripped in screw clamps. It is recommended to tighten screw clamps to arrive at an effective straightening of the saw on tempering.

The gradual transition from the thin section to the thick one in lathe gouges, as well as the sides next to the aperture in a hollow gouge should

be hardened so as to obtain minimum hardness.

Fitter's tools. Built-up tools such as flat- and round-nosed pliers, nippers and hand vice should be assembled and hardened with the jaws

open. The jointed sections should not be hardened.

To prevent cracking, cutters and combination pliers are quenched in kerosene or oil, while providing effective mixing of the bath. Large combination pliers should first be quenched in water and then transferred to oil. Hand shears are to be quenched in a disassembled state. The opening for the rivet should neither be heated nor hardened. The work is to be cooled in kerosene or oil.

When only the ends of carbon-steel tools (chisels, hammers, etc.) heated throughout are to be hardened, these ends are quenched by

the alternating method.

In this hardening method, first, the thin part is quenched, then the thick one, or vice versa. When hardening the thin part first, it should be immersed to a greater depth than that required; if the thick part is hardened first, the work should be heated up to the upper temperature range.

Spanners, according to Soviet Standard FOCT 2838-54, can also be manufactured from carburised steels, the depth of the case being 0.3-0.5 mm for spanners 2.5 to 4 mm thick and 0.6-1.0 mm for those

from 5 to 8 mm thick.

Heat-treatment of springs. When heat-treating springs, effective steps should be taken to prevent cracking, especially of those of the

smaller dimensions.

Holding in the furnace should be as short as possible. Small springs should be placed onto a heated pan. Springs from wire over 6 mm in diameter should be tempered at 670-700°C prior to hardening. Large spiral springs should be heated, on hardening and tempering, in special devices to prevent distortion. Prior to tempering, the springs are to be degreased. Best tempering results are obtained in saltpetre baths, holding time being from 10 to 12 minutes. When electric or flame furnaces are employed, tempering should take 20-40 min.

Brittleness, which develops in springs as a result of pickling or application of an anticorrosive coating (pickle brittleness), is eliminated

by heating for 1-2 hrs at 150-180°C.

When the above treatment does not remove the increased brittleness, the springs should be annealed, and then the entire heat-treat-

ment procedure repeated.

The thermo-mould method is the best means to manufacture flat springs from a heat-treated band. The spring band sheared to requisite profile is clamped in a device, shaped to finished spring, then heated for 20-30 min at 350-400°C, after which it is air-cooled together with the device. When the spring, upon discharge, is not up to the required profile, the configuration of the device must be corrected by the trial method. The said clamping devices are of the multiple-recess type.

Repairing worn plug gauges. Worn small-sized plug gauges from alloy and high-carbon steels can be repaired through tempering in an oil bath at 210-230°C. As a result of this treatment, the diameters of the gauges grow because of the disintegration of residual austenite

(Table 87).

Table 87

Heat-treatment Schedule for Repairing Thread Plug Gauges from Alloy and High-carbon Steels

Plug diameter, mm	Heating temperature, °C	Holding time, hrs
Below 25 25-35 35-40 40-60 60-80 80-150	225 - 235 225 - 235 215 - 225 205 - 215 205 - 215 200 - 210	} 2-5

Straightening of hardened articles. When hardened and tempered articles warp beyond the tolerated allowances, they are to be sub-

jected to straightening.

Tools from any grade of tool steel up to 15 mm in diameter or thickness, hardness numbers being $R_{\rm C} = 50$ and above, are straightened through brief, but not heavy hammering over the entire length on the concave side except for the cutting edges and threads, until full straightening is achieved. The tool should be laid on a hardened plate, while straightening is carried out. The configuration of the hammer striker should not be pointed in order to avoid indentations which cannot be removed by subsequent grinding.

Tools from high-speed steel are hand-straightened while hot, immediately after hardening; those larger than 15 mm in diameter are straightened under a press, being heated to 500-550°C. When trueing a tool the latter is bent in the direction opposite to that of the detected convexity, to a degree slightly in excess of that of the true shape; after a predetermined holding (determined from practice)

and cooling, the work is discharged and inspected. If the procedure has not given satisfactory results, it is repeated. The heating temperature (500-550°C) is checked by the thermocolours.

Articles whose hardness numbers do not surpass $R_C = 50$ are press-straightened while either cold, or preheated to tempering temperatures.

Files from high-carbon steel are effectively straightened by the use of a mallet or a special device, the hardening procedure being discontinued at 140-180°C.

Half-round files, prior to hardening, are curved in the direction

opposite to the flat face.

Flat, long articles are removed from oil at 150-200°C and inserted for straightening into special slotted plates (the slots being thicker than the articles by 0.1-0.15 mm), and re-cooled therein.

All articles without exception are re-tempered to remove stresses im-

mediately after press or mallet cold straightening.

Normalising, straightening and repeated hardening with the use of all means to prevent warpage should be carried out in the following instances:

1. When warpage is excessive, the shape of the article prohibiting trueing (cutting edges or through threads, sharp changes in section,

etc.).

2. When the article does not yield to trueing.

3. When large-sized carbon or alloy steel tools display excessive

Thermal straightening (thermo-trueing). Thermal straightening can be used whenever there is a need to straighten warped plates (Fig. 28).

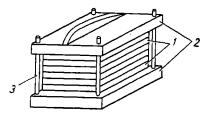


Fig. 28 Arrangement of plates in thermal straightening: 1-plates; 2-clamps: 3-bracing bolts

A packet of plates is clamped in a device and is held till heat-soaked in a furnace, the temperature being in the 240-280°C range for aluminium plates, and 400-600°C for those made of brass. The upper temperature range is used for straightening thicker plates.

A list of steel grades recommended for the manufacture of parts of lathes, automobiles and agricultural machinery, as well as tempering schedules and requisite hardness numbers, are given in Tables 88, 89 and 90.

Table 88

Steel Grades Recommended for the Manufacture of Spare Parts of Lathes and Presses, Their Tempering Temperatures and Hardness Numbers

Hardness numbers	Brinell				228-260 200-260 200-260 228-280	228-250 228-250	179-207
Hardness	Rockwell		56-62 56-62 56-62 56-62 56-62	45-50 45-50			
Tempering	recommended temperature, °C		180-200 180-200 180-200 180-200 180-200	320-400 320-400	620-670 620-670 620-670 600-650	550-620 550-600	Subject only to normalising
Steel	recommended	Gears	20X 15 20 12X H3 13X HBA	40X 45X	40X 35X 38XA 45X	45 40	45 50
	Name of parts and working conditions		Gears working at high peripheral velocities (4 m/sec and over), considerable bending and impact loads (gearbox gears of turning, furret and other lathes, as well as various other assemblies)	Gears working at medium peripheral velocities (2-4 m/sec), medium specific and light impact loads (individual gearbox gears of high-speed lathes, gearboxes, etc.)	Gears working at low peripheral velocities (up to 2 m/sec) and medium specific loads (gearbox gears, countershaft gearing, pump gears, etc.)	Gears working at low peripheral velocities (up to 1 m/sec) and medium specific loads (link and apron gears, etc.)	Non-critical gears working at low peripheral velocities (up to 0.3 m/sec) and low specific loads (changeable gears, quadrant gears, etc.)

Spindles

				228-260 200-260 200-260 228-280	228-250
56-62 56-62 56-62 56-62 56-62	45-50 45-50 ional	40-45 40-45	35-42 35-42 35-42 35-42 35-42		
180-200 180-200 180-200 180-200 180-200	320-400 320-400 Local hardening is optional	350-400 350-400	450-500 400-450 420-470 530-600 450-500	620-670 620-670 620-670 600-650	580-620 550-600
13XHBA 20X 15 20 12XH3	40X 45X Local	45 50	40X 35X 38XA XT XT 45X	40X 35X 38XA 45X	45
Heavy-duty spindles working in rolling and sliding friction bearings (spindles of turret and automatic lathes, etc.)	Spindles requiring high hardness (working in rolling friction bearings) as well as those working at medium loads in sliding friction bearings (tapered spindles of drilling, milling and other lathes)	Spindles working under light and medium load conditions in sliding friction bearings (spindles of turning, drilling and other lathes)	Spindles (working in rolling friction bearings), requiring increased hardness and adequate fatigue endurance (spindles of turning and gang lathes, etc.)	Spindles (working in rolling friction bearings, under medium loads), requiring increased strength	Light-duty spindles working in rolling friction bearings

Table 88 continued

Shafts working at high peripheral velocities of gearboxes, feed gearboxes, etc.) and and plain shafts. Norking in rolling friction bearings at edgearboxes, etc.) and plain shafts working in rolling friction bearings at peratus, character, degarboxes, etc.) and gearboxes, etc.) and plain shafts working in rolling friction bearings at peratus, character, degarboxes, etc.) and working in rolling friction bearings at peradequately high strength and wear resistance shafts working in rolling friction bearings at perature, downwing in rolling friction bearings at light and wear resistance splined shafts working in rolling friction bearings at light perquiring and toughness and toughness and toughness.					
recommended temperature, °C Rockwell Shafts		Steel		Hardness	numbers
Shafts 180-200 56-62 20 180-200 56-62 20 180-200 56-62 20 320-400 45-50 40 45-50 45-50 45 350-400 40-45 50 350-400 40-45 50 350-400 40-45 40 45-50 35-42 38 40-45 35-42 40 45 40-45 38 420-470 35-42 45 450-500 35-42 45 620-670 35-42 45 620-670 35-42 45 550-600 56-62 40 550-600 56-62 20 180-200 56-62 20 180-200 56-62 20 180-200 56-62	Name of parts and working conditions	recommended		Rockwell	Brinell
20X 180-200 56-62 15 180-200 56-62 20 180-200 56-62 40X 320-400 45-50 45 50 320-400 46-50 45 50 350-400 40-45 40X 450-500 35-42 35X 400-45 35-42 40X 450-500 35-42 45X 620-670 35-42 35X 620-670 35-42 45X 620-670 35-42 45X 620-670 35-42 45X 620-670 35-42 45 550-600 56-62 40 550-600 56-62 180-200 56-62 20 180-200 56-62 20 180-200 56-62		Shafts			
15 180-200 56-62 20 180-200 56-62 40X 320-400 45-50 45 320-400 40-45 50 350-400 40-45 50 350-400 40-45 35X 400-450 35-42 35X 400-450 35-42 40X 450-500 35-42 35X 620-670 35-42 40X 620-670 35-42 35X 620-670 35-42 45X 620-670 35-42 45X 620-670 35-42 45X 600-650 56-62 40 550-600 56-62 180-200 56-62 20 180-200 56-62 20 180-200 56-62	ifts working at high peripheral velocities	20X	180-200	56-62	
20 180-200 56-62 40 X 320-400 45-50 45 S 350-400 40-45 50 350-400 40-45 50 350-400 40-45 40 X 450-500 35-42 38 X A 450-500 35-42 40 X 620-670 35-42 40 X 620-670 35-42 35 X 620-670 35-42 45 X 620-670 35-42 45 X 600-650 35-42 45 X 600-650 35-42 45 Senebroom 550-600 56-62 15 Iso-200 56-62 20 Iso-200 56-62 20 Iso-200 56-62	3 m/sec and over) and increased bending loads	15	180-200	56-62	
40 X 320-400 45-50 45 X 320-400 45-50 50 350-400 40-45 50 350-400 40-45 40 X 450-500 35-42 35 X 400-450 35-42 40 X 450-500 35-42 40 X 620-670 35-42 45 X 620-670 35-42 45 X 620-670 35-42 45 X 600-650 56-62 40 550-600 56-62 56-62 15 180-200 56-62 56-62 20 180-200 56-62 56-62	gearbox shafts)	20	180-200	56-62	
45 X 320-400 45-50 45 50 350-400 40-45 50 350-400 40-45 40 X 450-500 35-42 35 X 400-450 35-42 40 X 450-500 35-42 45 X 450-500 35-42 45 X 620-670 35-42 35 X 620-670 35-42 45 X 600-650 56-62 45 550-600 56-62 15 180-200 56-62 20 X 180-200 56-62 20 56-62 56-62	ifts working in sliding friction bearings at pe-	40X	320-400	45-50	
45 350-400 40-45 50 350-400 40-45 40 X 450-500 35-42 35 X 400-450 35-42 45 X 420-470 35-42 45 X 620-670 35-42 35 X 620-670 35-42 35 X 620-670 35-42 35 X 620-670 35-42 45 X 620-670 35-42 45 580-600 56-62 40 550-600 180-200 56-62 20 180-200 56-62 20 180-200 56-62 20 180-200 56-62	ipheral velocities up to 3 m/sec (feed gearbox hafts, countershafts, etc.)	45X	320-400	45-50	
50 350-400 40-45 40X 450-500 35-42 35X 400-450 35-42 38XA 420-470 35-42 45X 450-500 35-42 40X 620-670 35-42 35X 620-670 35-42 35X 620-670 35-42 45X 620-670 56-62 45 580-600 56-62 40 550-600 56-62 20X 180-200 56-62 20 180-200 56-62 20 56-62 56-62	ifts working in sliding friction bearings at pe-	45	350-400	40-45	
40 X 450-500 35-42 35 X 400-450 35-42 45 X 420-470 35-42 45 X 450-500 35-42 45 X 620-670 35-42 35 X 620-670 620-670 45 X 600-650 600-650 45 560-600 56-62 40 550-600 56-62 180-200 56-62 20 X 180-200 56-62 20 X 180-200 56-62	ipheral velocities up to 2 m/sec (shafts of hange gearboxes, etc.)	20	350-400	40-45	
35.X 38.XA 400-450 38.42 45.X 450-470 35.42 40.X 620-670 35.X 620-670 45.X 600-650 45 45 560-620 45 180-200 15 180-200 56-62 20 180-200 56-62	avy-duty splined and plain shafts, requiring	40X	450-500	35-42	
38.XA 420-470 35-42 45.X 450-500 35-42 40.X 620-670 35.X 620-670 45.X 600-650 45 580-620 40 550-600 1shings 180-200 56-62 15 180-200 56-62 20 180-200 56-62	dequately high strength and wear resistance	35X	400-450	35-42	
45 X 450 - 500 35 - 42 40 X 620 - 670 35 - 42 35 X 620 - 670 620 - 670 38 X A 620 - 670 620 - 670 45 580 - 620 620 - 670 620 - 670 45 580 - 620 620 - 620 620 - 622 180 - 200 56 - 62 62 - 62 20 180 - 200 56 - 62 62 - 62		38XA	420-470	35-42	
40 X 620-670 35 X 620-670 38 XA 620-670 45 580-620 550-600 40 550-600 56-62 180-200 56-62 20 X 180-200 56-62 20 X 180-200 56-62 20 X 180-200 56-62		45 X	450-500	35-42	
38.X.A 620-670 45.X 600-650 45 580-620 40 550-600 1shings 180-200 56-62 20 180-200 56-62 20 180-200 56-62	fts (working in rolling friction bearings) requir- ng increased strength (shafts of tumblers, quad-	40 X 35 X	620-670 620-670		228-260 200-260
45	ints, feed gearboxes, etc.), as well as multisplined	38XA	620-670		200-260
45 580-620 40 550-600 18hings 20X 180-200 56-62 15 180-200 56-62 180-200 180	nafts working under medium-load conditions	45X	900-650		228-280
ushings 180-200 15 180-200 15 180-2000	ifts working in rolling friction bearings at light adds, as well as non-critical lightly loaded blined shafts	45 40	580 - 620 550 - 600		228-250 228-250
20X 180-200 15 180-200 20 180-200		shings	•		
0	dium-duty bushings working at high speeds and quiring increased strength, high wear resistance nd toughness	20X 15 20	180-200 180-200 180-200	56-62 56-62 56-62	
	0				

Heavily loaded bushings requiring high strength and adequate wear resistance 45x 320-400 45-50				228-260 200-260 200-260 200-260		228-260 200-260 200-260 228-280			
Heavily loaded bushings requiring high strength and adequate wear resistance by such ings bushings Worm Shafts Worm Shafts Worm Shafts Worm Shafts Heavy-duty worm shafts requiring high wear resist-life bushings Non-critical lightly loaded worm shafts of hand-driven mechanisms Claw clutches requiring high resistance to crushing frunning engagement) and increased tensile strength Non-critical claw clutches Cams, rollers, eccentrics, pawls of ratchet mechanisms of immers, stators of hydropumps and other parts subject to wear and requiring constancy of dimensions while in service Locating rings Engaging pins in presses Worm Shafts 180-200 180-200 180-200 180-200 180-200 180-200 455 Cams, rollers, eccentrics, pawls of ratchet mechaning constancy of dimensions while in service Locating rings Engaging pins in presses Worm Shafts 180-200 40X 620-670 620-670 620-670 88XA 630-670 88XA 6									
Heavily loaded bushings requiring high strength and adequate wear resistance Jig bushings Worm Shafts Worm Shafts Heavy-duty worm shafts requiring high wear resist- ance Non-critical lightly loaded worm shafts of hand- driven mechanisms Claw clutches requiring high resistance to crushing frunning engagement) and increased tensile strength Non-critical claw clutches Cams, rollers, eccentrics, pawls of ratchet mechanisms, formers, stators of hydropumps and other parts subject to wear and requiring constancy of dimensions while in service Locating rings Engaging pins in presses Loganians Worm Shafts 40X 38XA 45X ANXHMA SOXH 45XH3A	45-50 45-50 59-63		56-62 56-62 56-62	3	56-62 56-62 56-62		59-63 59-63	40-45 40-45	Above 50 Ditto Ditto Ditto
Heavily loaded bushings requiring high strength and adequate wear resistance Jig bushings Worm Shafts Worm Shafts Heavy-duty worm shafts requiring high wear resist- ance Non-critical lightly loaded worm shafts of hand- driven mechanisms Claw clutches requiring high resistance to crushing frunning engagement) and increased tensile strength Non-critical claw clutches Cams, rollers, eccentrics, pawls of ratchet mechanisms, formers, stators of hydropumps and other parts subject to wear and requiring constancy of dimensions while in service Locating rings Engaging pins in presses Loganians Worm Shafts 40X 38XA 45X ANXHMA SOXH 45XH3A									
Heavily loaded bushings requiring high strength and adequate wear resistance Jig bushings Worm Heavy-duty worm shafts requiring high wear resistance Non-critical lightly loaded worm shafts of hand-driven mechanisms Claw clutches requiring high resistance to crushing (running engagement) and increased tensile strength Non-critical claw clutches Cams, rollers, eccentrics, pawls of ratchet mechanisms, formers, stators of hydropumps and other parts subject to wear and requiring constancy of dimensions while in service Locating rings Engaging pins in presses	320-400 320-400 180-250		180-200 180-200 180-200	620-670 620-670 620-670 600-650	180-200 180-200 180-200	620-670 620-670 620-670 620-670	180-220 180-220	350-400 350-400	180-200 180-200 180-200 180-200
Heavily loaded bushings requiring high strength and adequate wear resistance Jig bushings Worm Heavy-duty worm shafts requiring high wear resistance ance Non-critical lightly loaded worm shafts of hand-driven mechanisms Claw clutches requiring high resistance to crushing (running engagement) and increased tensile strength Non-critical claw clutches Cams, rollers, eccentrics, pawls of ratchet mechanisms, formers, stators of hydropumps and other parts subject to wear and requiring constancy of dimensions while in service Locating rings Engaging pins in presses									
Heavily loaded bushings requiring high stren and adequate wear resistance Jig bushings Heavy-duty worm shafts requiring high wear resance Non-critical lightly loaded worm shafts of hadriven mechanisms Claw clutches requiring high resistance to crush (running engagement) and increased tenstrength Non-critical claw clutches Cams, rollers, eccentrics, pawls of ratchef mechanisms, formers, stators of hydropumps and ot parts subject to wear and requiring constated dimensions while in service Locating rings Engaging pins in presses	40X 45X XΓ	n Shafts	20X 15	40X 35X 38XA 58X	20X 15	40X 35X 38X A 45X	IIIX12 IIIX15	45 50	37XH3A 40XHMA 50XH 45XH
/ _unu	Heavily loaded bushings requiring high strength and adequate wear resistance Jig bushings	Worn	Heavy-duty worm shafts requiring high wear resist-	Non-critical lightly loaded worm shafts of hand-driven mechanisms	Claw clutches requiring high resistance to crushing (running engagement) and increased tensile strength	Non-critical claw clutches	Cams, rollers, eccentrics, pawls of ratchet mechanisms, formers, stators of hydropumps and other parts subject to wear and requiring constancy of dimensions while in service	Locating rings	Engaging pins in presses

Note. Hardening schedules for steel grades mentioned in the above table are presented in Tables 61, 64, 67.

Table 89

Steel Grades Recommended for the Manufacture of Some Automobile Parts and their Approximate Hardness Numbers

Name of parts	Steel grade	Hardness number, R _C	Remarks
Bushings	20, 18XFT	29-95	Carburise to 0.7-1.0 mm
Crankshafts	45	26	Surface hardening of journals
Camshafts	45	Ditto	Ditto
Injet valves	40X	22-28	
Exhaust valves	СХ8, ЭИ107	45	Thrust ring to be hardened
Piston rings	Pearlitic	98-106	Thermal stabilising on a mandrel in
	(431 11011	8	30 min
Standard parts, bolts, nuts, etc.	40X, 45	28-35	
Standard parts subject to friction	20X, 20	20-60	Carburise to 0.6-1.0, working section to be hardened
Piston pins	15X, 20	26	Carburise to 0.7-1.2 mm
Spherical pins	18XFT, 12XH3A	56	Carburise to 1.0-1.5, the head to be hardened
Spring pins	20, 20X	56	Carburise to 1.0-1.3 mm, the thread to be tempered to 35-40
Pivots of rotating cam	12X H3A, 12X2H4A, 20X3	56	Carburise to 0.7-1.2 mm

Steel Grades Recommended for the Manufacture of Some Agricultural Machinery Parts and

Their	Their Approximate Hardness Numbers	ardness Num	bers
Name of parts	Steel grade	Hardness number, R _C	Remarks
Share sidewalls and skim coverer of potato planter	65Г	38-50	
Bushings of self-propelled combines	35	45-50	Carburise to 0.6-1.0 mm to a length of 20 mm
Medium-duty shafts and axles	45-40 Cr.5	30-40 40-45	Depending on loads
Heavy-duty shafts and axles	Ditto	52-58	HF hardening
Field disks for ploughs, harrows, etc.	Cr.6,Л53 65Γ. 70Γ	35-50	
Twine holder	8,6	40-50	Local hardening
Chain links for elevators of potato diggers	65Γ	38-45	Ditto
Teeth of thrashing drums and con- caves	45, Cr.6	42-52	Ditto
Teeth of traverse rakes	65F, 70F	37-46	Pin hardness number $>269~H_{B}$
Chain wheel of combine drum shaft	15	54-62	Carburise to 1.0-1.5 mm
Ridger mouldboard vanes	Cr.5	33	
Agricultural scythes	Y7A, Y8A	40-50	
Crosspiece of Hooke's joint and cross-	Cr.2	40-55	Carburise to 0.8-1.2 mm

Name of parts	Steel grade	Hardness number, R _C	Remarks
Teeth of various types of cultivators	65F, 70F	38-50	Blade area to be quenched
Various types of plough-shares	Л65, 50, Ст.6	38-50	Working section to be quenched
Beaters, crushers	Cr.2	56-62	Working section to be carburised and quenched
Points of harrows and seeders	JI53, Cr.5	38	Working section to be quenched
Knives of feed mincers	V9, 65F, 70F	46-56	Ditto
Knives of beet-lifting machines	98	45	
Knives of sheafer knotter	99	50-58	Working section to be quenched
Knives and combs of sheep shearing machine	y2A, yBA	09	
Various types of mould-boards	Ст.2	50	Carburise to 1.0-1.5 mm
Ginning and lintering saws of cotton scrapers	82	30-35	
Plates and pawls of plough automatic device	Cr.2	45-50	Carburise
Pins of self-propelled combine	40	35-40	
Sections of harvester cutting devices	3.9	20-60	Working section to be quenched
Spring mountings of cultivators	65F, 70F	38-47	
Slide block of self-propelled combine	40	45-50	
Worm shafts of high strength and wear resistance	40X	38-45	
Worm shafts of high wear resistance	20, Cr.2	50-58	Carburise to 0.8-1.2 mm

Chapter VII

HEAT-TREATMENT OF CAST IRON

Cast-iron articles, depending on specifications, may be annealed, normalised, hardened, tempered or case-hardened.

Low-temperature annealing. Natural ageing, consisting of a prolonged maturing of articles in storehouses, was employed as a means of removing stresses in cast-iron articles. This was a long-term operation

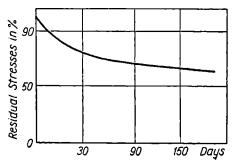


Fig. 29. Lowering of internal stresses in relation to the duration of natural ageing of cast iron

which, all things considered, did not remove all the stresses entirely

(Fig. 29).

Currently, a low-temperature annealing procedure, relieving all the stresses completely, has been successfully applied (Fig. 30); this procedure is carried out according to a schedule which is the same for all grades of cast iron: the articles are charged to a furnace heated to maximum 250-300°C, the furnace temperature is then raised and the said work is held over for a period ranging from 2 to 8 hrs, depending on its size and shape; after this, the furnace is cooled at a rate of 20-30°C per hour down to 200-150°C, and the articles are discharged in the air.

Low-temperature spheroidising annealing. This procedure is generally used for extra-tough and malleable cast irons; it greatly increases ductility, resilience and antifriction properties. This treatment is rarely applied to grey cast iron, as along with a lowering of

hardness and improvement of machinability it brings down sharply

its mechanical properties.

The annealing schedule comprises: heating (gradual for fancyshaped parts) to 670-700°C, holding from 1 to 4 hrs and cooling with the furnace.

Spheroidisation of white cast iron. This process is employed to reduce hardness and to improve machinability of white cast iron. The work is heated to 850-950°C and, after a through heat soaking, cooled with the

furnace; larger articles are air-cooled.

To remove the whitening effect in the thin sides of castings from extra-tough cast iron, the articles are annealed according to the following schedule: heating to 800-850°C, holding for 2-4 hrs, cooling to 650°C, holding for 4-6 hrs and cooling with the furnace to 200-175°C.

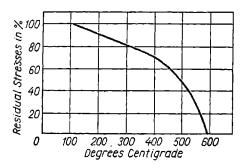


Fig. 30. Effect of heating temperature on removal of internal stresses according to published data

Normalising. It is employed to increase strength and wear resistance of grey cast iron. On normalising, the article is heated to 850-900°C and, after a through heat soaking, cooled. Intricate parts should be tempered, after normalising, at 600-650°C.

Hardening is intended to increase hardness, strength and heat re-

sistance of articles made of grey cast iron.

The hardening temperature in oil is 830-870°C, and in water 800-820°C. Odd-shaped parts, prior to hardening, are heated slowly whereas simple-shaped parts are heated rapidly by charging them into the furnace heated to the requisite temperature. When initial structure is ferritic, holding time is increased. Isothermal hardening of grey cast iron, which markedly reduces cracking and warping, is carried out at 830-900°C through immersion in a bath at 200-400°C.

Tempering. Tempering of cast-iron parts, after hardening, is imperative. The process temperature is from 200 to 500°C, depending on properties required. When maximum wear resistance is needed, the work is tempered at 200-250°C. The best combination of strength and wear resistance is attained in the 350-450°C range, the hardness averaging $H_{\rm p} = 321-418$.

Testing cast-iron articles for hardness. Rockwell hardness test, after quenching, fails to bring forth the exact picture of the hardening effectiveness, because of presence of graphite in the cast iron. Fig. 30 a illustrates an indentation produced by the Rockwell testing machine on the hardened surface of a tractor cylinder bushing, the load applied equalling 60 kg. To ascertain what amount of graphite is included in the indentation area of the diamond cone, a circle of equal diameter is drawn next to the indentation. The hardness number of the said bushing is 45 units $R_{C_{150}}$, while the hardness number of its metallic base determined by the ΠMT -3 device equals 62 $R_{C_{150}}$. It is evident that the R_C =45 is an average hardness number, which is a function not only of the degree of hardening but also of the amount of graphite (Brinell hardness number of only 6-10 kg/mm²) present in the cast iron.

Table 91
Causes and Correction Methods for Annealing Defects in Grey and Malleable Cast Irons

Defect	Detection	Chief cause	Correction methods
Low hardness of grey cast iron after stress- relieving an- nealing	Hardness test	Disintegration of cementite on heating above 600°C	Normalising at 900-950°C for plain-shaped parts and sub- sequent stress- relieving tem- pering
High hardness of white cast iron after annealing	Ditto	Insufficient femper- ature and/or holding on an- nealing	Repeated an- nealing at nor- mal tempera- ture and ade- quate holding time
Free structural cementite in malleable cast iron	Microstructure checking	Low temperature or insufficient holding at the 1st stage of spheroid- ising	Repeated an- nealing ac- cording to set schedule
Excessive amounts of la- mellar pear- lite in malle- able iron	Ditto	Non-compliance with cooling schedule or insuf- ficient holding at the 2nd stage of spheroidising	Repeated an- nealing at 710-730°C
Lamellar graphite in malleable iron	Ditto	Presence of free graphite in mal- leable iron	Irremediable defect

Improving of extra-tough cast iron is effected according to the following schedule: hardening in oil from 870-900°C and tempering at 500-600°C.

Mechanical properties: tensile strength up to 120 kg/mm², yield point up to 100 kg/mm², Brinell hardness number—375.

Heat-treatment of white cast iron. Malleable cast iron is obtained by annealing white-iron castings according to a number of technological flowsheets, two of which are given below.

Flowsheet No. 1. The articles are packed into boxes and covered with dry river sand. The boxes are heated to 1000-1050°C in a furnace. kept there at the said temperature for 6-10 hours and cooled with the furnace down to 680°C. The cooling in the 760-680°C range is slow (2-3°C. per hour), after which the cooling rate is immaterial. Prolonged holding at temperatures slightly below the A, point can be substituted for slow cooling.

Flowsheet No. 2. Heating of articles in an antioxidising medium up to 940-960°C, holding at the said temperature for 12 hours, cooling to 760°C, slow cooling (for 20-22 hrs) in the 760-680°C range, 10-hr cooling down to 550°C, subsequent cooling effected at any rate desired.

Table 91 lists various defects encountered in the heat-treatment of cast from and the methods required to remedy them.

Chapter VIII

HEAT-TREATMENT OF NON-FERROUS METALS

Copper and copper-base alloys. Copper is annealed to remove cold working effects and to reduce hardness. To this end, copper is heated to 500-700° C and then cooled after heat soaking. Any rate of cooling will suit the case, the cooling being effected with the furnace, in the air or water; when the work is water-cooled, the scale flakes off more readily than with the cooling being effected in the furnace or air.

The post-annealing mechanical properties of pure copper are tensile strength $\sigma_b = 20 \text{ kg/mm}^2$, relative elongation $\delta = 50\%$, Brinell hardness number $H_B = 35$ (approximate data).

Conventional designations for copper alloys are given in Table 92.

Table 92
Conventional Designations of Elements in Specifications for Copper and Nickel Alloys

Element	Designation	Element	Designation
Aluminium	А	Magnesium	Mr
Beryllium	Б	Manganese	Mu
Chromium	Х	Nickel	H
Copper	М	Silicon	K
Iron	Ж	Tin	O
Lead	С	Zinc	Ц

Brasses (copper-zinc alloys). The chemical composition of brasses can be identified according to the designation of the alloy, namely: the letters indicate the elements contained in the brass, while the figures specify the percentage of these elements. The letter "I" designates brass, while the subsequent letters stand for elements comprising the brass. The first figure designates the percentage of copper, the remaining figures specifying the percentage of the elements in the same sequence as these are included in the designation of the alloy. The amount of zinc is the balance to 100%.

Examples:

1. Π62—brass, copper—62%, zinc—38%.

^{2.} ЛЖМu59·1-1—brass, copper—59%, iron—1%, manganese—1%, zinc—39%.

Brass is annealed to remove cold working effects and restore the structure. On annealing, the parts are charged to a furnace heated to the required temperature. Brasses, even when slightly distorted, tend to crack when stored in moist atmosphere; because of this, all articles and half-finished pieces should be subjected to low-temperature anneal-

ing at 300-350°C.

Intermediate annealing subsequent to the rolling of brass, as well as annealing of blanks are carried out at temperatures listed in Table 93. Brass castings which should display minimum deformation in machining and service are annealed at 300-350°C with a 2-4 hr holding. Castings from brass JK80-3J should be homogenised at 750-760°C for 1-1.5 hrs in order to eliminate microporosity. When the said defect is highly pronounced, homogenising should be repeated twice (Table 93).

Table 93

Approximate Annealing Schedules and Mechanical Properties of Brasses

				imate proper	mechan- ties
Name of brass	Designation	Annealing tempera- ture, °C	Tensile strength, kg/mm²	Elonga- tion, %	Brinell hardness number
Tombac. Ditto. Semi-red brass. Ditto. Brass. Ditto. Ditto. Ditto. Aluminium brass. Ditto.	Л96 Л90 Л85 Л80 Л70 Л68 Л62 Л59 ЛА85-0.5	540-600 650-720 600-700 600-700 600-700 520-650 600-700 600-670 650-700 600-650	24 26 26 31 33 33 36 39 35 38	52 44 43 52 55 56 49 44 60 50	59-63 53 54 53 52 56
Ditto	ЛАН759-3-2 ЛАН59-3-2 ЛН65-5 ЛМц58-2 ЛЖМц59-1-1 ЛО90-1 ЛО70-1 ЛО62-1	600-650 600-650 600-650 600-650 650-720 560-580 550-650	38 50 38 44 47 28 35 38	50 42 65 36 40 62 37 37	50 117 65 85 80 57 48 85 85
Lead brass Ditto Ditto Ditto Ditto	ЛС74-3 ЛС64-2 ЛС63-3 ЛС60-1 ЛС59-1	600-650 620-670 620-650 600-650 600-650	35 34 35 35 42	48 60 45 50	55 50 80 75

Note. Cooling after annealing is effected in the air or within the furnace.

Bronzes. Depending on their chemical composition, bronzes belong to the tin or tinless varieties. Chief constituents of tin bronzes are copper and tin. Tinless bronzes are constituted by copper-bearing alloy comprising one or several additional elements: aluminium, silicon. manganese, etc. The chemical composition of bronzes is recognised by the designation of the alloy, as is the case with brasses.

On heat-treatment, the work can be charged to a furnace heated to the requisite temperature. Cooling, after annealing, can be carried out in the air or within the furnace. Annealing temperatures are adopted with relation to the thickness of the material. Temperatures lower by 10% as compared to those listed in Table 94 can be recommended for

bands and small-sized wire (up to 0.5 mm).

Table 94 Annealing Schedules and Approximate Mechanical Properties of Bronzes

		**	Approximate mechanical properties				
Name of bronze	Designation	Heating tempera- ture, °C	Tensile strength, kg/mm²	Elonga. tion, %	Brinell hardness number		
Tin-phosphor bronze	Бр.ОФ6.5-0.4 Бр.ОФ4-0.25 Бр.ОЦ4-3 Бр.ОЦС4-4-2.5 Бр.А5 Бр.А7 Бр.АЖ9-4 Бр.АМц9-2	600-650 600-650 600 600 600-700 650-750 700-750	34-45 34 35 30-35 38 42	60-70 52 40 35-45 65 70 40	70-90 55-70 60 60 60 70 110		
bronze	Бр.АЖМц 10-3 -1 .5	650-750	60 -6 4	27-30	12 0		
Aluminium-iron- nickel bronze Beryllium bronze Silicon-manganese	Бр.АЖН10-4-4 Any grade	700-750 6 5 0-750	60 50	35 30	140-160 100		
bronze	Бр.ҚМц3-1	600-680			80		
bronze	Бр.КН1-3 Бр.Мц5	650-750 700-750	45 30	1 2 40	80		
various grades		60 0 -650					

Castings from a number of tin bronzes, which display leakage when tested for water-tightness, are consolidated through annealing according to the following schedule: heating to 700-710°C, holding for 2 hours for each 25 mm of thickness and cooling within the furnace.

Spring bronzes which are marketed in a strain-hardened state cannot be annealed to facilitate bending. In this case tempering at 300-

320°C is sometimes permitted.

Beryllium bronzes should be quenched in water at 20°C as a maximum, while the transfer time from the furnace to the tank should be as short as possible. Slow quenching tends to impair springiness of this bronze. Dissociated ammonium is the best medium for the prehardening heating of beryllium bronze blanks.

Heating in dissociated ammonium makes it possible to detect the defects of the material prior to manufacturing of parts, as the hydrogen of the ammonium forms bulges or blisters on the blank surface by com-

bining with the oxides of the metal.

When heating in a usual chamber furnace in air atmosphere the fol-

lowing precautions should be envisaged:

1. Pans which carry the articles should be free from dirt, oil and scale.

2. The articles should be piled on the pan uniformly. Heating in bulk impairs the quality of the articles.

Holding time of heated articles should average 8-12 min at 760-

780° C.

Beryllium bronze is aged in an air furnace or saltpetre bath. Ageing

time is approximately 2-3 hrs.

It should be pointed out that sensitive elements (diaphragms, bellows, etc.) manufactured from bronze Ep.B2.5 do not show superior quality due to the fact that high beryllium content in the bronze frequently causes structural non-uniformity, increased brittleness, etc.

The Moscow Research Institute for Non-Ferrous Metal Working has recommended, for sensitive elements, a new grade of beryllium bronze, the BHT1.9 whose chemical composition is: beryllium 1.8-2.1%, nickel 0.2-0.4% and titanium 0.1-0.25%.

Diaphragms and springs from the said bronze possess high fatigue

endurance (durability) and lower hysteresis.

The heat-treatment schedule for BHT1.9 bronze is the same as for bronze Bp.B2.5. The low response of the former to heat-treatment conditions and its cost, being lower than that of the Bp.B2.5 bronze, should be of interest.

Tempering time of grade X0.5 chromium bronze is accepted to

be 6 hrs (Table 95).

Nickel is subject either to oxidising or bright annealing depending on requirements. Annealing temperature for nickel is in the 750-900°C range. Annealed nickel possesses a tensile strength σ_b of 50 kg/mm², elongation of 40%, Brinell hardness number H_B =70-90.

Bright annealing of nickel is carried out in the atmosphere of dried hydrogen from generator gas, of dissociated ammonium, etc. When controlled atmosphere is not available, bright annealing is effected as follows: parts or materials, subject to annealing, are placed into pots or boxes (cast-iron or steel), a small amount of charcoal is added thereto, the boxes are covered with lids, the seams are thoroughly luted with a mixture of ordinary clay and fireclay, and then charged to the furnace. After annealing the packets should not be opened until cold. When annealing coils of wire, the latter, prior to packing, are dipped

When annealing coils of wire, the latter, prior to packing, are dipped in a water solution of whiting and dried, to prevent sticking of coils.

Nickel alloys. The chemical composition of nickel and copper-nickel alloys can also be readily recognised by the alloy designation. When nickel is the basic element of the alloy, the first letter of the alloy designation is "H", while if copper is the basic element, the first letter of the alloy designation will be "M". The letters and figures that follow designate the elements and their percentage in the alloy.

Table 95

Approximate Hardening and Tempering Schedules for Bronzes

		Harden	ing	Tempering	
Name of bronze	Designation	Heating tempera- ture, °C	Cooling medium	Heating tempera- ture, °C	Brinell hardness number
				1	
Aluminium-iron bronze Aluminium-man-	Бр.АЖ9-4	850	Water	350	_
ganese bronze Aluminium-iron-	Бр.АМц9-2	800	Ditto	400	150-187
manganese bronze	Бр.АЖМц 10-3-1.5	830-860	Ditto	300-350	207-285
Aluminium-iron- nickel bronze Chromium bronze		980 950-1000	Ditto Ditto	400 400	400 max
Cunial	Бр.НА14-3 Бр.НА6-1.5 }	900	Water or air	500	260 210
Beryllium bronze	Бр.Б2 ВНТ1.7 БНТ1.9	760-780	Water	310-330 *	300 min
Silicon-nickel bronze	Бр.КН1-3	850	Ditto	450	150-200
	l	l	l	1	l

^{*} Bronzes used for diaphragms are tempered at 290-310° C.

Examples:

Nickel allovs:

^{1.} Alumel HMuAK2-2-1, averaging 2% manganese, 2% aluminium, 1% silicon, the balance (95%) being nickel and cobalt.

2. Chromel HX9.5, averaging 9.5% chromium, the balance being nickel and cobalt.

Copper-nickel alloys:

1. Copel MHMu43.0.5, averaging 43% nickel, 0.5% manganese, the balance being copper.

2. Cunial MHA13-3, averaging 13% nickel, 3% aluminium, the bal-

ance being copper.

Bright annealing of nickel and copper-nickel alloys is similar to

that of nickel (Table 96).

Copper-nickel alloys Cunial "A" MHA13-3 and Cunial "B" MHA 6-0.5 are toughened by means of heat-treatment according to the following schedule: hardening from 900°C in water and tempering at 500°C for two hours. Mechanical properties following this treatment are:

		MHA13-3	MHA6-0.5
Tensile strength	in kg/mm ²	90-95	65-75
Elongation in %		5	7
Brinell hardness	number	260	210

Low-temperature annealing of nickel and its alloys, aimed at removing internal stresses, is carried out in the 250-300° C range.

Magnesium alloys. Cast as well as wrought magnesium alloys (Table 97) are subjected to a variety of heat-treatments, the set

purpose of which is indicated in Table 98.

The work is heat-treated in furnaces of the following types: shaft, vacuum, chamber and bath-type furnaces (containing a mixture of potassium and sodium bichromates).

It is recommended, in order to prevent oxidation, to heat the work in a controlled gas atmosphere composed of a mixture of air with 0.7-1% of sulphurous anhydride is not available, the protective or controlled atmosphere is obtained by charging 3-4 kg of pyrites per 0.7-1 ton of the weight of the work.

The furnace temperature should be adjusted to within $\pm 5^{\circ}\,\text{C}$ with

the aid of an automatic regulator.

To avoid possible fusion, the heating of large articles should be a two-stage procedure: first, up to 330-340°C or 360-370°C, depending on hardening temperature, then, after a 2-4 hour holding, to the final required temperature Prolonged holdings on heating are necessary because of very slow consummation of the diffusion processes (Table 99).

To prevent cracking on hardening, it is not recommended to use cold

water.

Strict temperature control is to be ensured when heat-treating magnesium alloys, as the latter inflame readily. At no time should the requisite temperatures be surpassed. Articles to be heat-treated should be clean from magnesium dust, chips, burrs, grease and oil.

Defects possible in the heat-treatment of magnesium alloys are de-

scribed in Table 100.

Aluminium and its alloys. Heat-treatment of aluminium involves annealing at 370-400° C with subsequent air-cooling. After annealing

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Table 96

Brinell hardness number 120 - 130 150 - 200 130 60 - 70 70 70 82 38 50 - 60 70 85 - 90 75 - 90 120 160 Annealing Schedules and Approximate Mechanical Properties of Nickel and Copper-nickel Alloys Mechanical properties Elongation, % 50 max 18 min 35 min 35-40 23-28 35 35-45 40 36 Tensile strength, kg/mm² 75 max 45-50 60-70 38-40 40 38-45 40 25-30 39 40 40-50 40-55 40-64 64 Annealing temperature, C 800-850 800-850 850-900 900-950 780-810 600-780 700-750 750-760 800-850 800-850 700-750 850-900 750-850 750 550 500 Designation of alloys HMXKMu28-2.5-1.5 HX9.5 and HX9 HMuAK2-2-1 MHXMu30-0.8-1 MHI[15-20 MHI[Cl7-18-1.8 MH5 MHMu43-0.5 MHMu40-1.5 MHMu3-12 НМц2.5 НМц5 X20H80 X15H60 MH0.6 MH16 MH19 Manganese-bearing nickel Leaded German silver Name of alloy Copper-nickel alloy "TII" alloy Ditto German silver . Ferronichrome Nickel silver Copel . . . Nichrome . Manganin . "TE" alloy Constantan Monel . . Chromel Alumel

Table 97
Chemical Composition of Commercial Magnesium Alloys

		Chemi	cal composition	, %	
Grade	Aluminium	Zinc	Manganese	Silicon	Magnesium
		Foundry A	llloys		
МЛ1 МЛ2 МЛ3 МЛ4 МЛ5 МЛ6	2.5-3.5 5-7 7.5-9.3 9-11	 0.5-1.5 2-3 0.2-0.8 Up to 2.0	1-2 0.15-0.5 0.15-0.5 0.15-0.5 0.1-0.5	1-1.5 — — — — —	balance
		Wrought A	l <i>lloy</i> s		
MA1 MA2 MA3 MA4 MA5 MA8	3-4 5.5-7 6.5-8 7.8-9.2	0.2-0.8 0.5-1.5 2.5-3.5 0.2-0.8	1.3-2.5 0.15-0.5 0.15-0.5 0.15-0.5 0.15-0.5 1.5-2.5		The

Table 98

Conventional Designations, Uses and Varieties of Heat-treatment Processes Applied to Magnesium Alloys

Designation	Purpose	Heat-treatment applied			
T2	Removal of internal stresses, removal of strain-hardening and increase of ductility	Annealing			
T4	Improvement of mechanical proper- ties	Hardening			
Т6	Increase in yield point	Hardening and ageing			

the mechanical properties are: tensile strength σ_b —8-10 kg/mm², relative elongation δ -40-45%, Brinell hardness number H_B =20.

When only internal stresses are to be removed, aluminium is annealed at 150° C.

Two types of aluminium alloys are distinguished: the wrought alloys and the cast alloys.

The wrought aluminium alloys are divided into two groups: (1) alloys which are not toughened by heat-treatment; (2) heat-treatable allovs

(Table 101).

Aluminium alloys acquire good ductility as a result of annealing in saltpetre baths for 20-30 min or in electric furnaces in air atmosphere for 1-2 hours. The duration of heating is indicated in Table 102, while the temperatures for heat-treatment schedules are presented in Table 103.

Best ductility and minimum hardness of aluminium alloys are obtained after a 30-min annealing at 420°C with subsequent cooling, first, within the furnace at a rate of 30° C down to 280° C, and then in

the air.

Heating for the toughening of aluminium alloys is carried out in the furnaces mentioned above. Temperatures in excess of the upper range of hardening temperatures may cause burning revealed by sharp darkening of the surface, and by the presence of blisters and cracks.

Water temperature for hardening should not fall below 30° C. As machine parts grow both in size and complexity, it is recommended to increase the temperature of water, bringing it up to 70-80° C. An increasein temperature reduces the formation of cracks as well as warping.

The time interval between the removal of parts from the furnace

and their immersion in water should not exceed 20-30 seconds.

Parts heated in saltpetre baths should be thoroughly flushed and wiped with rags after hardening. Following hardening aluminium alloys toughen to a certain extent but are still as ductile as to permit distortion of the article.

After a certain lapse of time the alloy ages. The time interval between the hardening and the beginning of ageing is given in Tabl**e** 105. Ageing is speeded up through tempering at 150-170°C for

6-12 hours.

"Fresh"-hardened condition of the articles can be retained by holding them at a temperature ranging from 0 to -20° C after hardening.

Hardnened and aged duralumin can be restored to "fresh"-hardened condition through short-term heating (0.5-1 min) at 240-260° C. After the necessary working (stamping, etc.), the alloy is aged again.

Cast aluminium alloys. Cast aluminium alloys (Table 104), depending on requirements, are heat-treated; the conventional designations of the processes and their objectives are presented in Table 106.

Prolonged heatings are essential for the heat-treatment of cast alloys, because of their very coarse structure. Fancy-shaped parts are preheated in electric furnaces to 300-350° C (Table 107) prior to heating on hardening in salt baths.

When treating aluminium as well as magnesium alloys in furnaces with air atmosphere, furnaces with positive mixing of atmosphere

should be used to provide uniform heating (type $\Pi H-31$, etc.).

Table 99

Recommended Heat-treatment Schedule for Magnesium Alloys in Electric Furnaces with Air Atmosphere and Approximate Mechanical Properties (Undervalued for the Most)

ies	ness number	
opert	Brinell hard-	
Mechanical properties	Elongation, %, minimum	
Mechai	Tensile strengih, kg/mm²	
	gnilooD muibəm	
Ageing	Rolding, 211	
	Heating temperature, °C	Allous
dening	Gooling muibəm	A. Foundry Alloys
Annealing and hardening	,BaibloH 211	
 Anneali	Heating temperature, oC	
	Conventional designation of heat-treat-	
	Alloy	

Alloys	
Foundry	
Ą.	

	40	30	40	- 23	22	09	20	22	9		20		9	65		_
	2	က	9	က	4	2	7	4	2				က	-		
	6	6	91	91	21	22	15	21	22		15		21	21		
	1	1	1	1	1	Air	[1	Air	Ditto	J		1	Air		
	1	١	I	J	J	91	1	I	91	∞	1		1	4-8		
,	1	1	1	1	1	170-180	1	ı	170-180	or 195-205	ı		I	185-195		
•	Within the furnace	Ditto	Ditto	Ditto	Air	Ditto	Ditto	Ditto	Ditto		Within the	furnace	Air	Ditto		
	3-5										3-5		24-32	24-32		
	200-250	200.250	170-250	170-250	375-385	375-385	170-250	410-420	410-420		170-250		405-415	405-415		
	T2	T2	T2	T2	T4	Te	T2	T4	Te		T2		T4	T6		
	MJII	M.712	M3	MJ14			M.715				M.Л6					

pertics	-brsd Manira resonum esen
ical pro	Flongation, %
Mechanica	Tensile strength, kg/mm²
	gnifooO muibəm
Ageing	Holding, hrs
ď	Heating temperature, C
dening	SuilooD muibem
ing and hare	Holding, hrs
Annealin	Heating temperature, °C
	Conventional designation of heat-treat- ment
	Alloy

B. Wrought Alloys

50 70 70		80 64 55	
1887		7 15	
348		35	
1111		Air	
1111		 16.24 	
1111		170-180	
Air Ditto Ditto Ditto	ent	2-3 4-10 Hot water 3-6 Air 4-12 Hot water 4 Ditto 2-3 Air	
2.5.4 2.5.4 6.4	p Treatm	2-3 4-10 3-6 4-12 2-3	
340-400 Up to 400 320-380 320-350	Ste	330-340 375-385 350-380 410-420 410-420 280-320	
72 72 72	Т4	27,42,52	
MA1 MA3 MA4		MA5 MA8	

Notes. 1. For castings from MJ5 alloy, with the side thickness above 12 mm, the hardening temperature is 415 to 425°C, the holding time being 16.24 hours.
2. Holding time is taken without the time interval necessary for the heating.
3. Holding time varies within the limits indicated, depending upon the mass of the article.

9		Defects in Heat-treatment of Magnesium Alloys and Their Prevention	ment of Magnesium	Alloys and Their Pre	vention
	Defect	Symptom	Detection	Cause	Prevention
	Low mechanical properties		1. Testing for me-chanical properties ing or insufficient holding time	Low heating temper- ature on harden- ing or insufficient holding time	Control of furnace run. Correction by repeated heat-treatment with strict compliance to
	Local fusing	1. Strongly oxidised Visual inspection metallic prills on the	inspection Visual inspection	1. Coarse cast struc- ture	schedule 1. Heating with prelim- inary preheating
		Surface 2. Minute surface gas cavities and internal microcavities		Microstructure in 2. Rapid heating on spection 3. Non-uniform furnace temperature	2. Checking of furnace pyrometers 3. Lowering of heat-treatment temperature by
	Oxidation on heating	on Surface of the metal Visual inspection covered with powder from grey to black in colour. Minute cavities uncovered		1. Overheating or non-uniform furnace temperature 2. Air leakage 3. Water vapour	1. Overheating or Similar to those on fusing non-uniform furplus protective atmosnace temperature phere phere 3. Water vapour
	Grain growth in cast alloys during heat- treatment	Grain growth in Bright spots on ma-cast alloys chined surface before treatment	Visual inspection	present in the furnace During the casting process	Heating (prior to hardening) to 300° C for 1-2 hrs to remove internal stresses

Table 101
Chemical Composition of Aluminium Wrought Alloys

			Chemical o	ompositio	n, %	
Grade	Copper	Magnesium	Manganese	Silicon	Other elements	Alumi- nium
АМц АМг АМг5п Д1	 3.8-4.8	2.0-2.8 4.7-5.7 0.4-0.8	0.2-0.6	- - -	 	
Д6 Д7 Д16 Д1П	4.6-5.2 3.0-4.0 3.8-4.9 3.8-4.5	0.25-0.5	0.5-1.0 0.25-0.5 0.3-0.9 0.4-0.8		 	
ДЗП Д16П Д18П АВ	2.6-3.5 3.8-4.5 2.2-3.0	1.2-1.6	0.3-0.7 0.3-0.7 — 0.15-0.35	_ _ _ 0.5-1.2	- - -	апсе ——
AK AK2	3.5-4.5	-	 	4.5-6.0 0.5-1.0	Nickel 1.8-2.3 Iron 0.5-1.0	- The balance
AK4-1	1.9-2.5 1.9-2.5		_	0.5-1.2 —	Nickel 1.0-1.5 Iron 1.1-1.6 Nickel 1-1.5 Iron 1.1-1.6	
АК6 АК8 АЛД Д12 В95	1.8-2.6 3.9-4.8 — — — 1.4-2.0	0.4-0.8 0.5-0.9 0.8-1.3	0.4-0.8 0.4-1.0 — 0.9-1.4 0.2-0.6	0.7-1.2 0.6-1.2 0.8-1.0 —	Iron 0.2-0.5 Zinc 6.0	
		[<u>, </u>		Chromium 0.2	

 $\it Note.$ Chromium may be substituted for manganese in the same amounts as the latter in alloy grades AMr and AB.

Cast aluminium alloys with high magnesium content (AJI8) are to be heated, on hardening, in air furnaces instead of saltpetre baths, because of inflammation hazard. The interval between the discharge from the furnaces and the cooling should be reduced to a minimum. The more complex the shape of the article the higher should be the temperature of the cooling medium.

Titanium and its alloys. While being low in specific gravity (4.5) titanium has great tensile strength and ductility, i.e., $\sigma_h = 45.60 \text{ kg/mm}^2$. elongation $\delta = 25\%$ (Tables 108, 109).

Commercial grade BT1 titanium and its alloys take on considerable

cold working when rolled without heating (stamping, etc.), and they should be annealed in air atmosphere or vacuum to restore ductility.

Table 102 Approximate Heating Time On Hardening of Aluminium Wrought Allovs (Minutes)

		Thi	ckness o	r dian	neter o	f work	, mn	n		
Heating equipment	Up to 0.8	0.8-2.5	2.5-5.0	5-12	12-20	20-50	60	70	80	90
Saltpetre bath	8	10	12	15	30	40-60	6 0	70	80	90
Air electric fur- nace	12	20-30	40	80	90	110	130	130	180	180

Notes. 1. Holding time is counted off from the moment the furnace reaches

requisite temperature after charging.

2. Holding time is reckoned bearing in mind the greatest cross-sectional dimension of the part.

3. Excessive holding of plated alloys impairs the properties of the plated

4. In case of repeated hardening, the time of heating should be cut down by half.

Annealing temperature for plain titanium is 510-570° C, while that for titanium alloys is 650-750°C.

Finished articles and sheet stock are annealed at lower temperatures and reduced holding time as compared to half-finished products and bulky articles. Prolonged holding, especially at high temperatures. causes scaling and brittleness.

Zinc is annealed at 50-100°C whenever its softening is required. After annealing, the mechanical properties of zinc are: tensile strength $\sigma b = 7-10 \text{ kg/mm}^2$; elongation $\delta = 10-20\%$.

Silver and silver-platinum alloys are annealed at 650-700° C with cooling being effected in water, preferably acidulated.

Ta ble 103

Heat-treatment Schedules and Typical Mechanical Properties of Wrought Aluminium Alloys

				-		-	1		
	An	Annealing	Hardening	ning	Ageing	ing	Mechanical		properties
Grade	Heating temperature, °C	Cooling muibəm	Heating temperature, oC	Cooling mulbəm	Heating temperature, °C	BuibloH emit	Tensile strengih, kg/mm²	Elongation, %	Brinell sandasa number
AMn	350-410	Air or water		Not a	- Not applied	_	13	20	30
AMr	350-410	Ditto		Di	Ditto	_	20	23	45
AMr5n	340-370	Air		D	Ditto		27	23	20
Д1	340-370	Ditto	1	1	1		21	18	45
	l	ı	495-510	Water	15-20	4 days	42	18	100
Д6	340-370	Air	1	1	J	1	22	15	20
	l	1	497-503	Water	15-20	4 days	46	15	105
Д16	340-370	Air	l	l	1	1	21	18	42
		1	495-505	Water	15-20	4 days	47	17	105
ДЗП	340-370	Air	ı	i	-	Į	17	20	45
	ı	ı	490-500	Water	15-20	4 days	34	50	80
Д18П	340-370	Air	i	i	1	1	16	24	38
	1	!	490-505	Water	15-20	4 days	90	24	20
			_						

Table 103 continued

	Anı	Annealing	Hardening	ning	Ag	Ageing	Mechanical		properties
Grade	Heating temperature, C	Cooling medium	Heating temperature, O°	gailooD muibem	Heating temperature, Oo	gnibíoH 9mit	Tensile strength, kg/mm²	Elongation,	Brinell hardness number
AB ("avial")	340-370	Air	ı	1		<u> </u>	13	24	30
	i	I	515-530	Water	150-160	6 hrs	33	12	92
AK				-	Not subjec	Not subject to above treatments	treatment	S	
AK2	350-460	Air	I	1	1	1	-	i	1
		l	510-520	Water	165-175	15-18 hrs	42	13	100
AK4	350-460	Air	ı	ī	l	ı	ı	ı	1
	[ı	525-540	Water	165-175	15-18 hrs	44	10	110
AK6	350-460	Air	l	I	i	1	ı	1	1
	ı	I	505-515	Water	150-160	12-15 hrs	42	13	105
AK8	350-460	Air	1]	I	1	l	1	1
		1	495-505	Water	175-185	5-8 hrs	49	13	130
AK9			520-535	Water	130-160	5 hrs	39	10	115
B95	420-440	Air	l	1	ı	i	56	13	I
	1	ı	465-475	Water	120-125	24 hrs	55	16	150
_				_		_			

Chemical Composition of Aluminium Foundry Alloys

			Chemi	Chemical composition, %	%	
Grade of alloy	Silicon	Copper	Magnesium	Мапданеѕе	Other elements	Aluminium
АЛІ	1	3.75-4.5	1.25-1.75	1	Nickel 1.75-2.25	
A.712	10.0-13.0	1	100	0000	1 1	
AJ13	#.0-0.0 #.0-10.5	0.0-0.1	0.12-0.30	0.25-0.5	1	
A.115	4.5-5.5	1.0-1.5	0.35-0.6	1	I	
A.716	4.6-6.0	2.0-3.0	l	ı	ŀ	
A.117]	4.0-5.0	l	1	1	
А.Л8	1]	ı	ə
А.Л9	6.0-8.0	1		ı	1	ווכ
A.7110B	4.0-6.0	5.0-8.0		1		sia
AJ111	6.0-8.0	1	0.1-0.3	1	Zinc 10.0-14.0	3 q
АЛ12	1	9.0-11.0	l	!	I	ə
A.Л13	0.8-1.3		4.5-5.5	0.1-0.4	1	чј
АЛ14В	0.8-0.9	1.5-3.0	0.2-0.6	0.2-0.6	1	Ľ
AJ115B	3.0-5.0	3.5-5.0	ı	0.2-0.6	1 .	
АЛ16В	3.0-5.0	2.0-4.0	1	0.2-0.5	Zinc 2.0-4.0	
AJ117B	3.0-5.0	1.5-3.5	Ī	0.2-0.6	Zinc 4.0-7.0	
AJ118B	1.5-2.5	7.5-9.5	l	0.3-0.8	Iron 1.0-1.8	
ВИ-11-3	0.8-1.5	ı	10.5-13.0	İ	Beryllium 0.07 Titanium 0.07	

Note. The letter B at the end of the alloy grade designation indicates that the castings are manufactured from pig aluminium foundry alloys.

Table 105

Time Info	erval During	Time Interval During Which Ductility of Aluminium Wrought Alloys Is Preserved After Hardening	Wrought Alloys Is Pr	eserved After Hardening
Grade	Grade of alloy	Time, hrs	Grade of alloy	Time, hrs
Д1 Д6 Д16 АК4 АК4-1	6 4.4 4-1	2-3 2-3 2-3	AB AK6 AK8 B95	6-8-8-8-8-8-8-8-8-8-8-8-8-8-8-8-8-8-8-8
Convention	onal Designati	Table . Conventional Designations, Varieties and Uses of Heat-treatments of Aluminium Foundry Alloys	at-treatments of Alum	Table 106 inium Foundry Alloys
Designation		Heat-treatment		Use
T1 T2	Ageing at ter Annealing at	Ageing at temperatures up to 200° C Annealing at temperatures up to 300° C	For light-duty par Dimensional stab	For light-duty parts rapidly cooled when cast Dimensional stabilisation of parts and re-
T4 T5 T6	Hardening ar Hardening ar Hardening ar	Hardening Hardening and partial ageing Hardening and full ageing to maximum hard-		moval of stresses Increase in ductility Increase in strength and yield point Attainment of both maximum strength and
17	Hardening and stabilis	Hardening and stabilising annealing at temper- atures above 200°C		For parts working at elevated temperatures
T8	Hardening and softeni atures above 300° C	Hardening and softening annealing at temper- atures above 300° C		For small parts requiring increased ductility with high magnesium content in the alloy

Table 107

Recommended Heat-treatment Schedules for Aluminium Foundry Alloys Treated in Air Atmosphere Furnaces

meine Machanical aconoction	ageing mechanical properties	Tensile strength, kg/mm². Flonga- flonga- tion, % Brinell instdness	30	0.0	se to 16 4 50		17 1 70	0 12 - 65	0 21 - 75	0 20 1 70	0 18 2 65	0 20 1.5 70	0 23 3 70	0 16 - 65	0 20 0.5 70	0 18 1 65
tomposition and	rempering and	Holding time, hrs	4: V P 6		2-4 Furnace to	_	5 Air	2-4 Ditto	5 Dit	5 Dit	3-5 Dit	5-15 Dit	15 Ditto	5-10 Dit	5 Dit	5 Dit
Appealing	Ameaning,	Heating tempera- ture, °C	910 930	210-230	280-300		175-185	280-300	170-180	225-235	325-335	170-180	170-180	175-185	175-185	225-235
A Luitace	nardening	Cooling muibəm	Water 50 100° C 22	water 50-100 C of	1		1	ı	Water 50-100° C	Water 50-100°C	Water 50-100° C.	1	Water 50-100° C	1	Water 50-100° C	Water 50-100°C
	паг	Holding and, amit	•	4-7	1		}	1	5-12	4-6	4-6	1	5-8	1	2-8	2-8
		Heating tempera- D° ,97ut	210 690	026-016	1		1	!	510-520	510-520	510-520	1	530-540	ı	520-530	520-530
I	и	Convention designatio of heat- treatment	E	c T	T2		ΙΙ	T2	T6	T7	T8	T	T6	Tı	T6	T7
		Grade of alloy	-	AJII	АЛ2		АЛЗ					A.Л.4		АЛБ		

Table 107 continued

Mechanical properties	Brinell hardness number		75
noa '	-FBonga- fion, %	1000 646	4
Mechanical propert	Tensile strengih, kg/mm²	20 20 20 20 20 20 20 20 20 20 20 20 20 2	25
and ageing	Cooling muibem	Air Air Air	1
tempering	Holding and ,smil	33-6	1
Annealing,	Healing tempera- Jo, enut	280-300 145-155 145-155 175-185 177-180 280-300	
Hardening	BulfooD muibəm	Water 50-100° C Water 50-100° C Water 50-100° C Water 50-100° C Water 50-100° C Water 50-100° C Water 50-100° C	 Not applied Not applied Not ater 20-100° C
Ha	Holding time, hrs	10-15 10-15 10-15 10-15 10-15 5-12 5-12 5-12 7-6 Not	 Not 10-24
	Heating tempera- ture, C	520 - 530 520 - 530 520 - 530 430 - 440 530 - 540 530 - 545	420-430
Is	Convention designation of heat- treatment-	172 175 176 176 175 175 175 175 175	T4
	Grade of alloy	АЛ6 АЛ7 АЛ8 АЛ10 АЛ11 АЛ11 АЛ12 АЛ12 АЛ13 АЛ13 АЛ15В	АЛ18В ВИ-11-3

Note. The mechanical properties of heat-treated alloys, with the exception of AJ3 and AJ12, refer to sand castings. Strength characteristics of chill mould castings are superior to those above.

Chemical Composition of Titanium Alloys

								565				
		Ba	Basic components,	ints, %					Addition	Additions, % (max)	lax)	!
Grade of alloy	Al	ზ	Mo	Mn	Λ	Ë	Fe	Si	υ	z	0	Н
ВТЗ	4-6.2	2-3	I	1	l		0.8	9.0	0.1	0.05	0.2	0.015
BT3-1	4-6.2	1.5-2.5	1-2.8	1	I		8.0	0.4	0.1	0.05	0.5	0.015
BT4	4-5	l		1-2	I		0.3	0.15	0.02	0.05	0.15	0.015
OT4	2-3.5	l	1	1-2	1	90	0.4	0.15	0.1	0.02	0.15	0.015
OT4-1	1-2.5	1		0.8-2.0	l	nslad	0.4	0.15	0.1	0.05	0.15	0.015
BT5	4-5.5	ı	[I	1	ЭцТ	0.3	0.15	0.05	0.04	0.15	0.015
BT5-1	4-5.5	1	1	2-3	l		0.3	0.1	0.1	0.05	0.2	0.01
BT6	5-6.5	ſ		1	3.5-4.5		0.3	0.15	0.05	0.04	0.15	0.015
BT8	5.8-6.8	.1 :	2.8-3.8	1.	ı		0.4	0.35	0.1	0.02	0.5	0.01
_					_							

is at Desparties of Titanium Allove

Properties Units BT3 BT3-1 BT4 OT4 BT5 BT6 BT8		Mechanical and Physical Properties of Litanium Alloys	Physica	l Proper	ties of Tit	anium A	lloys		
	Properties	Units	BT3	BT3-1	BT4	OT4	BT5	BT6	BT8
Tensile strength kg/mm² kg/mm² 85-115 95-120 80-96 90-100 105-118 Yield point Limit of proportionality kg/mm² kg/mm² 85-105 80-96 55-65 70-85 80-90 95-110 Relative elongation of area kgm/cm² 96 25 26	Tensile strength Yield point Limit of proportionality Relative elongation Relative reduction of area Resilience Hardness Modulus of elasticity Modulus of shear Poisson's ratio Shearing strength Specific gravity Electric resistance Coefficient of linear expansion (×10°) Heat conductivity	kg/mm² kg/mm² kg/mm² % kgm/cm² kg/mm² kg/mm² cal/cm·sec·°C	95-115 85-105 70-80 10-16 25-40 3-3-6* 11,000 4,250 0.3 65-70 4.46 1.58 8.4	95-120 85-110 70-85 10-16 25-40 3-6* 11,500 4,300 0.3 4.5 1.36 8.6	80-90 70-80 50-60 115-22 20-30 11,000 4,200 0.31 4.6 	70-85 55-65 15-40 25-55 60-70** 4,000 4,000 4.55	80-95 70-85 65-80 112-25 30-45 30-45 10,400 4,250 0,3 65 65 1.08 1.08	90-100 80-90 70-80 8-13 30-45 4-8 11,300 11,300 11,6 8.41 0.018	105-118 95-110 75-85 30-15 30-55 3-6 11,000 4,250 0.3 65-70 4.47 1.61 8.4

* Rockwell scale C; ** Rockwell scale B; *** Brinell hardness number.

Chapter IX

FORGING OF FERROUS AND NON-FERROUS METALS

The present chapter considers heating, forging and cooling of blanks from ferrous and non-ferrous rolled stock.

1. FORGING OF STEEL

Heating rate. The heating of blanks prior to forging is carried out, as a rule, at a maximum rate (induction heating included). Rapid heating increases furnace throughput and reduces scaling. Slow preforging heating is employed for steels with low heat conductivity, namely:

1. For medium- and high-alloy steels greater than 50 mm in crosssection. Steels with total content of alloying elements exceeding 2.5% are regarded as medium-alloy steels, while high-alloy steels are those which contain over 10% of alloying elements.

2 For alloy steels with increased carbon content and high-carbon

steels over 160 mm in cross-section.

3. For steels in stressed condition. Slow heating is generally ensured by charging the blanks in the lowest temperature zone, usually near the furnace door, and then by pushing it gradually towards the high temperature zone. Under all conditions heating should be uniform, excluding pronounced local overheating of the blanks. Charging in bulk is to be avoided and uniform distribution of the charge throughout the furnace ensured.

Forging temperature ranges and cooling conditions applicable to blanks are presented in Table 110. Steels attain best ductility in the

temperature ranges indicated.

Too high temperatures at the beginning of forging cause overheating and burning. Completion of forging at temperatures considerably in excess of those mentioned in the table provokes grain growth and formation of cementite lattice and embrittles the steel. When forging is concluded at temperatures below those indicated in the table, steel be-

comes strain-hardened and cracks are possible.

Cooling rates indicated in the table ensure the best structure of the forgings. The formation of cementite lattice may take place even when high-carbon steel is very slowly cooled from normal temperatures. Therefore, these steels are cooled in the air down to 700° C, heating colour being as a guide, and then at a slower rate. Practically, this method is employed for forgings with thin sides, as well as with abrupt changes in section in which air-cooling from beginning to end may result in cracking.

Table 110

Approximate Temperatures for the Beginning and Completion of Forging and Cooling Schedules for Steel Blanks

	Тетрега	ture, °C	Cooling	schedules for the	Cooling schedules for the below-mentioned o	dimensions
Grade of steel	Beginning of forging	Comple- tlon of forging	Up to 50 mm	51-100 mm	101-200 тт	201-300 mm

Carbon Structural Steels

In the air Ditto	Ditto Ditto Ditto Ditto Ditto In the air In the pile In the pit Ditto Ditto
700	750 800 800 800 800 800
1300	1250 1250 1230 1200 1200
Cr.0, Cr.1, Cr.2, Cr.3, 10, 15 Cr.4, Cr.5, 20, 25, 30, 35 Cr.6, Cr.7, 40, 45, 50	55, 60 15F 20F, 30F, 10F2 40F, 50F, 60F, 65F 30F, 35F2, 40F2 45F2, 50F2

Alloy Structural Steels

In the pit	-
In the air	Ditto In the pile In the pit
In the air	air n the pile
	In the air
800	888 800 800
1220 1220	1200 1250 1230
	X, 45X,
15X, 20X 30X	35X, 38XA, 40X, 50X, 20X3 15XΦ, 20XΦ 40XΦ, 50XΦ

the air In the air the air the air In the pit In the pit In the pit In the pit In the pit In the furnace the air Ditte	In the pile In the pit In the furnace the air In the pit Ditto	Ditto Ditto pile In the pile In the pit In the furnace In the pile In the pile In the pile In the pit In the air In the pile In the pit In the air In the pile In the pit In the air In the pile In the pit air In the pile In the pit In the furnace air In the air Ditto air In the pit In the pit air In the pile In the pit Ditto		
In the In	In the air In the In	In the air In the pile In the In the In the In In In In In In In In In In In In In	In the air In the air In the air	In the pit In the pii
800 800 820 820	88888888 800 000 800 000	800 220 220 8820 8820 8820 8820 8820	800 800 850 850 850 850	820 820 850 850
1260 1180 1200 1180	1150 1170 1180 1240 1240	1220 1180 1220 1220 1180 1180	1160 1160 1160 1160	1200 1180 1180 1180
12M, 15M, 20M 330M, 30XMA 12XM, 15XM, 20XM 35XM, 35X2M 33XC, 37XC, 40XC	40XF, 50XF 35XF2 18XFT, 18XFM 40XFM, 55C2, 60C2 27CF 35CF	20XFC, 25XFC 20XFC, 35XFC 30XFCHA 25H 30H 20XH 40XH, 45XH, 50XH 12XH2, 12XH3	37X H3A 12X2H4 20X2H4 35X ЮA, 38X MЮA 35X MΦA 25X2MΦA	13X HBA 18X HBA, 18X HMA 25X HBA 12X2H3MA, 18X2H4MA

Table 110 continued

	Temperature,	ıture, °C	Cooling schedules for the below-mentioned dimensions
Grade of steel	Beginning of forging	Comple- tion of forging	Up to 50 mm 51-100 mm 101-200 mm 201-300 mm
33XH3MA 40XHM 30XH2MФA, 45XHMФA 13H2A, 13H5A, 21H5A	1180 1150 1180 1180	850 850 850 800	In the pit In the pile In the pit In the furnace In the pit In the pit In the furnace In the air In the pile In the pit In the furnace
		7	Ball-bearing Steel
шх6, шх9, шх12, шх15 шх15СГ	1110	850	Cooling in the air to 700°C, then in the pit
		21	Wear-resistant Steels
F13	1150	006	In the air
		Heat-re	Heat-resisting Stainless Steels
1X13 2X13 3X13, 4X13	1150 1150 1120	820 820 800	In the furnace Ditto Ditto

In the air In the iurnace In the air Ditto Ditto Ditto Ditto Ditto Ditto Ditto Ditto Ditto Ditto Ditto Ditto Ditto Ditto Ditto Ditto Ditto In the furnace Ditto In the air	Carbon Tool Steel In the air Ditto Cooling in the air to 700°C, then in the pit Ditto Ditto Ditto Ditto
820 820 820 820 820 820 820 820 820 820	800 800 820 820 820 820
100 1050 1050 1200 1200 1180 1150 1150	1120 1120 1100 1100 1080
X17, X25, X28 9X18 0X18H10, 1X18H9, 2X18H9 X23H13 X23H18 4X9C2 X5M 3X13H7C2 4X18H25C2 1X18H25C2 1X18H25C2 1X18H115 1X17H2 X6CM, X7CM, 4X14H14B2M X14H14B2M	V7, V7A V8, V8A V9, V9A V10, V10A V11, V11A, V12, V12A V13, V13A

Table 110 confinued

	Temperature,	ture, °C	Cooling schedules for the below-mentloned dimensions	dimensions
Grade of steel	Beginning of forging	Comple- tion of forging	Up to 50 mm 51-100 mm	201-300 mm
			Alloy Tool Steel	
7X3	1 1150	820	Cooling in the air to 700°C, then in the pit	ı the pit
8X3	1110	820	Ditto	
60X, X, X6	1120	800	Ditto	
XO5	1120	830	Ditto	
X12, X12M	1100	860	In the pits or in the furnace	6)
Х12Ф1	1120	860	Ditto	
Ф	1120	840	Cooling in the air to 700°C, then in the pit	n the pit
B1	1120	800	Ditto	
ΧΓ	1100	830	Ditto	
XFC	1080	830	Ditto	
4XC	1160	800	In the piles In the pits	In the furnace
6XC	1160	820	Ditto	
9XC	1120	840	Cooling in the air to 700°C, then in the pit	n the pit
8ХФ	1100	860	Ditto	

85XΦ	1100	860	
4XB2C, 5XB2C	1140	800	In the piles In the pits In the furnace
6XB2C	1150	820	Ditto
5XBF	1160	800	Ditto
9XBL	1120	820	Cooling in the air to 700°C, then in the pit
XBL	1100	830	Ditto
3X2B8	1160	820	Ditto
4X8B2	1160	820	Ditto
XB5	1100	850	Ditto
5XHM	1200	820	In the piles In the pits In the furnace
5X FM	1180	800	Ditto
5XHT, 5XHB	1180	820	Ditto
5XHCB, 5XHC	1180	870	Ditto
_			
		7	High-speed Steels
P18, P9, P9Ф5, P10K5Ф5, P9K10, P14Ф4	1180	096-006	In the pits or in the furnace
P18K5, P18K0, P18Φ2, P18K5Φ2	1200	006	Ditto

Large and simple-shaped forgings from the steels mentioned above can be air-cooled to room temperature. To avoid cracks, no water is to be splashed on the hot forgings and, furthermore, the latter should never be thrown on a moist floor or metallic plates, or cooled in the draught. The higher the content of carbon and alloying elements in the steel, the greater care should be taken in dealing with the cooling schedule. Cooling pits should be filled with sand, ash or cinder. It is recommended to cool high-alloy steel forgings in pits with warmed-up sand or cinder; furthermore, each forging should be covered up separately.

Distinctive forging features of tool and high-alloy structural steels. To obtain most favourable structural characteristics and best mechanical properties, the steel should be forged from all sides, this being of great importance for high-alloy and high-speed steels larger than 50 mm in section, in order to disintegrate the carbides and to distrib-

ute them evenly over the entire forging.

The forging procedure should be started with rapid light strokes. As the work cools, the force of the strokes should be increased, the frequency remaining unchanged. Light strokes at the end of forging may cause cracks; at the same time, care should be taken to avoid overheating the steel which may result from too heavy blows. A correct forging schedule will ensure a through deformation of the forging and its slow cooling during the hot-working. Excessive feed of blanks per blow of hammer should be avoided to prevent internal cracks. No sharp angles should be allowed to appear in forging manipulations; if these angles appear they should immediately be hammered in because subsequent hammering may cause cracks at the cooled angles.

Table 111 lists forging defects of steel and their correction.

2. FORGING OF NON-FERROUS METALS

Table 112 presents the temperatures of hot working of copper and

copper alloys.

Table 113 lists temperature ranges for forging aluminium alloys liable to plastic working. Alloys of grades Д6 and Д16 are not subject to forging. The alloys mentioned are forged only in exceptional cases,

the utmost attention being given to the forging schedule.

All tools used for the forging of aluminium alloys, namely: strikers, dies, etc., should be preheated to 250-300°C. Forgings are to be cooled in a heated pit. It is not recommended to throw them on a cold floor. To avoid sticking of the alloys to the die, the surface of the latter is oiled with a lubricant consisting of 15-20% graphite and 85-80% oil (or fuel oil).

Table 114 presents temperatures ranges for the hot working of nickel

and its alloys.

Table 115 presents temperature ranges for the forging of magnesium

alloys subject to plastic working.

All tools used for the forging and stamping of magnesium alloys should be heated to 250-320° C. Forging and stamping should be effected by means of hydraulic and steam hydraulic presses, ensuring low deformation rates.

Steel Forging Defects

Cause	Correction and prevention
Oxygen contained in fur- nace gases combines with iron	Furnace atmosphere control. Rapid heating of the metal and normal forging temperature range. In calculations the loss of metal through oxidation and waste is assumed to equal 2%
Furnace gas oxygen com- bines with carbon. De- carburisation is pro- moted by silicon, tung- sten, vanadium, mo- lybdenum, contained in steel	Ditto
Rapid cooling of the metal during and after forging	Local rapid cooling of forgings is to be avoided. Hammer in sharp angles, preheat the forging in time, avoid cooling in draughts, on damp floors, etc. Fine cracks should immediately be cut out
Poor-qualify metal, in- sufficient ductility of the blank core. Weak forging, excessive feed per hammer blow	Heat the blank through. Forge the piece quickly and vigorously. Feed- ing rate should aver- age 3/4 of the diame- ter or side of the cross- section of the forged blank
Forging completed at too high a temperature. Slow cooling of hypereutectoid steels from forging temperatures	Strict compliance to forg- ing temperature range. Correction: normalising from Ac _m + 20-30°C, then annealing accord- ing to schedules usual for the steels concerned
	Oxygen contained in furnace gases combines with iron Furnace gas oxygen combines with carbon. Decarburisation is promoted by silicon, tungsten, vanadium, molybdenum, contained in steel Rapid cooling of the metal during and after forging Poor-qualify metal, insufficient ductility of the blank core. Weak forging, excessive feed per hammer blow Forging completed at too high a temperature. Slow cooling of hypereutectoid steels from

Defect	Cause	Correction and prevention
Flaky fracture characteristic of high-speed steels	steel completed at too	Strict compliance to forg- ing temperature sched- ule
Cracking of the blank	Burning of the metal	Control of forging temperature and holding time

Table 112

Temperature Ranges for the Hot Working of Copper and Copper Alloys

Grade	Temperature range, °C	Grade	Temperature range, °C
Copper Brasses Л96 Л90 Л80 Л68 Л62 ЛС59 ЛА77-2 ЛАН 59-3-2 ЛЖМц 59-1-1 ЛК 80-3 ЛМц 58-2 ЛН 65-5 ЛО 90-1 ЛО 70-1	900-800 850-775 900-850 870-820 850-750 850-750 820-730 770-720 750-700 730-680 850-750 730-680 850-750 900-850 900-850	ЛС 60-1 ЛС 59-1 Вronzes Бр.ОФ6.5-0.4 Бр.ОФ4-0.25 Бр.ОЦ4-3 Бр.А5 Бр.А7 Бр.АЖ9-4 Бр.АЖМц10-3-1.5 Бр.АЖН10-4-4 Бр.АМц9-2 Бр.КМц3-1 Бр.КН-1-3	820-780 780-640 770-750 780-750 750 880-830 850-750 825-775 900-850 850-800 850-800 910-890
ЛО 62-1	750 - 700	Бр.Мц5	850-800

Forging temperatures for titanium and its alloys are in the 800-900°C range. Heating prior to forging should be carried out in two stages to reduce scaling and prevent grain growth, first up to 700-750°C, and then to 900°C. Holding time for blanks at high temperatures should be as short as possible, averaging 30 sec at the most per 1 mm of the maximum thickness of the blank.

 ${\it Table~113} \\ {\it Forging~Temperatures~for~Aluminium~Alloys}$

	Temperatu	re, °C
Grade	Beginning of forging	End of forging
АК2 АК4 АК6 АК8 В95 Д1 Д6 Д16	470 470 470 460 440 460 450 450	380 380 380 400 380 400 400 400

 ${\it Table~114}$ Hot Working Temperatures for Nickel and Some of Its Alloys

Grade	Description	Temperature range, °C
НТ НМц2.5 НМц5 НМЖМц2.8-2.5-1.5 МНЖМц30-0.8-1 МН19 МН5 МН5 МНА13-3 МНА6-1.5	Commercial nickel Manganese-bearing nickel Ditto Monel metal Nickel silver Ditto Copper-nickel alloy Cunial A Cunial B	1250-1140 1250-1150 1150-1100 1150-975 960-900 1030-980 1000-950 980-750 980-750

Table 115
Forging Temperatures for Magnesium Alloys

	Temperatur	e, °C
Grade -	Beginning of forging	End of forging
MA1 MA2 MA3 MA4 MA5 MA8	410 410 400 350 385 420	260 230 230 300 300 280

Chapter X

MANUFACTURE OF BUILT-UP CUTTING TOOLS

Built-up cutting tools are manufactured by means of butt welding of high-speed and structural steels for the shank tools, as well as by welding or brazing of cutting tips to the holders of single- or multi-point tools. Steels of grades 40, 45, 40X, 45X are usually used when manufacturing holders and shanks. Steel of grade V7 is also used for the holders of fine cutting-off tools.

Butt welding. It is effected in a butt-welding machine. The blanks should be descaled, cleaned from dirt and other deposits which prevent a perfect contact with the grips of the welding machine. Table 116 lists an approximate projection length for blanks, the melting allowance for the welding process included. Difference in the cross sections of welded blanks should not exceed 1 mm. If this difference surpasses 1 mm, then the larger blank should be machined to the diameter of the smaller one, on a length equal to approximately 3 or 4 welding allowances. The transition from the larger size to the neck should be filleted with a radius not less than 1-2 mm.

Table 116

Approximate Data on Projection Lengths and Allowances for Butt Welding

	Projection length, mm		Welding all	owance, mm
Blank diameter	High-speed steel	Structural steel	High-speed steel	Structural steel
Up to 10 10-20 21-25 26-30 31-35 36-40 41-46 46-50	6-8 8-10 10-12 12-14 14-16 16-18 18-20 20-22	10-12 12-15 15-18 18-22 22-24 24-28 28-32 32-38	3 4 4 4 4 5 5	2 2 2 3 3 3 3 3 3

Whenever it is necessary to weld blanks larger than those specified by the welding machine rating, the articles are recessed in the butts (Fig. 31) so as to reduce the cross-section, and then welded. An opening

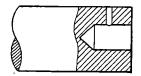


Fig. 31. Large-section blank machined in order to be welded on a machine of insufficient rating

3-5 mm in diameter should be provided in the recessed part to ensure the evacuation of gases.

Table 117
Dimensions of Shank Tools with Section
Recessed for Butt Welding

Tool	Diameter of blanks, mm	Diameter of recessed section, mm
End milling cutter Drills, reamers, counter-bores	30-35 35-45 45-55 55-65 65-75 75-85 30-35 35-45 45-55 55-65 65-75 75-85	14-16 16-20 20-25 25-30 30-35 35-40 14-20 20-28 28-35 35-42 42-50 50-56

Table 117 presents recess diameters for various types and sizes of shank tools. In order to avoid cracks, the welded blanks are charged to a furnace at 730-760° C. Following charging the blanks are held over till the blanks charged last are heated through; after this, the blanks are removed to the air or cooled together with the furnace.

For structural steel, the quality of the welding is checked after annealing by knocking the blank twice, at a distance 5-8 mm from the welding seam, against the angle of a massive metallic plate, turning the blank 90° after the first blow. For high-speed steels, annealing is checked by means of the Rockwell hardness test at 2-4 mm from the seam. The hardness number should not exceed $R_C = 25$.

Tips from high-speed steel are welded by the use of welding compounds, the composition of which is presented in Table 118. Welding compounds are prepared by grinding ferromanganese and other components to fine powder, sieved through a 0.2-0.3 mm mesh. The welded faces of the holder and tip should be machined but not ground to ensure the strength of the joint. The size of the tips should approximate that of the holder seat. If the tips are well trimmed, the work can be hardened immediately following welding, use being made of the welding heat. If the size of the tip does not fit the seat (particularly, when the tip is larger than the seat), burning of the tool may occur, and, therefore, the tools concerned should be hardened after sharpening.

Table 118

Composition of Welding Compounds for Welding
High-speed Steel Tips

Application of compound	Ferroman- ganese, Mn I	Borax or glass	Copper
 For tools subject to sharpening only For tools in which the welding seam is subject 	67-80	33-20	_
to various methods of machining	60-65	15 -2 0	15-20

Brazing of carbide tips. The brazing of carbide tips is effected by the use of hard solders. Table 119 presents the composition of solders, recommended by a research institute VNII, as well as their applications. In shop practice, whenever the manufacture of complex solders meets with difficulties, brass is used for brazing multi-point tools while copper is employed for various other tools. The solder is used in the form of foil, thin wire, fragments cut from sheet or band, chips or filings. Fluxes, the composition of which is given in Table 120, are employed to prevent oxidising of the welded surfaces.

Commercial borax is generally employed when being calcined at 800°C prior to use and ground to fine powder. Calcined borax is to be kept in a well-plugged recipient so as to avoid air moisture absorption. Apart from powders, aqueous water solutions of fluxes are also used. They are prepared by dissolving 1 kg of the flux in 3-5 litres of water. The tools (chiefly with the slots for the tips being coated) are boiled

for several minutes in such a solution prior to brazing.

The welded faces of the holder and tip should fit one another snugly. The clearance between the holder and the tip should not exceed 0.05 mm, while that between the sides of the slots in a multi-point tool and the tip should not exceed 0.15 mm. The tool tips are brazed in forges, furnaces, baths, resistance-welding machines and induced-heat installations. Brazing in an induced-heat installation is a highly productive operation, which is, moreover, noted for its up-to-date technical level.

Soldering Compounds Recommended for the Manufacture of Carbide Tip Tools and Their Applications

					mpositio conter			
-Soldering compound	Grade	Соррег	Nickel	Manga- nese	Other	Zinc	Melting point, ⁶ C	Recommended application
Copper Brass	М1 Л62	99.9 61.5	_	-			1083 905	Furnace brazing Electrical brazing of light-duty tools of the BK
Nickel brass	ЛН 58-5	58.5	5.0	_	<u></u> .	псе	850	duty tools of the
Manga- nese brass	ЛМц 58-5	58.5	_	4.25	Iron 0.75	e balance	850	duty tools of the
Nickel- manga- nese brass	ЛНМц 56-5-5	56.5	4.5	4.5	Ditto	The	900	TK group Brazing of heavy- duty tools of the TK group
Alumini- um- nickel- manga- nese bronze	Бр. АНМц 5-3-2	90.5	3.0	1.5	Alu- mini- um 5.0		1100	

Fig. 32 shows inductors used for the brazing of cutting tools, as well as for the multi-point tools, each tip being brazed separately. This

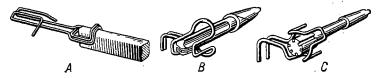


Fig. 32. Inductors for heating by induced heat: a- for brazing and welding of tips on to cutter holders; b- for brazing of drills; c- for brazing of reamers

method cannot be employed for gang milling cutters with negligible clearance between the tips, as tips, next to those soldered, will heat to

a high temperature, thus causing spoilage of the work. A method for simultaneous brazing of carbide tip gang milling cutters in induced-heat installations is described below.

Table 120
Fluxing Compounds Recommended for Brazing Carbide
Tools and Their Applications

Flux compos	ition	Molting	Recommended application Copper brazing of group BK tips		
Component	Amount in °, _e by weight	Melting point, °C			
Commercial borax	100	741			
Commercial borax Boric acid	90 10	€00-€50	Brazing of group BK tips with low-meltingsolders		
Commercial borax Potassium fluoride	70 30	800 - 850	Brazing of group TK tips with all solders, as well as brazing of car- bide alloys onto chro- mium steels		

A groove is made in a refractory brick, to the dimensions of the bottom inductor, with the aid of a cape chisel (Fig. 33). The brick 4, with the inductor 1 inserted in the groove, is placed on the table, fixed in a tapered bushing of the bottom part of the hydro-hoist of the Γ 3-46 apparatus (or on any other similar table mounted on any other type of apparatus).

The brick is covered with sheet asbestos 3, the milling cutter (which is "boiled" in a flux solution prior to the operation) is placed atop and the inductor is seated over the latter; after this the two inductors are

clamped in the holder.

While clamping the upper inductor 2 should be slightly lifted, so as to avoid contact with the cutter. The bottom coil of the inductor 2 should clear the cutter by 2-3 mm. It is intended to heat the body of the milling cutter directly at the cutting tips. At the same time the latter are also heated through heat conduction.

The upper coil of the inductor 2 is situated 8-12 mm above the tips. Its destination is to ensure, in conjunction with the inductor 1, the initial heating of the body of the milling cutter and the tips and to melt

the solder.

The rate of heating can be controlled by varying the spacing between the cutter and the inductor coils. For the bottom inductor, the said distance is adjusted by varying the thickness of the asbestos pad. The brazing technique for multi-point cutter is simple enough:

1. The cutter is placed between the clamped inductors on the asbestos pad. To facilitate the operation, the upper inductor is forced slightly upwards.

2. The flux is spread additionally around the tips with the aid of

a spatula.

3. Current is switched on and the cutter is heated through to approximately 800-900° C, after which heating is cut out and solder is added by means of tweezers. The hard solder is in the form of two half-rings fashioned to the diameter of the cutter (Fig. 33).

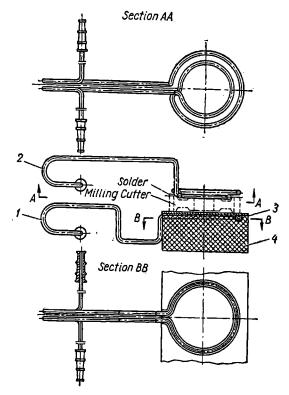


Fig. 33. Device for simultaneous brazing of multi-blade milling cutters with carbide tips on induced heat apparatus

4. Heating is switched on again and the cutter is brazed. The brazing of a 150 mm dia and 15-mm thick milling cutter with 16 tips takes 2-2.5 minutes. The solder fills the gaps entirely. Table 121 describes tool brazing defects and measures which should be taken to prevent them,

Defects in the Brazing of Built-up Tools and Their Prevention

Defects	Cause	Prevention							
Manufacture of Brazed and Welded Tools									
The solder does not flow in the gaps	sides of the groove and	When assembling the tools the gaps should be limited to 0.05-0.15 mm. No continuity of calking is necessary Normal heating temperatures and sufficient holding time should be ensured.							
The solder forms balls on melting	Presence of oxides on the brazed surfaces Soldered surfaces have been ground or lapped	Sufficient amounts of flux should be fed The soldered faces should not be ground or lapped							
Voids under the tips	1. Maladjustment of tips and holders 2. Switching on and off of current after the melting of solder 3. Premature pressing of the tip	tures should be up to the recommended values 2. After the hard solder has started melting, the current should not be switched off till the termination of brazing							
Blackening of the seam	Insufficient feed of flux Insufficient heating	Sufficient amounts of flux should be fed Normal heating temperatures and sufficient holding time should be ensured							

Defect	Cause	Prevention
Burning of the holder at the point of con- tact with the electrode	Contact surface is too small Weak contact High degree of heating and prolonged holding	Contact area should be equal to the product of the holder width at the point of contact by the thickness of the tip Contact pressure on the holder should be increased Heating temperature and holding time should be decreased
Cracks in carbide tips	Rapid cooling after brazing Contact between the electrode and the tip Non-uniform heating of tip because of great thickness	Slow cooling after brazing should be ensured Electrode should be lowered to prevent contact with the tip Reduce the thickness of the tip
	Manufacture of Butt-weld	ded Tools
Incomplete pen- etration	Too low heating temperature Insufficient upsetting strength	heated till the butts melt
Voids	Insufficient upsetting be- cause of too low ma- chine power rating	The work is to be welded in a machine of ade- quate rating or the blanks should be re- cessed according to Ta- ble 115
Annular cracks in high-speed blanks at 1-3 mm from the welded seam	Local welding stresses are caused by too rapid cooling of the welded blank	After welding the blank should be immediately charged to a furnace and then, after holding, cooled with the furnace

Table 121 continued

Defect	Cause		Prevention
Overheating	Excessive heating blanks	of the	1. Keep strictly to the projection lengths recommended in Table 114 2. Follow strictly the heating schedule

Chapter XI

EQUIPMENT, FUEL AND ACCESSORY MATERIALS FOR HEAT-TREATMENT SHOPS

I. FURNACES

According to the fuel used, furnaces can be of the solid-fired, liquid-fired, gas-fired or electrically heated type.

Solid-fuel furnaces are used only when other types of furnaces are

not available.

Table 122 presents technical data on electric furnaces.

Metallic resistors (Table 123) are used for furnaces with working temperatures up to 1100° C, while carborundum resistors (Table 124) are

used for those with higher temperatures.

In three-electrode $\overline{\text{CII-2}}$ and $\overline{\text{CII-3}}$ bath-type, salt furnaces, the electrodes are set on the walls of the working space at approximately 120° to each other. On heating, the current passes through the heated work and, as the resistance of the metal is lower than that of the salt, overheating of the article can occur, especially if the latter is nearer than 25-30 mm to the electrodes.

Bath-type C-20, C-25 and C-45 furnaces with submerged electrodes are provided with one pair of electrodes each, the latter being situated near the wall at 12 to 25 mm from each other. With this arrangement of electrodes, the current does not flow through the work and there is no danger that the latter would be overheated. The fact that the electrodes are spaced at a short distance from each other gives rise to powerful electromagnetic fluxes, which mix the salt and equalise its temperature. Large C-50 and C-100 bath-type furnaces are provided with

three pairs of electrodes each situated near the back wall.

Another advantage of the bath-type furnaces with submerged electrodes is the possibility of changing the electrodes during the run of the furnace (hot reconditioning). After relining, bath-type furnaces can best be dried with the aid of electric heaters. The working space, where the heater is placed, should be covered with asbestos. Drying is continued till the furnace shell heats to approximately 50-60°C; drying time for the CII-2 furnace averages 30-40 hours. When drying by wood or coal, air should be fed to the bottom part of the working space to ensure adequate combustion; otherwise, the fuel will smoulder with ensuing poor drying of the furnace and possible eruption of the salt on its initial melting. Steels of the heat-resisting grade X23H13.

TO RATINGS OF ELECTRIC FURNACES USED FOR HEAT-TREATMENT OF METALS

	Comments			
٠,	Maximum throughpu Kgjhr			
work-	Maximum ing tempe C° taruf			
14 th	Total weignot to to the total single for the total			
slons,	Height			
Il dimen mm	dignal			
Overa	41P!M			
асе	JdgisH			
king sp.	Length			
Wor	Width or diame- ter			
. 3u	Power rati			
	type			
	Furnace			

Chamber Furnaces with Metallic Heating Elements

	Furnace with spherical hearth and carriage
	50 125 200 275 350 300
phere	950 950 950 950 860
t m o s	96.2.8
a) For Operation in Air Atmosphere	1,400 1,600 1,440 1,400 1.790 2,020 2,050 2,250 2,000 2,200 2,560 2,450 2,360 2,860 2,500 2,280 5,030 2,440
in A	1,600 1,790 2,250 2,560 5,030
ation	1,100 1,400 2,050 2,200 2,360 2,280
Oper	250 450 500 550 600 600
For	650 1,200 1,800 1,820
a)	300 450 600 750 900 970
	15 30 45 60 75 100
	H15 H30 H45 H60 H75 HIII-100A

	Heating elements situated on roof, sidewalls and hearth
b) For Operation in Controlled Atmosphere	30
nı o s p	950
ed At	0.29
1011	760
Cont	820 1,245 1,245
n in	820
ratio	140
. Оре	400 500
b) For	250
	12.8
	H25X50

Heating elements situated on roof, sidewalls and hearth Heating elements situated on roof, sidewalls, hearth, doors Furnace with spherical hearth and push-type carriage	A two-chamber furnace. Data in the numerators refer to the bottom chamber, those in the denominators—to the upper chamber
50 25-35 80 50-70 150 90-120 220 130-140 400-450 230-280 300	ents 50 120 25
1000 1000 1000 860 860	g Elem 1300 1300 1300 1300
0.00 - 1.00 0.00 0.00 0.00 0.00 0.00 0.0	Heating 0.75 2.1 3.6 1.5
790 790 790 790 790 790 790 790 790 790	Carborundum Heating Elements 1,140 1,400 0.75 1300 1,600 1,770 2.1 1300 1,2,100 2,000 3.6 850 1,325 1,800 1.5 1300 1,325 1,800 1.5 1300 1,50
1, 195 1,	Sarborundum 1,140 1,400 1,600 1,770 2,100 2,000 1,325 1,800
870 870 870 1,550 1,650 1,650 1,800 1,900 1,900 1,900	with (
200 200 200 320 320 400 500 615	naces 200 250 400 180
650 450 800 1,000 1,300 1,300 1,700 1,100 1,100	Cham ber Furnaces 250 360 200 300 400 250 325 410 180 255 360 175
300 300 400 400 400 500 500 650 650 850 910	Cham 250 300 520 325 250
112 112 113 118 118 110 110 110	15 30 50 19
H30X65 H30X45 H40X80A H40X80A H50X100 H50X100 H65X130 H65X130 H65X110 H85X110 H85X110	OKB-333C OK5-210 OK5-194A

Table 122 continued

	11,	Maximum throughpu kgiht Onnn e ents		140	125	330	225 265		1001	280 550		50 Single charges in kg 100 150 220 400 500
- ਮ -	Maximum work- ing tempera- ture, °C			950	320	950	1300		029	650	naces	950 950 950 950
	Total weight, metric tons		naces	2.1	0.8 0.0	6.4	5.8	naces	1.5	1.8	ng Fur	2.3 3.6 5.0 7.7
sions,		Helght	ng Fui	1,900	3,290	4,120	2,820 3,550	ng Fur	1,900	2,090 3,040	rburisi	1,980 2,320 2,390 2,760 2,880 3,220
Overall dimensions,	E	Length	lardenı		2,50	2,780	1,500	emperi	1,430	1,540 3,260	type Ca	1,790 1,790 1,970 1,970 2,170 2,170
Overal		Width	Shaft-type Hardening Furnaces	_	1,550	. 07	1,500	Shaft-type Tempering Furnaces	500 1,460 1,430 1,900	650 1,540 1,540 2,090 ,220 2,530 3,260 3,040	Muffle-	1,420 1,420 1,590 1,590 1,770
ace	E	Height	Shaft	800	9,200	2,500	1,470	Shaft	200	650 1,220	Shaft-and Muffle-type Carburising Furnaces	450 600 600 900 900 1,200
rking st	dimensions, mm	րերը	 	1	000	3	300		1	11	Shaft	111111
Mo	dime	Width or diame- ter		450	86	000	88		400	500 950		300 300 450 600 600
	'Bul	Power rat		30	8	38	88		24	36		25 35 60 75 105
		Furnace (ype		III30	11135	0211	Г65 Г95		пн-31	(11H-31D) ПН-32 ПН-34		11.25 11.35 11.60 11.75 11.105

Furnaces
Gas-carburising
Muffleless
Shaft-type

	270 380
50 Single charges 220 500 600	270 380
2225 5225 5225 520 520 520 520 52	270 380
	270 380
1050 1050 1050 1050 1050 1050 650 650 650 650 650 850 850 850 1300 1300 1300	
	875
1.8 3.7 4.0 5.2 1.3 1.3 1.3 1.3 1.3 1.3 1.3 1.3	14.43
2,295 3,325 3,325 3,325 1,000 1,000 1,635 3,795 2,220 2,200 2,200 2,200 2,200 2,200 1,320	2,090
11,865 2,120 2,120 2,270 3 and C 1,900 1,900 1,900 1,900 1,600 1,650 1,650 1,650 1,650	7,285 8,245
450 1,370 1,865 2,295 3,900 2,010 2,120 3,315 3,325 4,200 2,160 2,270 3,870 5,200 2,010 2,120 3,325 4,200 2,160 2,270 3,870 5,200 3,870 1,900 1,900 1,900 1,900 1,900 1,900 1,900 2,220 3,000 1,970 2,200 3,795 1,970 2,200 3,795 1,970 2,200 3,795 1,380 1,160	2,850 7,285
	200
Shaft- 	4, 180 5, 140
300 450 450 450 320 300 200 300 340 375 375	375 575
100 100 100 100 100 100 100 100	130
ШЦН5А ШЦН5А ШЦН65А ШЦН95А ШЦН95А ОКБ-3016 ОКБ-3018 ОКБ-3018 ОКБ-3019 ОКБ-3019 ОКБ-3019 ОКБ-3019 СС-20 СС-25 СС-25 СС-25 СС-45 СС-20 СС-25 СС-36 СС-36	

Table 122 continued

	re.							Capacity 21 litres	capacity 40 miles	Temperature contr	through a rheosta Automatic temperatu	control Automatic temperatu	control Automatic temperatu control
06 06 06	nosphe	111				11			_ saoy	. 1	1	!	1
<i>rnaces</i> 180 830 920	Pusher-type Furnaces for Heating Work in Hydrogen Atmosphere	1500 1750 1400				900		3000	_ ~	1000	1000	1000	250
ling Fu 2.1 5.2 9.0	Hydro	2.7 0.9 1.6				7.5		0.1	U.1 t-treat	485 0.042 1000	485 0.04	566 0.06	700 0.55
Annea 1,750 3,040 3,735	ork in	1,920 1,800 1,800	naces	2,500	1,900	$\begin{vmatrix} 5,940 \\ 6,100 \end{vmatrix}$	SI	620	ı usu in Hea	485	485	999	700
ng and 4,850 4,785 5,700	ting W	5,800 2,540 3,370	Vacuum Furnaces	1,700 2,250	2,26)	$\frac{4,500}{4,670} \begin{vmatrix} 3,740\\4,670 \end{vmatrix}$	Oil Baths	380	basn s	430	492	603	220
Barrel-type Hardening and Annealing Furnaces - 2,750 1,330 4,850 1,750 2.1 180 - 1,200 2,300 4,785 3,040 5.2 830 - 2,000 1,350 5,700 3,735 9.0 920	for Hea	98 1,390 5,800 1,920 620 1,000 2,540 1,800 750 1,250 3,370 1,800	Vacui	1,700	1,750 2,260 1	$ \frac{4,500}{4,670} $	0	280	urnace	445	445	525	585
type H 2,750 1,200 2,000	rnaces			350 450	370	1-1		300 280	tupe F	8	82	95	1
Barrel	уре Fu	1,000		1.1	18	1,090			oratoru	210	210	263	390
400 200 310	usher-t	250 100		300	250	1,100		350	Lab	137	137	175	410
19 30 70		22 22 29		35 35	20.	<u> </u>		6/3	* /o -	11.75	1.75	2.6	1:1
OKB-130C OKB-128C B-70		ЦЭП-214 · ЦЭП-219А ЦЭП-220А	± *	OKB-704 11311-245	10E-11E)I	OK5-744 OK5-745		MB-21 MB-40	01-01-	0-ЦW	MΠ-1	MI1-2	III-0.05
9 -969													

249

Notes. 1. Furnaces equipped with metallic heating elements are fed directly from the 220 or 380 V mains. Electode-equipped furnaces, as well as those with carborundum heating elements, necessitate step-down transformers. 2. Pusher-type T-240B and T-165B furnaces admit two pans each on the width of the working space. 3. Crucibles for alkall baths as well as mixers should be manufactured rather from carbon than stainless steel.

perature perature

perature control 1eostat

etc., as well as of low-carbon grade 10, 15, 20, Cr.2, etc., are used for

the manufacture of the electrodes of bath-type furnaces.

Table 125 shows comparative data presented by the "Frezer" Works and relating to electrode durability in three- and two-electrode bath-type furnaces. The author considers these data as somewhat undervalued. Thus, a C-20 furnace in the plant, the author is connected with, has shown a run of 16 days (a total of 106 hours) at 1220-1260°C between replacements of electrodes.

Of great importance, when consumption of electrodes in two-electrode furnaces is considered, is the furnace start-up method. The elec-

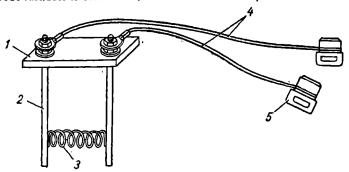


Fig. 34. Coil for melting of salt in salt baths: 1-asbestos-cement; 2-pin; 3-coil from Ø 4 mm, 1-1.6 m wire; 4-MPF wire; 5-contact blade

trodes crumble generally during the start-up of the furnace, the trouble spot being the line of contact of graphite powder with the electrodes. The furnace is heated up at the highest voltage, and if contact between the graphite powder and the electrode is lost, an arc strikes up and destroys the electrode. To reduce electrode wear, the heating-up operation should be carried out at the highest voltage only to the moment when the salt melts between the electrodes, after which graphite should be immediately removed and the voltage lowered. The most efficient method for initial, as well as repeated starting up of a submerged-electrode furnace is with an electric heating coil. After the melting of the salt, the coil is removed and subsequent heating is effected through electrodes. At the end of operation, after the furnace has been switched off, the coil is immersed in the molten salt to a 10-30 mm depth, and the salt solidifies. Repeated starting of the furnace necessitates only the switching-on of the coil current.

Motor generators as well as vacuum-tube oscillators are used to heat up articles by means of high-frequency current.

Tables 126 and 127 present technical data on motor generators and vacuum-tube oscillators manufactured in the Soviet Union.

Motor generators are used for heating up articles larger than 16 mm in diameter, the depth of the hardened layer being not less than 2 mm. Vacuum-tube oscillators are generally employed for heating up minute

Properties of Metallic Resistors

Grade of alloy	Maximum working temperature, °C	Specific resistance in ohm-mm²/m		
X15H60	1000	1.10		
X20H80	1100	1.15		
X20H89T3	1150	1.27		
Х13Ю4	1000	1.26		
0X23Ю5A	1200	1.35		
0Х23Ю5	1200	1.35		
0Х25Ю5	1200	1.40		

Table 124
Chemical Composition and Properties of Carborundum Resistors

		Chen	ical (ompo	ositio		Specific			
Туре	SiC	SiO	С	Al	Fe	· Si	CaO + MgO	Maximum temperature, °C	resistance in ohm mm²m at 20°C	
Si lit Globar	94.4 96.0	3.6 1.5	0.3	0.2	0.6	0.3	0.6	1500 1500	1,000-2,000 930-1,950	

Table 125
Comparative Data on Durability of Electrodes
in Bath-type Furnaces

	Electrode durability in days						
Working temperature, °C	Three-electrode furnaces	Furnaces with sub- merged electrodes					
Up to 600 From 600 to 1000 From 1000 to 1300	90-110 10-13	40-50 18-24 1.5-2					

and odd-shaped parts, the depth of the hardened skin averaging from a fraction of a millimetre to 2-3 mm.

Tubular-type T9H heaters are most efficient for the heating of saltpetre, alkalis, oil and other materials used in heat-treatment technology. Tubular-type electric heaters are curved tubes from carbon or stainless steels, which house insulated electric heating coils. The length and the power rating of the heating unit, as well as the tube material are selected in accordance with the bath volume, heating temperature and the composition of the material to be heated. Table 128 lists approximate data which may be used as a guide for the choice of tubular-type heaters.

The feeding of protective atmosphere to furnaces can be effected by an endothermic gas generator, type OKB-724, having the following rating:

Output 30 m3/hr 30 kW Power rating 220/380 volts Voltage Phases 3/1 Connection of resistors Star/Series Working temperature 1050 - 1100°C Composition of protective atmosphere: 36-40% hydrogen 18-20% carbon dioxide The balance nitrogen Natural gas, town gas, reduced Raw materials propane-butane compounds 6 m³/hr Raw gas consumption Overall dimensions: 1,440 mm width length -2,350 mm 2,500 mm height 2.750 kg Weight

The dimensions of bath-type furnaces working on liquid or gas fuels are standardised (Table 129).

Tables 130 and 131 indicate types of burners which have found widest industrial use.

Approximate furnace throughput rates are presented in Table 132.

2. REFRACTORIES

Firebrick is chiefly used for the building and re-lining of heat-treatment furnaces.

Electric furnace lining is patched with special-shaped refractory brick or with plain firebrick shaped to the configuration desired.

Lutes composed of 25% fireclay and 75% crushed firebrick are used for overhauls.

Data on thermal insulating materials and refractories are given in tables 133 and 134.

3. FUELS

Fuel oil with approximate calorific power of 9,600 kcal/kg is used as liquid fuel.

Various gases are used as gas fuels, the composition and calorific power of which are presented in Table 135.

Technical Data on MI3-type High-frequency Hardening Installations

	מווסושווס מוויים מיולה יויפון היויה מיוים מיוים מיוים מיוים וויים	Jpc mgn-meduci			BIGHT S				
, i , E	Installation equipment	.	.ga/	sdo '	Overal	Overall dimensions.	sions.	suc	tete no,
Type of installation		Designation	Power rati	Frequency	Width W	Length	t flgleH	: Weight, to	Cooling was onsumption in the consumption of the consumption of the consumption of the cooling was a second of the cooling of
MF3-52	High-frequency generator: High-frequency converter Driving motor etartor	FC-50X1/2.5 TIB-50/2500 EH5191, 35 A 9	20	2500	1040	1120	1040	2.1	
	2. Induced-heat hardening in- stallation	37.5075	50	2500	1325	2280	2135	2.0	4.5
MF3-102	1. High-frequency generator: a) High-frequency converter b) Auto transforms doubter	FC-100X1/2.5 FC-100/2500	100	2500	1040	1495	1040	2.5	
y 1	2. Induced-heat hardening installation	3C-100/2.5	100	2500	1325	2280	2135	2.0	4.5
MI 3-108	High-frequency generator: Aligh-frequency converter Alifo-frequency converter Alifo-frequency converter	FC-100X1/8 IIBB-100/8000	001	8000	1150	1685	1110	16.5	-
	2, Induced-heat hardening installation	3C-100/8	100	8000	1325	2280	2135	2.0	4.5

Notes. 1. Type 3C induced heat installation is composed of the following units: hardening, hardening control and pouring and high-frequency contactor.

2. Type C generator comprises (save the converter and the starter): measuring instruments desk, control unit and confactor board.

3. All three installations are supplied either as a complete unit (MF3-52AK, etc.) or in separate racks (MF3-52AB, etc.). The latter difficer from the complete installation in that the high-frequency transformer TBM-2 and the condenser battery are situated in separate racks, which widens the possibility of incorporating the installation in hardening machines and automated productlo lines.

Table 127
Technical Data on Vacuum-tube Oscillator

		ė.,	Overall	ldimensi	ons, mm	ټـ ا	Cooling water
Type of oscillator	Power rating, kW	Frequen- cy, cps	Width	Length	Height	Weight, tons	Cooling water consumption, m ⁹ /hr
ЛГ3-10 A ЛЗ-13 ЛЗ-37 ЛП3-37 ЛЗ-67 ЛП3-67 ЛП3-107 ЛЗ-207	8 10 30 30 60 60 100 200	300 - 450 300 - 450 60 - 74 60 - 74 60 - 74 60 - 74 60 - 74		1,170 1,120 2,200 8,550 2,200 3,400 2,200 2,800	2,150 2,050 2,250 2,250 2,250 2,250 2,250 2,500	0.75 0.85 2.7 3.2 3.1 3.4 3.8 6.0	0.54 1.2 1.75 2.0 2.25 2.5 2.5 2.25 7.0

Table 128.
Technical Data on Tubular-type Electric Heater

Designation	Overall length of the tube, metres	Power rating, kW	Tube material	Application
НММ, НВ, НП	From 0.5 through 6 m, (1! sizes with gradations every 0.5 m)	From 0.5 to 7	Carbon steel	Heating of air, water and oil
нвг	From 2 through 6 m (8 sizes with gradations every 0.5 m)	From 2 to 7	Carbon steel	Heating of alkalis and salt solution
HCЖ, HMЖ, HBЖ	From 2 through 6 m (8 sizes with gra- dations every 0.5 m)	From 2 to 7	Stainless steel	Heating of saltpetre and salt

4. TEMPERATURE CONTROL INSTRUMENTS

Manometric thermometers are used to measure, record and control temperatures in low-annealing bath-type furnaces and isothermal hardening tanks.

hardening tanks.

A manometric thermometer comprises essentially a thermal bulb, a capillary, a tubular spring and indicating or recording devices.

Table 129
Technical Data on Fuel-oil and Gas Bath-type Furnaces

Designa bath-type	tion of furnaces	Crucible di		e , kg	Average throughput.
Gas-fired	Fuel-oil- fired	Diameter	Depth	Crucible weight, kg	kg;hr
птвг-1	птвм-1	200 200	350 535	49 73	20 35
ПТВГ-2	птвм-2		350 535 610	62 86 95	35 50 60
птвг-з	птвм-з		535 610	130 146	70 80
ПТВГ-4	ПТВМ-4		535 610	160 210	100 125

Manometric thermometers are classified as gas thermometers (filled with nitrogen), liquid thermometers (filled with xylene, mercury or methyl alcohol) or steam thermometers (filled with a liquid turning to steam at a definite temperature).

Table 138 lists data on manometric thermometers used in heat-

treatment departments.

Temperatures from -200° to $+500^\circ\text{C}$ are also measured with resistance thermometers known for their very accurate readings. These thermometers are manufactured either from platinum resistance wire, the $9T\Pi$ type, with 11-a and 12-a scales and measurement range from -200° to $+500^\circ\text{C}$, or from copper resistance wire, the 9TM type, with 2-a scale and measurement range from -50° to $+15^\circ\text{C}$. The many types of resistance thermometers available are classified according to design, application and temperature ranges. Thermometers are manufactured with protective sheaths from low-carbon or heat-resistant steels, from 0.5 to 2 m in length. Indications provided by resistance thermometers are read with the aid of measuring instruments (Table 137).

Thermocouples are the most widely used means for measuring temperatures (Table 138). Table 139 lists and describes briefly pyrometer millivoltmeters (galvanometers) operating in conjunction with thermocouples and telescope of radiation pyrometers. Care should be taken that the graduation of thermocouples and galvanometers be similar. Thermocouples are to be connected to measuring instruments by means of compensating leads, or, when the latter are not available, with plain insulated copper conductors.

Compensating leads are manufactured in impregnated cotton sheath—the ΚΠΟ brand, in lead sheath—the ΚΠC brand, in steel wire

Technical Data on Burners

••		lecurical Data on Duriers	419		613					*
		Nozzle		Capacity in kg'hr at air pressure of, min water gauge	hrata!	r pressu	re of. m	ш wate	r gauge	Diameter
Type of burners		diameter, mm	150	300	400	450	9009	009	700	main, mm
One-stage, large-capacity	ФОБ-1	16 20	3.5	3.7	4	11	4.2	11	1 !	38
	ФОБ-2	20 26	473	8.5	7	11	8 12		11	50
	ФОБ-3	45 50	14	21 25	30	1.1	33	11	11	75
	ФОБ-4	, 20 80	32	25 45	41	11	45	11		100
Two-stage, large-capacity	ФДБ-1 ФДБ-2 ФДБ-3 ФДБ-4	50 70 90 90	1111	[]]]	1111	20 30 48 76	22 34 52 82	24 37 56 89	25 40 60 95	50 75 100 125
Low-pressure	A-40 A-30 A-60 A-52	30 40 25 25	1111	11 11 32 32	18 13 44 36	1111	20 14.5 50 41	22 16 55 46	24 17 58 49	100
	A-75 A-95 A-125	75 95 125	111	54 80 135	62 95 148		69 105 170	76 115 185	81 120 205	125 150 200

Thermal Characteristics of Low-pressure Long Flame Gas Burners

Type of burner		Capacity,	Diameter air mai	Diameter of gas and air mains, mm	Pressure in r and veloc	Pressure in mm water gauge and velocity in m/sec	Size of gas-air
			gas main	air main	gas	aîr	
With separate gas and air feeds	ГЩ-1 ГЩ-1½ ГЩ-2 ГЩ-2%	5-15 10-30 20-60 40-100	25 37 50 63	25 37 50 63	50-100 10-15	100-200	22.5×50 21×75 24.5×118 24.5×118
Slot type	ГЩ-3 ГЩ-4 ГЩ-5	60-140 100-200 150-300	75 100 125	75 100 125	\$ 50-100 10-15	100-200	53×120 58×150 68×183
Tangential with turbulent flame		50-100 100-200 150-300 200-400 300-600 500-1000	50 65 85 100 125 150	65 80 100 125 150 170	Not less than Not less 80-100 100-1 10-15 15-2 Mixture velocity 10-12 m/sec	than Not less than 0 100-150 15-20 xture velocity 10-12 m/sec	870 890 8115 8140 8170
With turbulent flame	17.2 17.2 17.3 17.5 1.6	100 160 250 400 630 1000	67 85 106 132 170 212	80 100 125 200 250 250	Not less than 80-100 10-15 Mixture 10-12 10-12	ss than Not less than 100 100-150 150 15-20 Mixture velocity 10-12 m/sec	ø80 ø100 ø125 ø200 ø250

Table 132

Approximate Furnace Throughput Rates (in kg/m²hr)
for Various Heat-treatment Operations

		He	at-treatmen	t furnaces	
Heat-treatment operation	Chamber	Car-type	Pusher- type, continuous	Conveyor, electric	Rotary- hearth
Normalising Annealing Hardening Carburising: solid carburiser gas carburiser Gas cyaniding	120-150 40-60 120-160 8-12	60-100 35-50 60-80 8-12	150-200 50-70 150-200 15-20 40-50 80-100	120-220 	160 - 200 150 - 200 15 - 18
Tempering	90-110	60-80	100-150	100-150	_

Table 133

Data on Thermal Insulating Materials

Material	Maximum temperature of use, °C	Volume weight, kg/dm³
Unburnt diatomite in lumps Unburnt diatomite powder Burnt diatomite powder Diatomite brick Foamy diatomite brick Fluffed asbestos Asbestos milboard Glass wool Glass fibre Foam concrete Boiler cinder	800 900 1000 950 900 500 500 300 600 600 300 800-950	0.68 0.3 0.55 0.55-0.75 0.40-0.55 0.8 0.90-1.2 0.80 0.25-0.30 0.10-0.20 0.40-0.50 0.7-1.0

sheathing—the KIII brand, in flexible sheaths—the KIIIO type, and

also in special-purpose sheaths.

When choosing thermal control instruments, it is recommended to bear in mind the following points. Type ЭПП, ЭПД and СП potentiometers are the most precise instruments, with permissible error not exceeding $\pm 0.5\%$ of the upper temperature range; for the type MПП-054 potentiometer the error does not exceed $\pm 1\%$, while for the rest of the instruments described in Table 138 the accuracy of measurement is within $\pm 1.5\%$. The upshot of this is that preference should be given

Table 134
Refractoriness and Allowable Working Temperatures
of Refractories

Refractory	Refractoriness. °C	Allowable tem- perature range. °C
Firebrick, grade A	1690 ∤ 1900-1950 1850-2000	1300 - 1400 1250 - 1300 1200 - 1250 1600 - 1650 1600 - 1700 1400 - 1500 2000 700 - 750

 ${\it Table~135}$ Mean Chemical Composition and Calorific Power of Gases

			cal composi	tion, pe	r cent		po-
Gas	Carbon dioxide	Сагьоп топохіde	Methane	Heavy hydrocar- bons	Hydrogen	Nitrogen	Calorific wer, kcal/n
Blast-furnace gas Coke gas Lighting gas Generator gas (from coal) . Generator gas (from shale, Estonia) Natural gases: Saratov Dashava Kuibyshev Ukhta	11 2 6 4.5 18 — 0.8 0.2	27 7 16 25 11 0.7 0.2	Up to 1 23 20 2 2 24 94.3 97.8 75.80 93	$ \begin{array}{c c} -\\ 2\\ 10\\ 0.2\\ 6\\ 2.7\\ 0.5\\ 8-10\\ 1.2 \end{array} $	3 58 25 13 39 1.8 0.2	58 8 23 55.3 2 1.2 1.3 11-13 1.2	900 3900 4500 1350 4760 8800 8500 8500 8150

to potentiometers when treating aluminium and magnesium alloys, beryllium bronze, as well as when tempering steel or performing various other heat-treating operations within narrow temperature ranges; in heat-treatments requiring wider temperature ranges, indicating, controlling or recording millivoltmeters (which are considerably cheaper than the potentiometers, yet are as reliable) are to be preferred. The same considerations apply to ratiometers and balanced bridges operating with errors up to $\pm 1.5\%$ and $\pm 0.5\%$ respectively.

List of Manometric Thermometers and Their Brief Characteristics

Instrument	Туре	Purpose	Measuring range, °C	Length of capillary, min
Manometric remote control thermometer, gas-filled	TF-970	Indicating	0-300	20.40 and 60
Ditto ,		Indicating, with	[·	•
	TΓ-410	signalling device Recording, with clock		20.40 and 60
Ditto	TΓ-610	drive Recording, with syn-	0-300	20.40 and 60
-		chronous motor	0-300	20.40 and 60
Regulating and recording	TF-618	Recording, with syn- chronous drive and signalling device	0-300	20.40 and 60
thermometer	04-TΓ -			
	-410	Regulating and record- ing, with clock drive	0-200	_
	04-TΓ- -610	Regulating and re-		
		cording, with syn- chronous drive	0-200	_

Note. Specifications indicate extreme ranges of temperatures measured. Practical ranges are: 0-120, 0-160, 0-200 and 0-300°C

High temperatures can be measured with optical or radiation pyrometers, data on which are listed in Tables 140 and 141.

The radiation pyrometer of type PAΠUP designed to measure surface temperature of heated bodies, 35 to 200 mm in diameter, consists of: (1) type TEPA-50 focussing telescope; (2) type ΠΥΘC-54 resistance panel (not required for the 400-1000°C and 600-1200°C temperature ranges); (3) type 3APT-53 protective fixtures, as well as auxiliary instruments (millivoltmeter MΠΙЩΡ-53, MCЩΡ-54, electronic potentiometers ЭΠΠ16 and ЭΠД02).

The telescope should be spaced at approximately 350-1500 mm from the radiation source.

The surface temperature of a body can also be measured with thermocolours, made from substances which change their colour when subjected to temperature variations.

Name	T	Brief characteristics
Name	Type	Brief characteristics
Indicating ratiometer with profile scale Electromechanical balanced	ЛПр-53	For measuring temperatures
bridge with automatic re- cording on a tape chart . Same with electric regulat-	АУМ	For measuring temperatures in one, three or six points
ing device	АУМР	For measuring and regulating temperatures in one or three points
Same with pneumatic regulating device	АУМРП	Same for liquid- or gas-fuel furnaces
Self-recording alternating- current automatic electron- ic bridge with disk chart	ЭМД-202	For measuring temperatures in one point without regulation
Same with electric three-po- sition regulating device	ЭМД-212	For measuring and regulating temperatures in one
Same with pneumatic iso- dromic regulating device	ЭМД-232	For measuring and regulating temperatures in liquid- and gas-fuel furnaces
Electronic automatic bal- anced bridge with position- al regulating device and		ard-and gas ract rathaces
tape chart	ЭМП-209	For measuring, recording and regulating tempera- tures in one, three, six or 12 points
Same with pneumatic iso- dromic regulator	ЭМП-209	For measuring, recording and regulating temperatures in liquid- or gas-fuel furnaces
		rumaces

Note. It is not recommended to use alternating-current instruments ЭМД type wherever there is a possibility of inducing stray currents in the measuring circuits. Direct-current bridges, marked ЭМД-102, ЭМД-112 and ЭМД-132, should be preferred.

Thermocouple Characteristics

.	uo	Designatio	Ľ	X	ΧK		ΧX	MK	
Compensating leads	Material and colouring	Electronegative	Alloy "ΤΠ" (green)	Constantan (brown), alumel (black)	Copel (yellow)		Copel (yellow)	Copel (yellow)	_
CO	Material	Electropositive	Copper (red)	Copper (red), chromel (vio- let)	Chromel (violet) Copel (yellow)	Plain electric cord	3.2 Iron (white)	Copper (red)	
ш	eter d, m	Wire dlam mostly use	0.5	3.2	3.2		3.2	3.2	
Jge, °C	per	Short- turn use	1600	1300	800	1	800	200	_
ture rar	Upper	Prolonged use	1300	1000	009	1000	009	350	_
Temperature range,		Lower	 0	0	0	300	0	0	_
		Negative	Platinum	Alumel	Copel	CA	Copel	Copel	_
Material		Positive	Platinum- rhodium	Chromel	Chromel	НК	Iron	Copper	_
		Туре	1111	XA	XK	HK-CA	ЖК	MK	

Note. Thermocouples of the HK-CA type are operated without correction for the cold junction.

Table 139

Brief Characteristics of Temperature Control Instruments Coupled to Thermocouples or Radiation Pyrometer Telescope

Туре	Brief characteristics	Substitut- ed for types
МПП-054 МПЩпл-54	For measuring temperatures For measuring temperatures. Not to be coupled to thermo-	ГНКП, МП-08 ГНЗС, МС-08
МПЩпр-54	telescope type PII For measuring temperatures	ПГУ, МПБ-46
МСЩпр-154	For measuring and recording temperatures in one point	Cr-1
	Same in six points For measuring and regulating tempera-	СГ-3 СГ-6 КГ, ЭРМ-47
СП-1 СП-3 СП-6	For measuring and recording temperatures in one, three	
СПР	For measuring and rec- ording tempera- tures in one or three points	
СПРП	For measuring and recording temperatures in liquid- or	
ЭПД-02	For measuring temperatures in one point	эпд-07
ЭПД-12	For measuring, recording and regulating temperature in one point	ЭПД-17
	МПП-054 МПЩпл-54 МПЩпр-54 МСЩпр-154 МСЩпр-654 МРЩпр-654 СП-1 СП-3 СП-6 СПР	МПП-054 МПШпл-54 МПШпл-54 МПШпр-54 МСШпр-154 МСШпр-154 МСШпр-154 МСШпр-154 МСШпр-154 МСШпр-354 МСШпр-354 МСШпр-654 МСШпр-654 МРШпр-54 СП-1 СП-3 СП-6 СП-6 СПРП Помазитіпд Темазитіпд Темазитіп Темазитіп Темазитіп Темазитіп Темазитіп Темазитіп Темазитіп Темазитіп Тема

Instrument	Type	Brief characteristics	Substitu- ted for types
Same with pneumatic isodromic controlling device	ЭПД-32	For measuring, recording and regulating temperatures in liquid or gas-fuel furnaces	эпд-37
Electronic potentio- meter with rotating scale and stationary temperature indica- tor	ЭПВ-01	For measuring tem- peratures	ЭПУ-18 and ЭПУ-28
Automatic electronic potentiometer with electric positional controlling device and recording on a tape chart	ЭПП-09	For measuring, recording and regulating temperatures in one or three, six and twelve points	
Same with pneumatic isodromic control- ling device	′ЭПП-09	For measuring, recording and regulating temperatures in liquid- or gas-fuel furnaces	

Table 140
Radiation Pyrometer Characteristics

Conventional designation	Name	Scale, °C	Tolerated error		
ОППИР-09 ОППИР-20-55 ОППИР-30-55	Optical pyrometer Ditto Ditto	1. 800-1400 2. 1200-2000 1. 800-1400 2. 1200-2000 1. 1200-2000 2. 2001-3000	±21 ±30 ±30 ±50		

Note: The pyrometers are to be spaced at 0.7 m and over from the heat source to be measured.

Telescope	Measurement	Tolerated erro)r	Gradua.
optics	ranges, °C	Temperature range	Error	tion
Quartz glass Glass K-8 Same	400-1000 * 600-1200 700-1400 900-1800 1100-2000 1200-2200	400-699 700-899 900-1099 1100-1400 900-1099 1100-2000 1200-2000	$\pm 12 \\ \pm 14 \\ \pm 18 \\ \pm 22 \\ \pm 18 \\ \pm 22 \\ \pm 22$	Гр.Р1 Гр.Р1 Гр.Р2 Гр.Р3
<u> </u>	1400-2550	2001-2200 2201-2500	±24 ±28	rp.P4

^{*} Temperature ranges not specified by Soviet Standards.

Each thermocolour changes its colour at a definite temperature which is a function of its chemical composition. Thermocolours are intended to measure surface temperatures from 40° to 580°C on open heating.

As practice shows, temperatures can also be determined by heat colours. It should be borne in mind that a change in shop lighting may be the cause of considerable errors when this method is applied (Table 142).

Table 142
Approximate Measurement of Temperatures by Heat Colours

Heat colour										Approximate temperature, °C				
														530-580
										÷	:			580-650
		٠,									i			650-730
											•			730-770
			٠.									:		770-800
١.														800-830
														830-900
														900-1050
													,	- 1050-1150
														- 1150-1250
			· · · · · · · · · · · · · · · · · · ·											_ , , , , , , , , , , , , , , , , , , ,

Table 143
Ink Composition for Recording Instruments

Material	Quantity, %
Pure glycerine (spec. gr. from 1.21 to 1.26) Pure alcohol (spec. gr. from 0.79 to 0.81) Distilled water	22.9 7.15 68.70 0.1 1.15

Note. The following chemicals are used as dyes: methyl violet, methylene blue or light blue, eosin, orcein, etc.

Method of preparation: the colouring agent is dissolved in alcohol with gradual addition of water by portions. Glycerine and glue are heated till they mix perfectly. The cooled mixture is added by portions to the dissolved colouring agent while mixing continuously. After complete mixing of glycerine, glue and colouring agent, the mixture is allowed to stand for 4-5 hours, after which it is filtrated.

Accessory materials, used in heat-treatment, are given in Table 144,

Table 144
Approximate Consumption of Some Auxiliary Materials
Used in Heat-treatment

Material	Consumption in grams per 1 kg of treated articles	Remarks
Hardening Common salt, etc	10	Heating in the 800-1000° C
Barium chloride	. 20	range Same for temperatures above 1000° C
Spindle oil	20-30 1	Cooling medium Cooling medium
Tempering		
Potassium or sodium nitrate	30-40 10 20 10	For heating

Table 144 continued

Material	Consumption in grams per 1 kg of treated articles	Remarks
Carburising Solid carburiser	60, 100	
Kerosene	60-100 20-30 20-30 20-30	For shaft furnace Same Production of pyrolisis
Solar oil	8	gas Purification of pyrolisis gas
Nitriding		
Ammonia	5-50	The smaller the size of articles treated, the greater the consumption
Cyaniding		
Cyanide compound "ГИПХ" Sodium cyanide Calcium chloride Sodium chloride (common	20 20 50	
salt)	30 20	Neutralising
Cleaning	[
Sand, size of particles from 1 to 1.5 mm Shot ø 0.8-1.0 mm	50-80 Up to 1	
Pickling	,	
Sulphuric or hydrochloric acid	30-40	
Crack checking		
Crocus	20	

· Chapter XII

SAFETY ENGINEERING

SAFETY REGULATIONS FOR FUEL-OIL FURNACES

The following procedure is to be adhered to when starting up a fuel oil furnace: (1) scavenge the furnace with air to evacuate accumulated explosive gases; (2) introduce an electric-ignition device or a lighted torch on a long, slim rod into the firing chamber, then open, first, the air valve, then the fuel valve; (3) after the stable flame has been obtained, remove the device and increase gradually fuel and air feed.

The lid of the charging door should be kept open during the lighting

operation.

Do not peer into the firing chamber. Avoid drips and pools of fuel oil underneath the burner and keep the latter in perfect condition.

In case of an abrupt cut in fuel oil or air feed, as well as after the termination of operations, first shut off the fuel valve, then the air valve.

SAFETY REGULATIONS FOR GAS FURNACES

Industrial gases are dangerous because of toxicity. When mixed with air they present explosion hazards.

Methane content in the mixture	Behaviour of the mixture on ignition
Up to 4% From 4 to 15% Above 15%	Neither burns nor explodes Does not burn, but explodes Rurns but does not explode

Explosive range of an air-gas mixture may vary depending on composition, moisture and temperature. Leaks should not be searched for with the aid of a torch. An aqueous solution of soap should be employed for this purpose.

The following procedure is to be performed when firing a gas fur-

пасе:

l Open the slide valve and the lid of the charging door, switch on air supply, and scavenge the furnace with air for 5 minutes, after which cut the air supply off.

2. Open the main gas valve, and scavenge the gas main for 5 minutes, discharging the gas through a riser.

3. Ignite the gas primer and gradually switch on the gas supply.
4. Adjust the combustion of the fuel in the gas burner by varying

air supply.

5. When igniting one should stand on one side of the burner; glass goggles are to be on to protect the eyes. When laying off a furnace, cut the air supply, first, then turn off the gas.

It should be borne in mind that 0.1% of hydrogen sulphide and

over 0.002% of carbon dioxide cause serious poisoning.

FIRST AID IN GAS POISONING

When a person has been gas-poisoned it is imperative to take the injured person out of premises to the fresh air, unbutton his clothes and not let him sleep until medical personnel arrives. If the person does not breathe, artificial respiration should immediately be applied.

RULES FOR SAFE OPERATION OF SALT BATHS

Operation of salt baths using melted salts, alkalis or lead requires strict keeping to all safety regulations.

1. Ventilation systems should be inspected prior to starting-up op-

eration.

- Goggles and gloves are obligatory when operating the salt baths.
 It is strictly forbidden to put one's head under the draught hood.
- 4. Lead or salt should be added to the bath in small portions; the added salt should be well dried, while lead should be thoroughly preheated.
- 5. Parts and hardening devices should be dipped into crucibles filled with salt or lead only after being completely dried and preheated to 100-150°C. Cold parts may cause eruptions of salt or lead from crucibles, which may burn the operator.

6. Spilled salt or lead should be covered with dry sand. Flooding

with water is prohibited.

7. Saltpetre ignites on heating above 550°C. A combination of saltpetre, charcoal, soot and grease leads to explosions. The use of saltpetre should be avoided. When the use of above components is unavoidable, controlling apparatus with sonic signalling devices are imperative.

8. Special attention should be given to the drying of a salt electrode furnace after relining. Inadequately dried furnace may explode after

a few hours' operation, splashing all salt out.

RULES FOR SAFE OPERATION OF CYANIDE BATHS

Cyanide salts are most violent poisons. The combination of cyanide salts with acids produces a very poisonous gas—the prussic acid. Only well-trained operators should be permitted to operate cyanide baths.

One should adhere strictly to the following safety rules when working with the cyanide bath-type furnaces:

1. Check the ventilation prior to operation. The operator should not

start working if ventilation defects are detected.

2. Furnace shell doors should be opened only when charging or discharging parts and adding salts.

3. It is forbidden to handle cyanide salts with bare hands.

- 4. Salt should be added to the melted bath by small portions, with the furnace doors closed, and only when well dried. Goggles, gloves and respirators should be used.
- 5. Hardened parts should be thoroughly water-flushed to remove traces of cyanide salts.

6. All tools stained with cyanide salts should be stored under

draught.

- 7. No saltpetre or potassium bichromate should be admitted to the melted bath, as this unavoidably causes explosions.
- 8. It is strictly forbidden to smoke or eat in the shop, as this presents mortal danger.
- 9. Prior to eating and smoking it is necessary to thoroughly wash hands and rinse the mouth.
 - 10. It is forbidden to wash hands in hardening and washing tanks.
- 11 In case the ventilation fails, shut the furnace off immediately, leave the shop and inform the managing personnel.

12. All scratches or wounds, however small, should be bandaged pri-

or to work.

13. If one is suffering from a cold or some other indisposition, medical care should be sought immediately.

RULES FOR DECONTAMINATING WATER FROM CYANIDE HARDENING AND WASHING SHOPS

Decontamination of 1 litre of water necessitates 3 grams of iron vitriol and 1 gram of soda ash (assuming sodium cyanide content averages up to 0.1%).

The operation should be carried out as follows:

1. Prepare the necessary amount of iron vitriol and soda, in conformity with the tank capacity.

2. Dissolve iron vitriol in water and add soda.

- 3. Pour the prepared mixture into the hardening or washing tank and mix it thoroughly.
- 4. Drain decontaminated water, simultaneously opening the water tap to bring the concentration still further down.

SAFETY RULES FOR OPERATION OF EQUIPMENT

1. Fuel oil or grease spilled on the floor should be covered with generous amounts of sand and swept away.

2. Prior to work sleeve cuffs should be tied up and dangling bits tucked away,

3. No moist blanks should be charged as this may cause violent flame eruptions.

4. When quenching slim articles with through holes, orient the latter away from shop personnel to avoid burns from vapour or oil.

- 5. Do not switch on bath-heating appliances when aqueous solutions are thick or frozen (blueing solution, etc.), as eruptions of solutions and burns would be unavoidable.
- 6. When quenching articles with compressed air the operator should put the goggles on and interpose a wire screen protecting him from the rebounding scale.

7. Water should be poured first into sulphuric acid pickling tanks;

sulphuric acid is added then in a thin jet.

8. Undried parts, tongs, hooks should not be dipped into a tempering oil bath while the latter is hot.

SAFETY REGULATIONS FOR HANDLING ELECTRICAL EQUIPMENT

1. Moist clothes or foot-wear, damp floor or mats, etc. increase electric' shock hazard.

2. Only well-grounded electrical equipment is authorised for opera-

tion.

3. Switches and starting buttons should not be switched on by means of pokers, tongs and similar metallic objects.

Rubber mats are imperative when operating electric furnaces.

FIRE PROTECTION

1. Pans with sand should be available underneath the burners when liquid fuel is being used.

2. Sand and spade from the fire-extinguishing box should never be

used for operation needs.

3. Fuel-oil storage tanks should be situated far from fire.

4. It is strictly forbidden to heat up the fuel-oil mains with a torch or a blow torch, or to heat fuel oil, stocked in storage tanks, by dipping hot pieces of metal.

5. All quenching and tempering oil baths should be provided with

lids to prevent air access, if the oil ignites.

6. On hardening it is forbidden to heat kerosene above 38°C.

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TO THE READER

Peace Publishers would be glad to have your opinion of the translation and the design of this book.

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